

***Mesoscale simulations of stimuli-sensitive
polymer networks***

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Polymer Networks

□ Types of polymer networks

- ☞ Natural (cytoskeletal structures)
- ☞ Synthetic (hydrogels)

□ Properties of polymer networks

- ☞ Highly permeable (porosity $\sim 0.75 - 0.98$)
- ☞ Extremely flexible (elastic modulus $\sim 1 - 103$ kPa)
- ☞ Mechanically sturdy (support external loads)
- ☞ Sensitive to external stimuli (light, pH, T, etc)

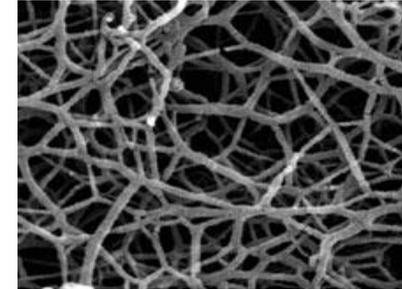
□ Applications

- ☞ Smart and responsive materials
- ☞ Drug delivery
- ☞ Tissue engineering

□ Goal

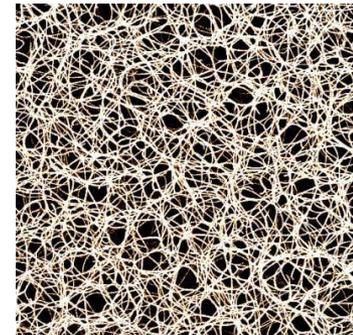
- ☞ Develop mesoscale model that can capture mechanical and transport properties of responsive polymer networks

Synthetic fibrous network

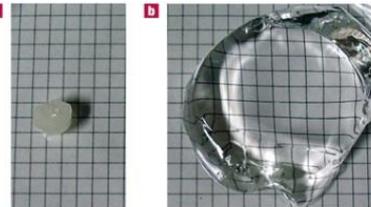


Tysseling-Mattiace *et al.*,
J. Neuroscience, 2008

Actin network



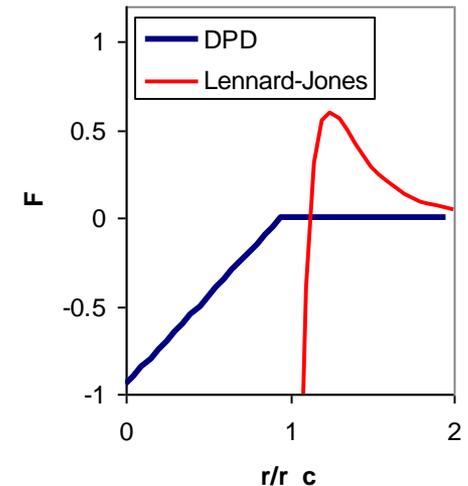
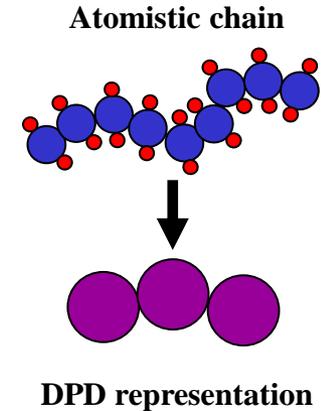
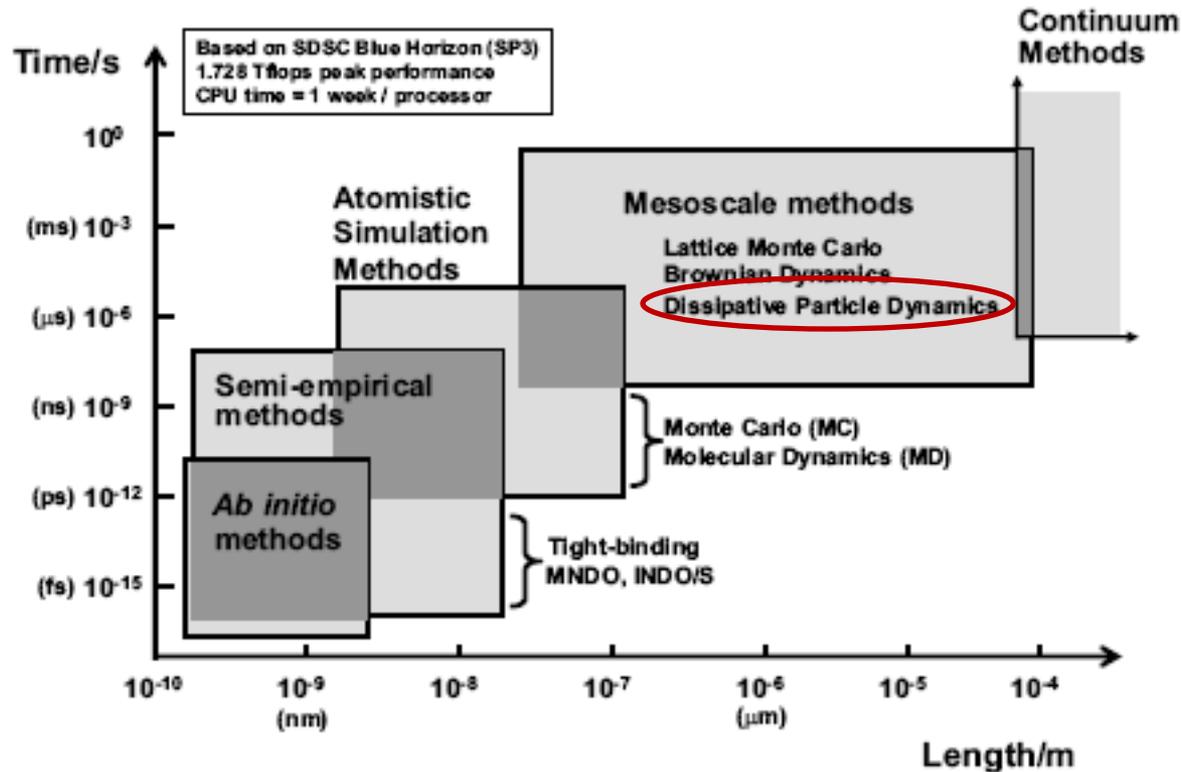
Gel swelling



Ono *et al.*,
Nature Materials, 2007

Schmoller *et al.*,
Nature Communications, 2010

Simulation Approach



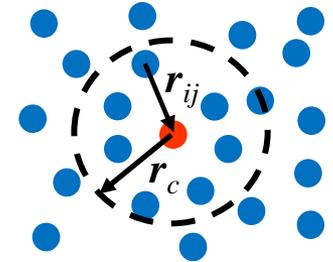
□ Dissipative Particle Dynamics

- ☞ Use soft potential to represent cluster of atoms
- Allow simulations with larger time/length scales

Dissipative Particle Dynamics (DPD)

□ Newtonian time evolution of many-body system (similar to MD)

$$\frac{d\mathbf{r}_i}{dt} = \mathbf{v}_i \quad \frac{d\mathbf{v}_i}{dt} = \sum (F_{ij}^C + F_{ij}^D + F_{ij}^R) \hat{\mathbf{r}}_{ij}$$



☞ Repulsive force

– Accounts for compressibility

$$F_{ij}^C = a_{ij} \left(1 - r_{ij} / r_c\right)$$

☞ Dissipative drag force

– Mimics viscosity

$$F_{ij}^D = -\gamma \left(1 - r_{ij} / r_c\right)^2 \left(\hat{\mathbf{r}}_{ij} \cdot \mathbf{v}_{ij}\right)$$

☞ Stochastic force

– Represent thermal fluctuations

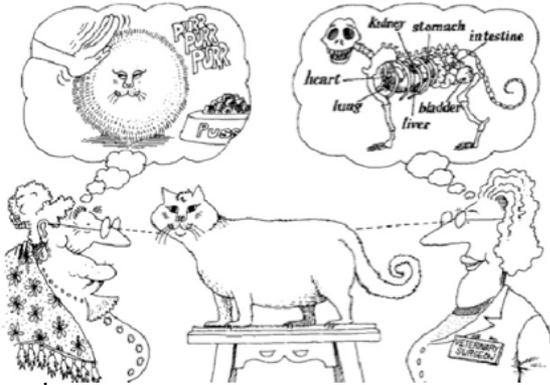
$$F_{ij}^R = \sqrt{2\gamma k_B T} \left(1 - r_{ij} / r_c\right) \zeta_{ij} / \sqrt{\Delta t}$$

□ Pair-wise and central forces to preserve hydrodynamics

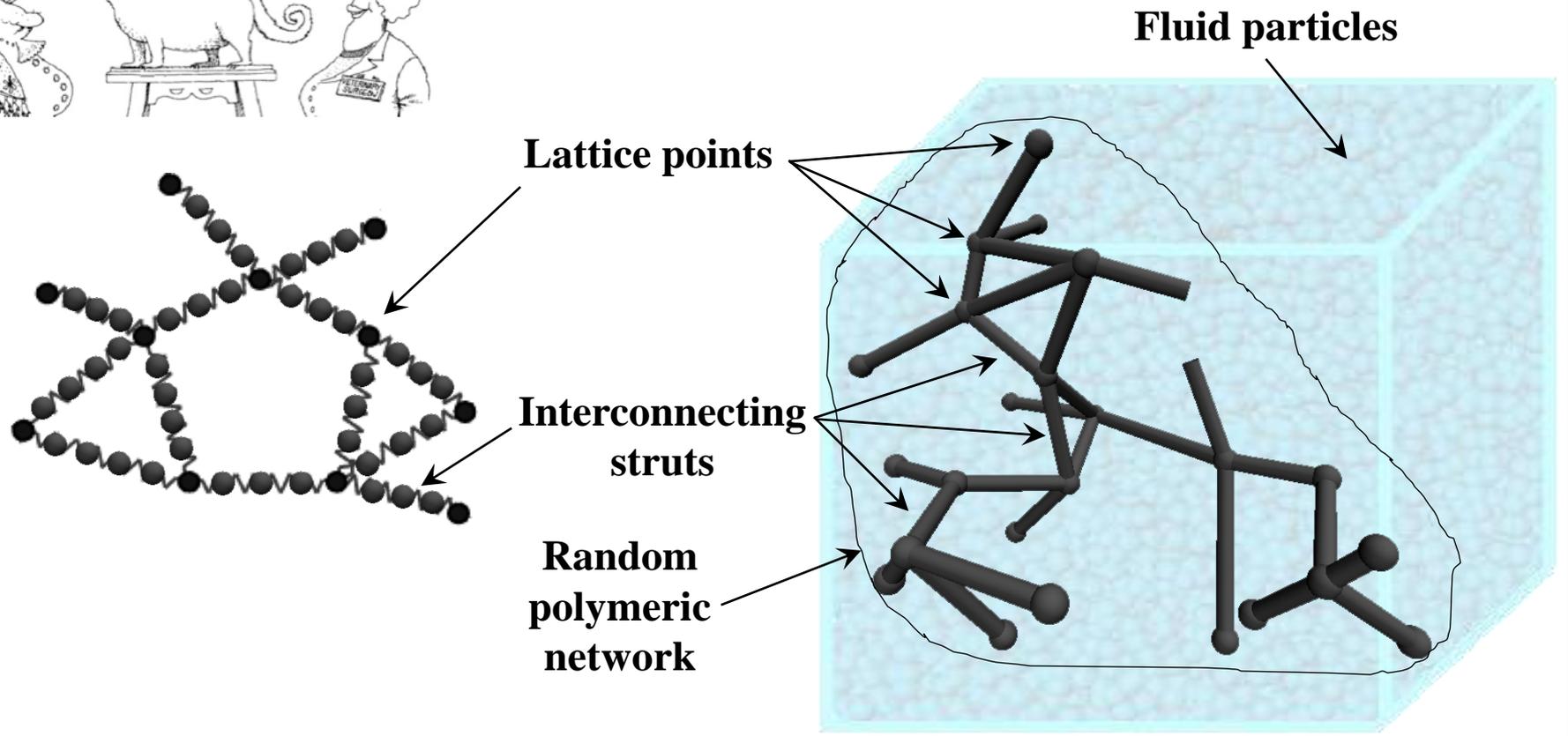
□ Departure from Molecular Dynamics (MD) :

☞ Soft conservative interaction potential

☞ Coarse-grained in time and space (“fluid elements”)



Mesoscale Model of Polymer Network



- ❑ **Model polymeric networks as random lattice of interconnected filaments**
 - ☞ Replicates microscopic architecture at mesoscale level

Polymer Network Model

□ Network is created in two steps

- ☞ Randomly distribute N cross-linking nodes in simulation box
- ☞ Connect each node to the C closest nodes

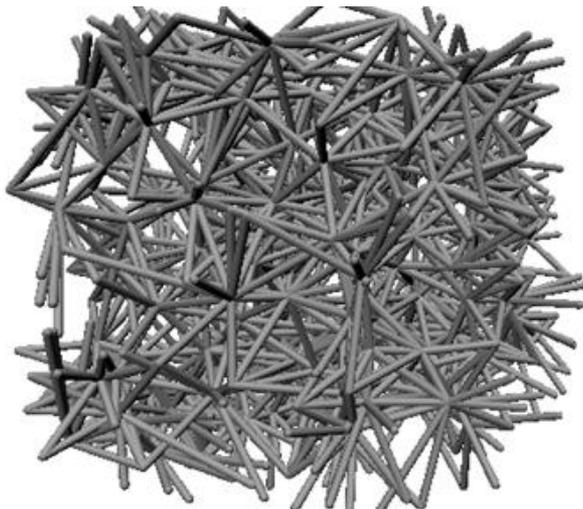
N = Number of cross-linking nodes

C = Average connectivity (cross-linking density)

$$\text{Porosity } (\varepsilon) = \frac{\text{Volume of voids}}{\text{Total volume}}$$

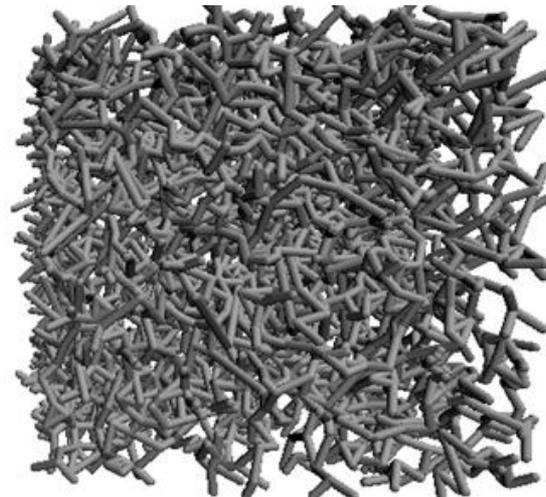
$N = 250$

$C = 12$



$N = 2000$

$C = 3$



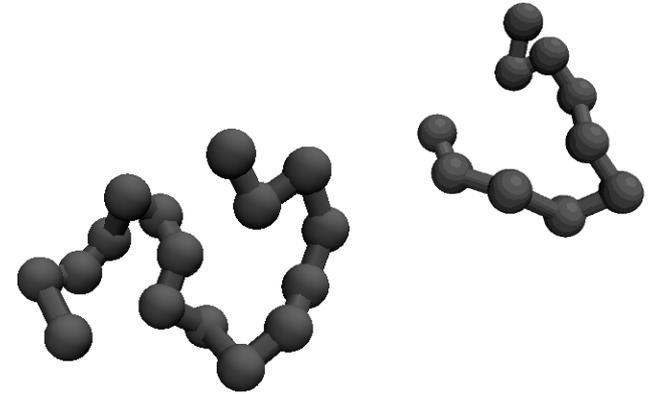
$\varepsilon \approx 0.81$

□ Accurate control over network properties

Macromolecules and Nanoparticles

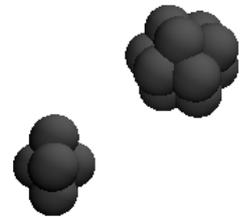
❑ Bead-spring model to model polymer chains

- ☞ FENE spring to model flexible polymers
- ☞ Harmonic stretching and angle (triple) potentials to model semi-flexible polymers



❑ DPD beads with fixed relative position to model rigid particles

- ☞ Size is characterized by effective hydrodynamic radius

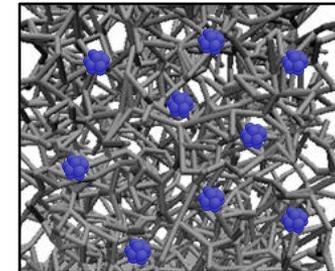
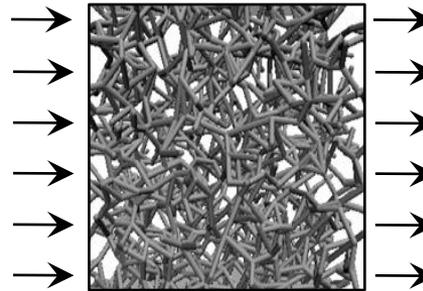


❑ DPD potentials account for interactions with surroundings

Transport Through Polymer Networks

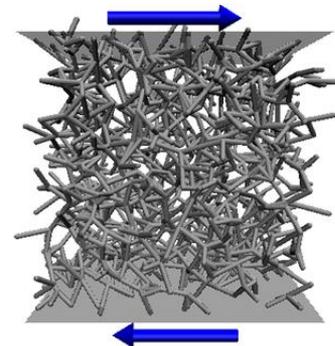
□ Study transport through polymer networks

- ☞ Permeation
- ☞ Self-diffusion
- ☞ Particle diffusion
- ☞ Chain diffusion



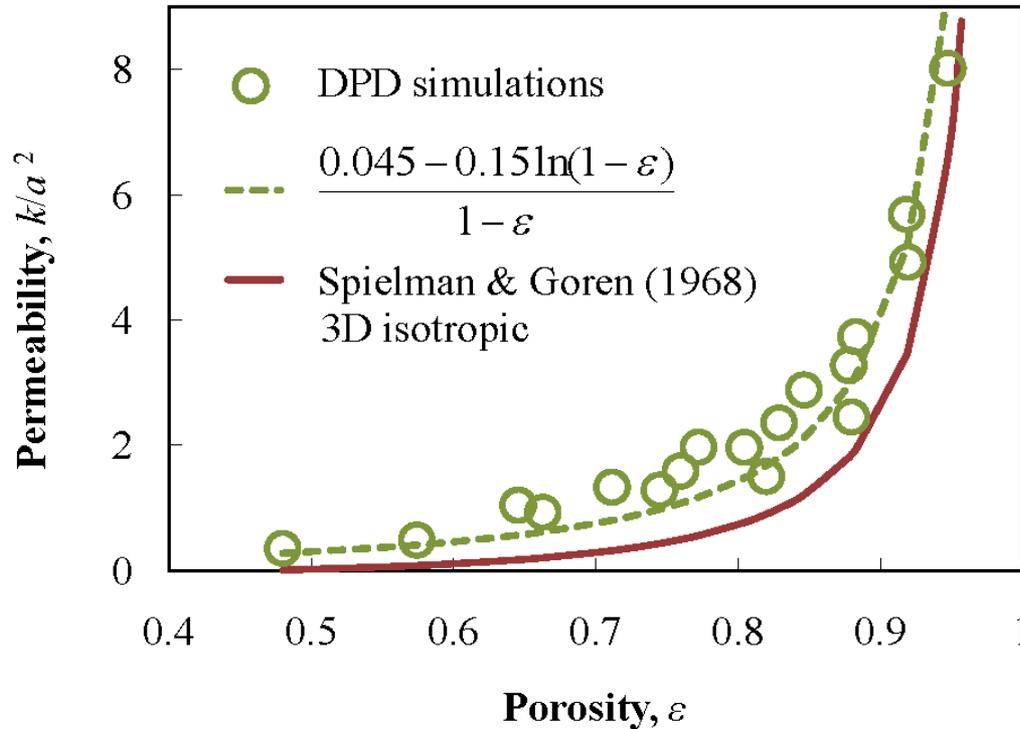
□ Quantify dependence of transport properties on network geometry

□ Probe effect of mechanical deformation on permeation and diffusion

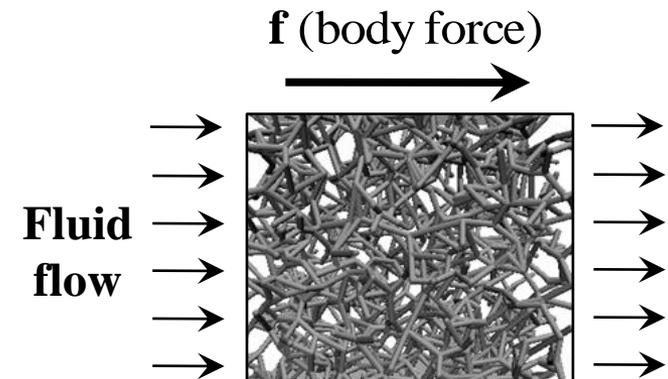


Network Permeability

a = Radius of filaments



$$\varepsilon = \frac{\text{Volume of voids}}{\text{Total volume}}$$

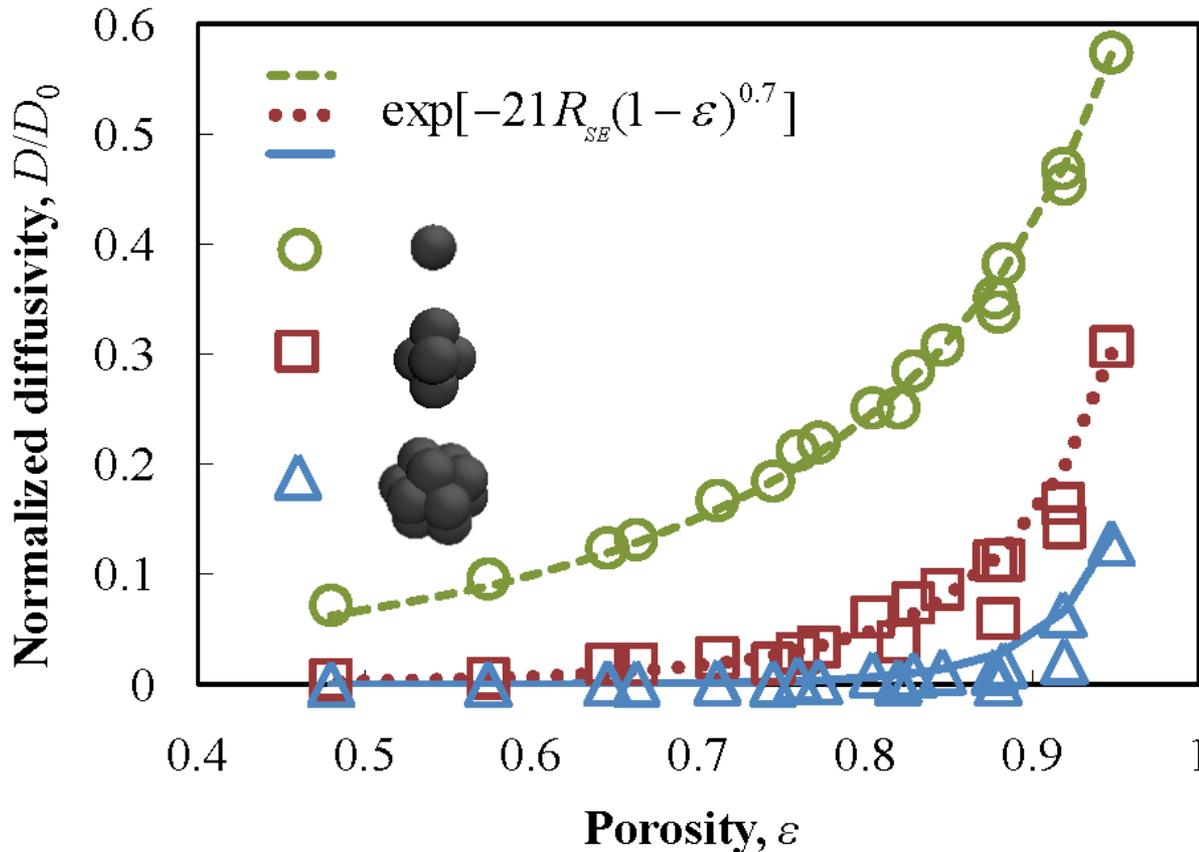


$$\text{permeability } (k) = \frac{\mathbf{q} \cdot \mathbf{v}}{\mathbf{f}}$$

□ **Normalized permeability is independent of network internal structure**

- ☞ Solely function of fraction of void volume (porosity)
- ☞ Good agreement with theory and experiment

Diffusion of Rigid Particles



$$\frac{D}{D_0} = \exp[-\alpha R^\delta (1-\varepsilon)^\nu]$$

Phillies, J. Chem. Phys., 1989

Phillies *et al.*, J. Chem. Phys., 1989

$$\frac{D}{D_0} = \frac{D_{\text{with network}}}{D_{\text{without network}}}$$

$$\nu = 0.7$$

experiments of
Seiffert & Oppermann,
Polymer, 2008

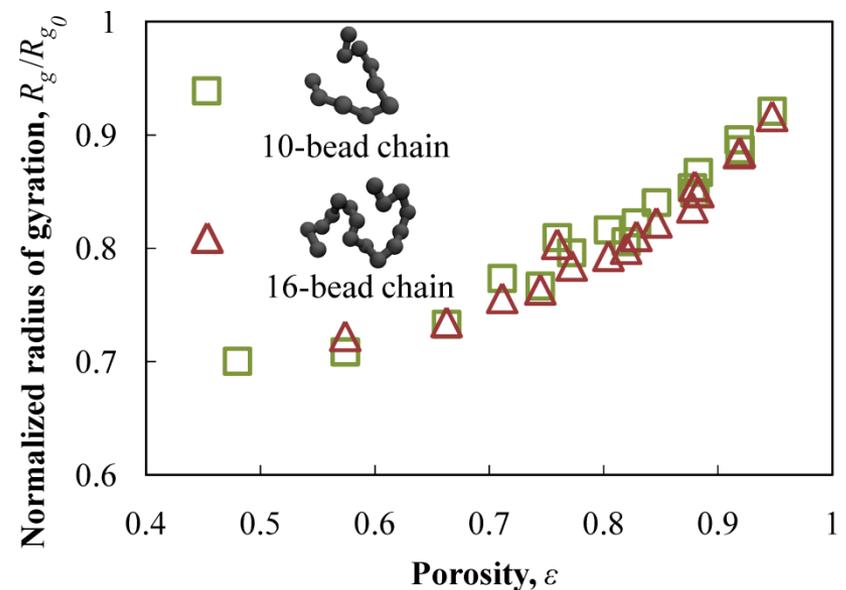
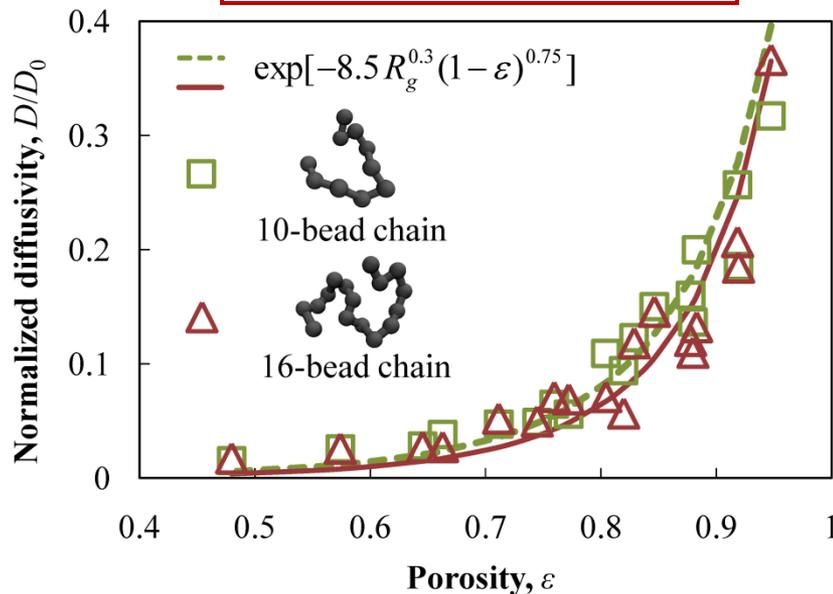
- ❑ Normalized diffusivity is also solely function of porosity
- ❑ Good agreement with hydrodynamic scaling model and experiment

Diffusion of Polymer Chains

$$\nu = 0.8$$

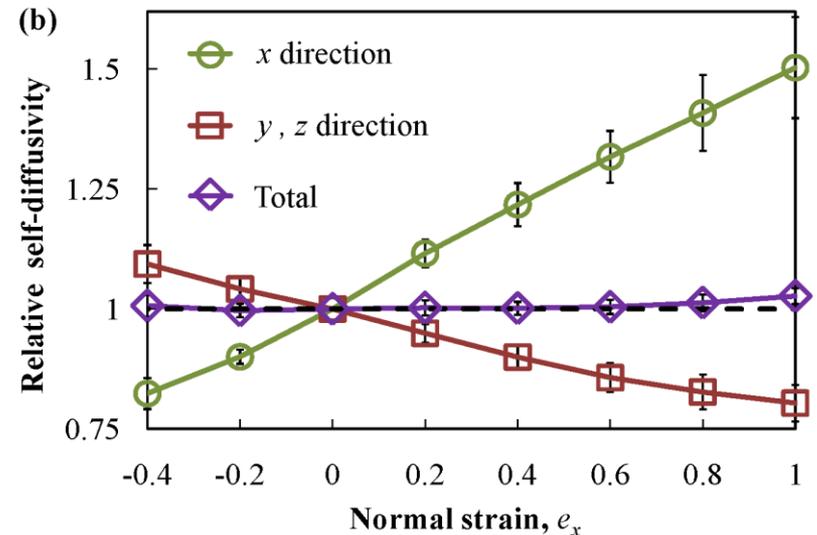
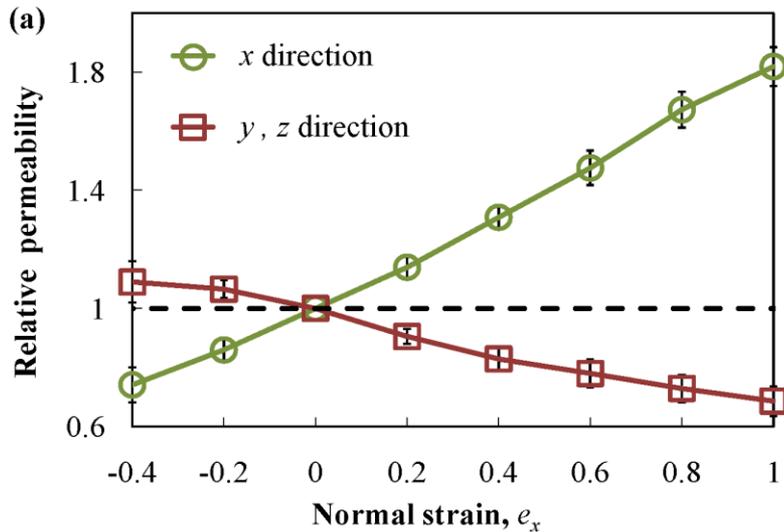
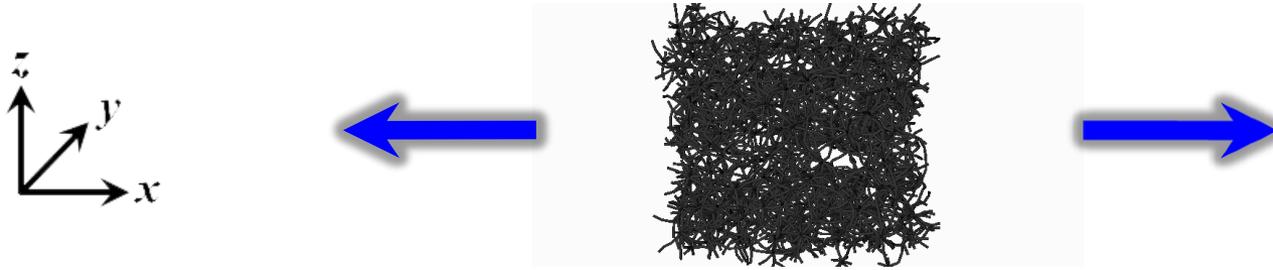
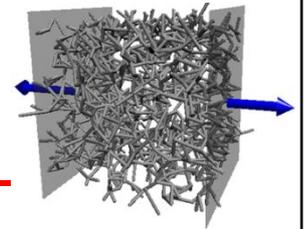
experiments of
Seiffert & Oppermann,
Polymer, 2008

$$\frac{D}{D_0} = \frac{D_{\text{with network}}}{D_{\text{without network}}} = \exp[-\alpha R^\delta (1-\varepsilon)^\nu]$$



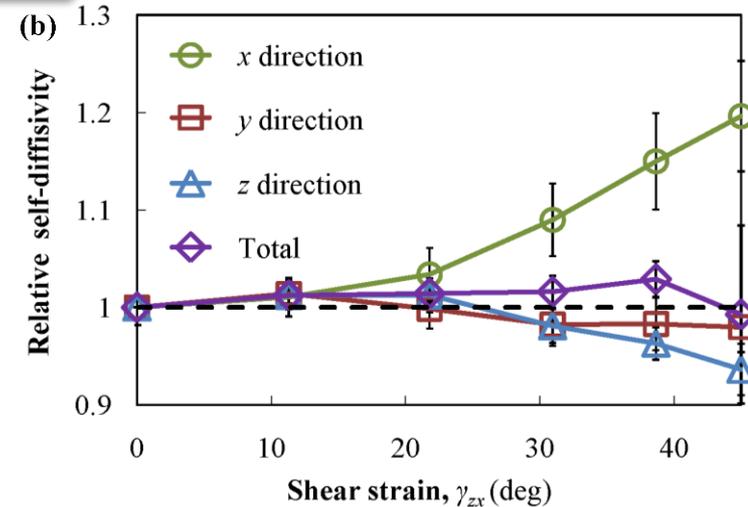
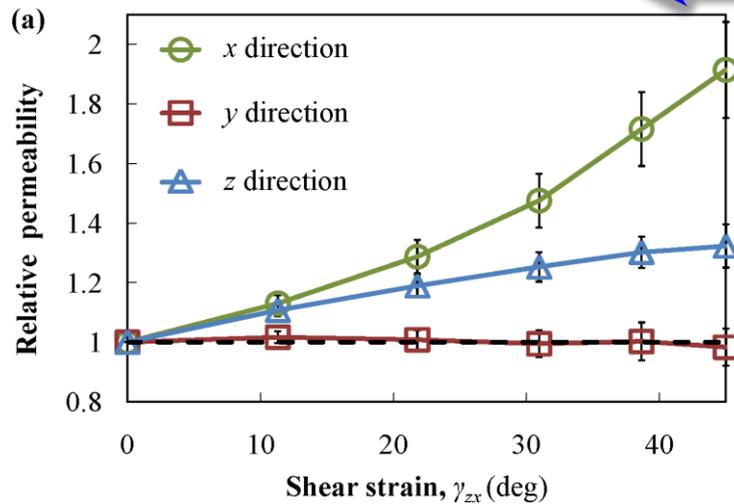
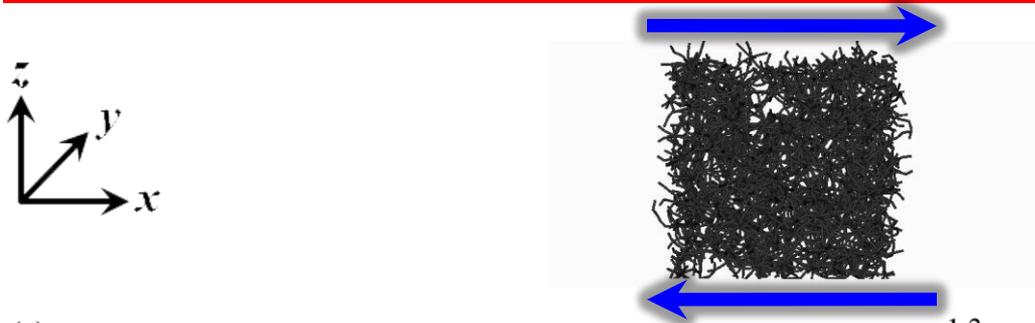
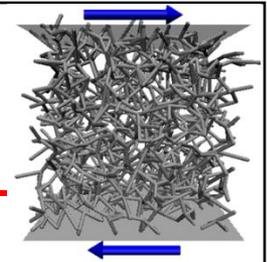
- ❑ Good agreement with theory and experiment
- ❑ Smaller diffusion coefficient for particles comparing to polymer chains
 - ☞ Change in radius of gyration of polymer chains

Transport Under Axial Deformation



- ❑ Data averaged over 18 networks ($N=250,500,1000$ & $C=4,6,8,10,12,14$)
- ❑ Stretching enhances permeation and diffusion in direction of deformation
 - ☞ Opposite effect in transverse direction
- ❑ Total diffusivity remains unchanged under axial deformation

Transport Under Shear Deformation

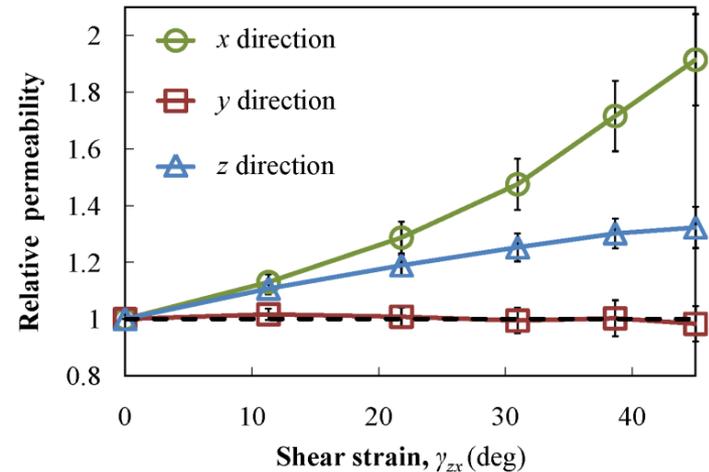
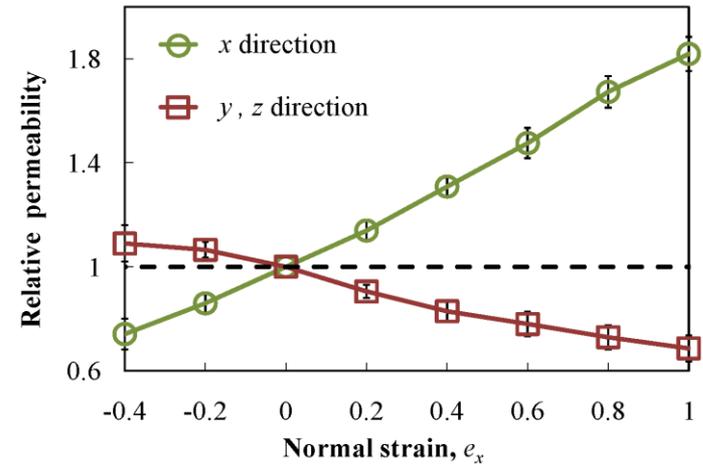
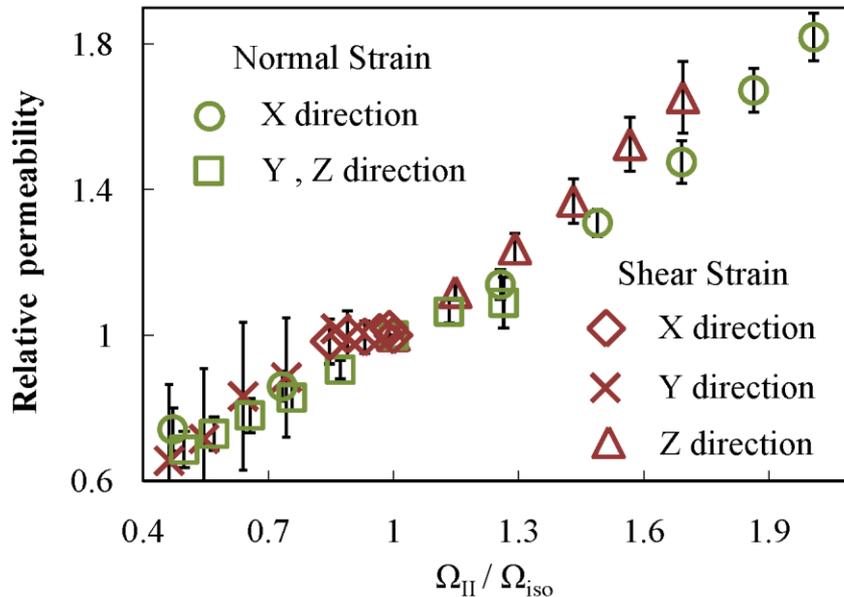


- ❑ Data averaged over 18 networks ($N=250,500,1000$ & $C=4,6,8,10,12,14$)
- ❑ Permeability in y direction remains unchanged under shear deformation
 - ☞ Filaments rotate in xz plane
- ❑ Total diffusivity remains unchanged under shear deformation

Permeability in Principal Direction

- Characterize degree of alignment by second order orientation tensor

Relative permeability in principal directions of orientation tensor



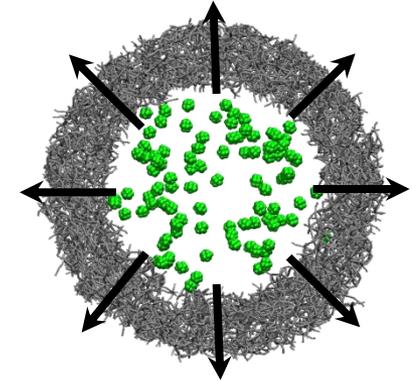
- All data collapses into single master curve

↳ Nearly linear dependence on magnitude of orientation tensor eigenvalues

Release From Responsive Capsules

□ Release from drug delivery microcapsules

- ☞ Controllable
- ☞ Non-destructive
- ☞ Sensitive to external stimuli

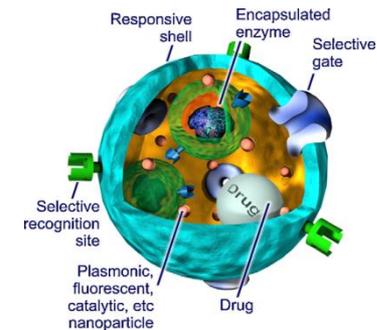


□ Study release from responsive microgel capsules

- ☞ Swelling/deswelling volume transitions

□ Mesoscale model for responsive polymer networks (gels)

- ☞ Micromechanics of polymer network
- ☞ Explicit solvent
- ☞ Diffusive and advective transport
- ☞ Swelling/deswelling volume transition

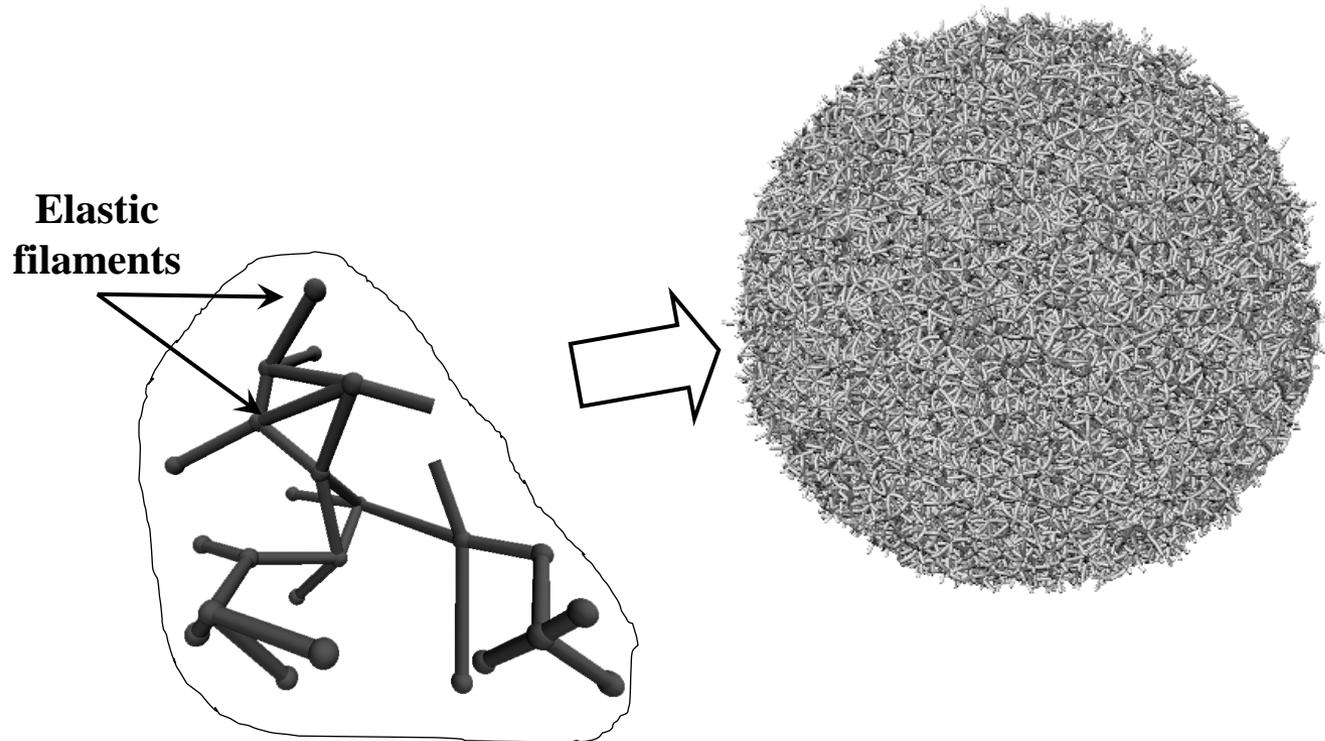


Motornov *et al.*
Prog. Polym. Sci., 2010

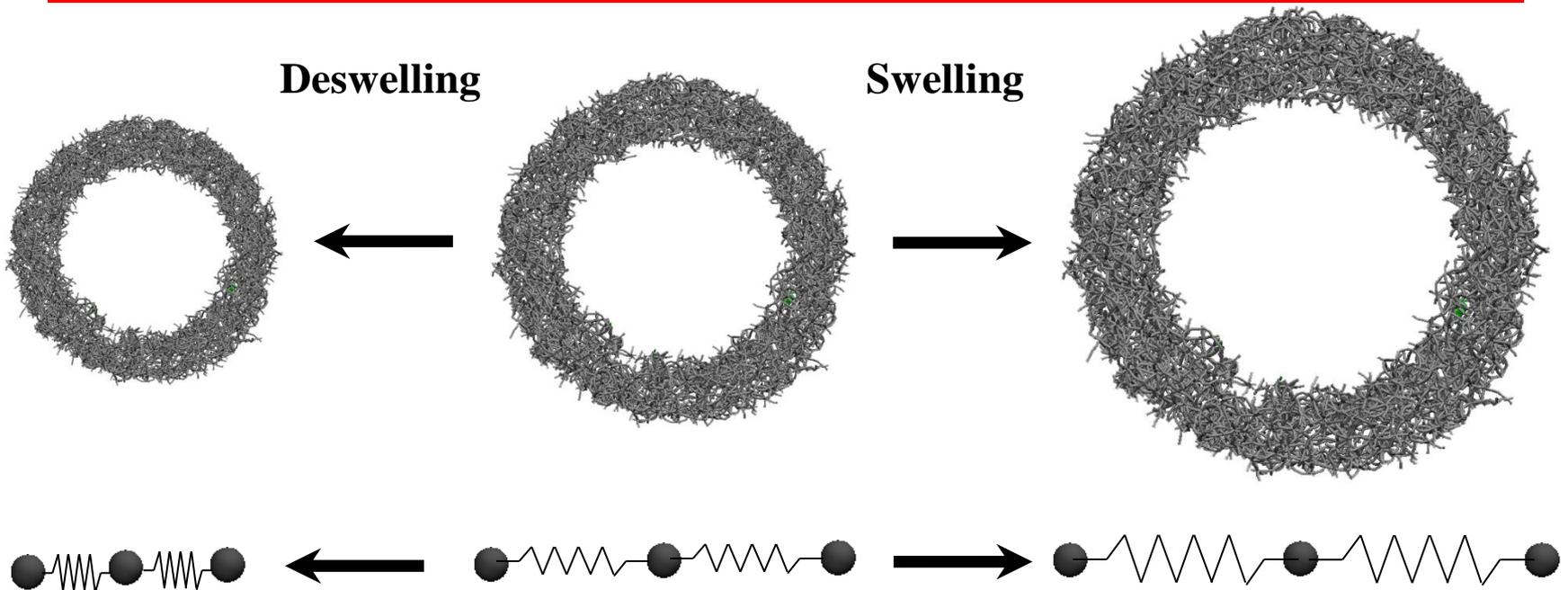
Responsive Microgel

□ Random network of interconnecting elastic filaments

- ☞ Filaments of DPD beads connected by stretching and bending springs
- ☞ Spherical capsule is formed by trimming homogeneous network



Modeling Gel Volume Transition



□ **Network volume transition is modeled by varying equilibrium length of network filaments**

- ☞ Accounts for internal stresses that force network to shrink or expand
 - Increase equilibrium length of springs to model swelling
 - Decrease equilibrium length of springs and strength of DPD potentials to model deswelling

Swelling Kinetics of Spherical Capsules

- Swelling kinetics is set by balance between hydrodynamic and elastic forces

$$\tau_{sphere} = \frac{R_{final}^2}{\pi^2 D_0}$$

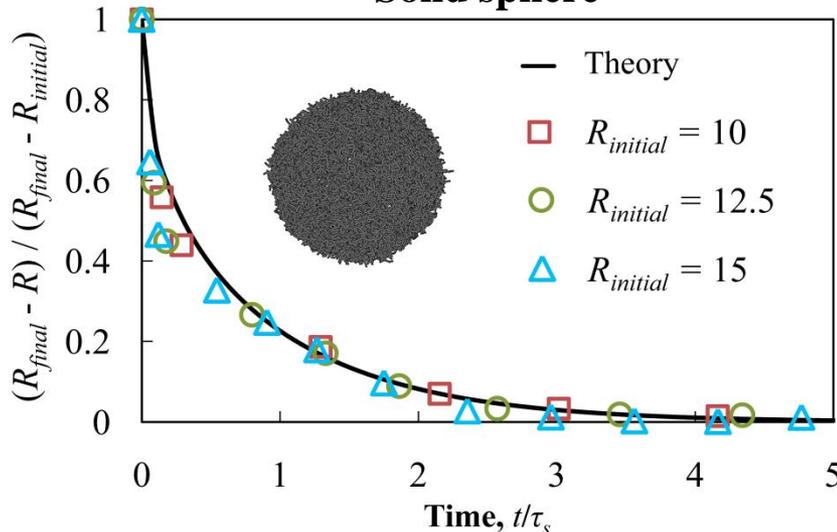
Tanaka and Fillmore,
J. Chem. Phys., 1979

Tanaka's model ($t > \tau$): $\frac{R_{final} - R}{R_{final} - R_{initial}} \cong \exp(t/\tau)$

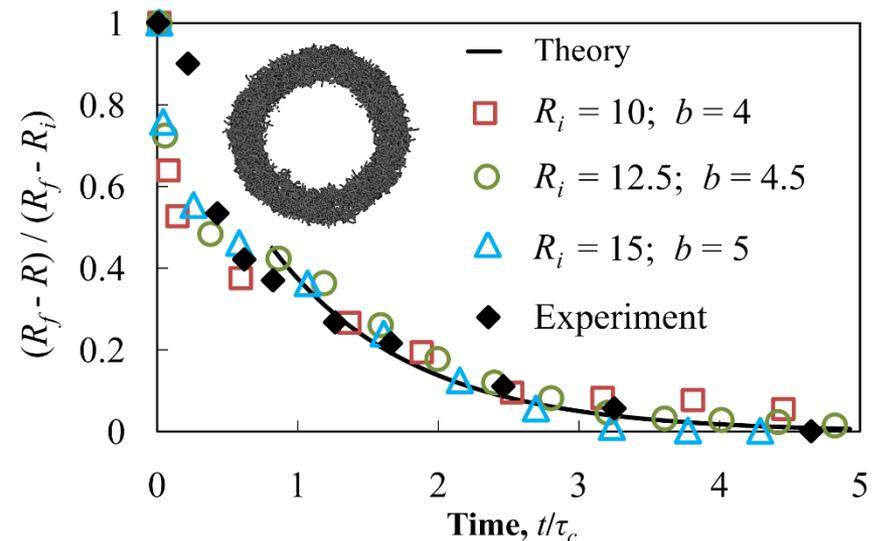
$$\tau_{capsule} = \frac{R_{final}^2}{9D_0}$$

Wahrmund *et al.*,
Macromolecules, 2009

Solid sphere



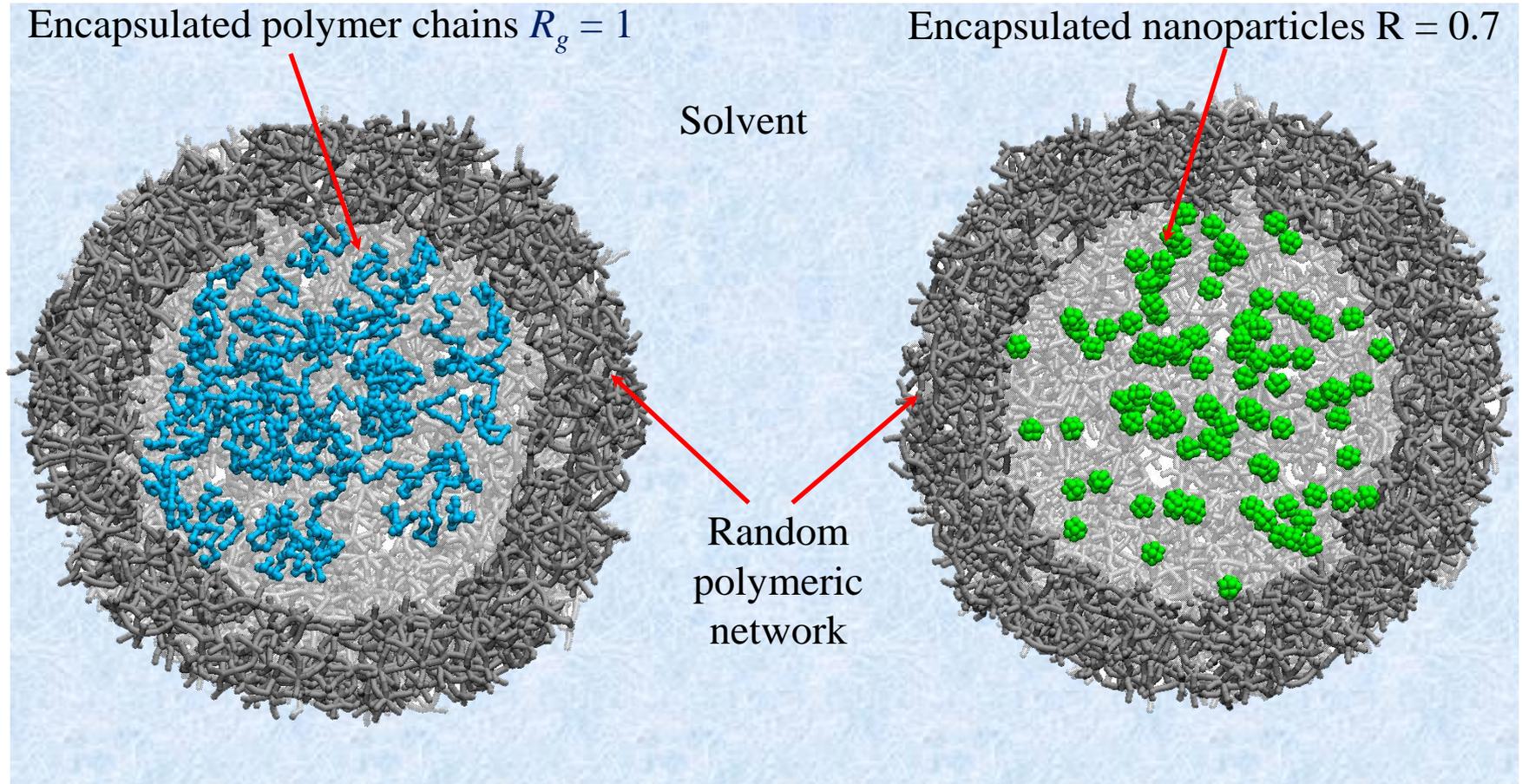
Hollow sphere



- Very good agreement with theoretical models

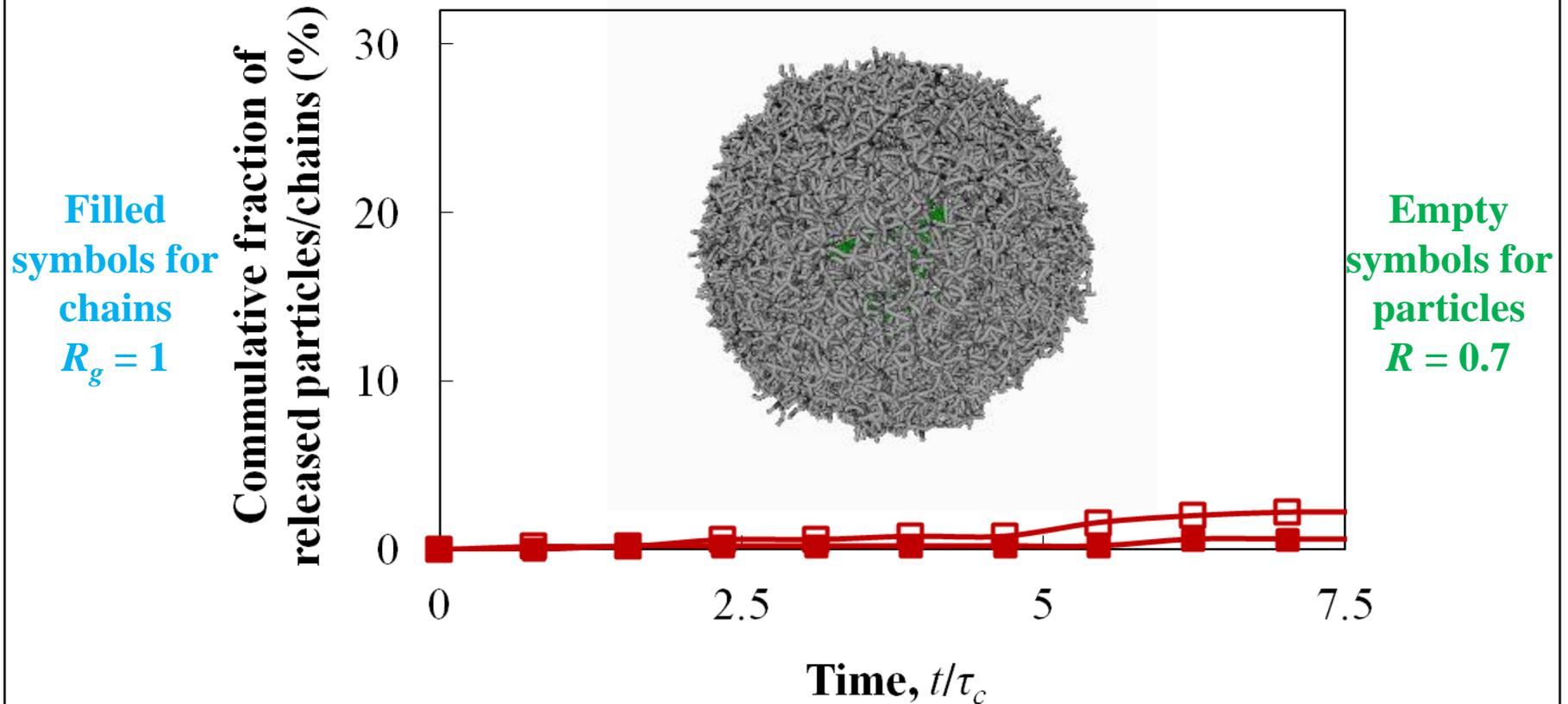
- Weak dependence of capsule relaxation time on the membrane thickness

Capsules Filled With Particles and Polymer Chains



- Probe release of macromolecules and nanoparticles from responsive microgel capsules

Release From Capsules at Initial Equilibrium

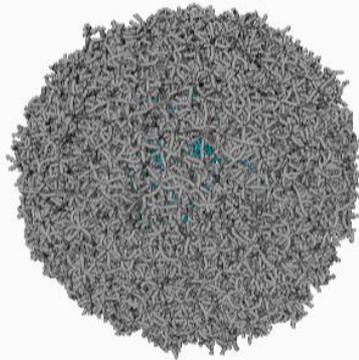


□ No release from capsule at initial state

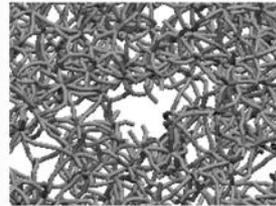
☞ Characteristic size of solutes is comparable with network average mesh size

Release From Swelling Capsules

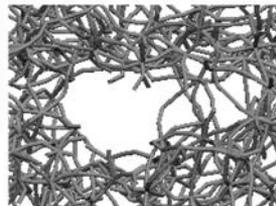
Capsule with macromolecules



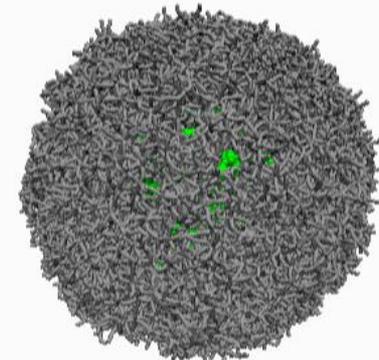
Initial



Swollen



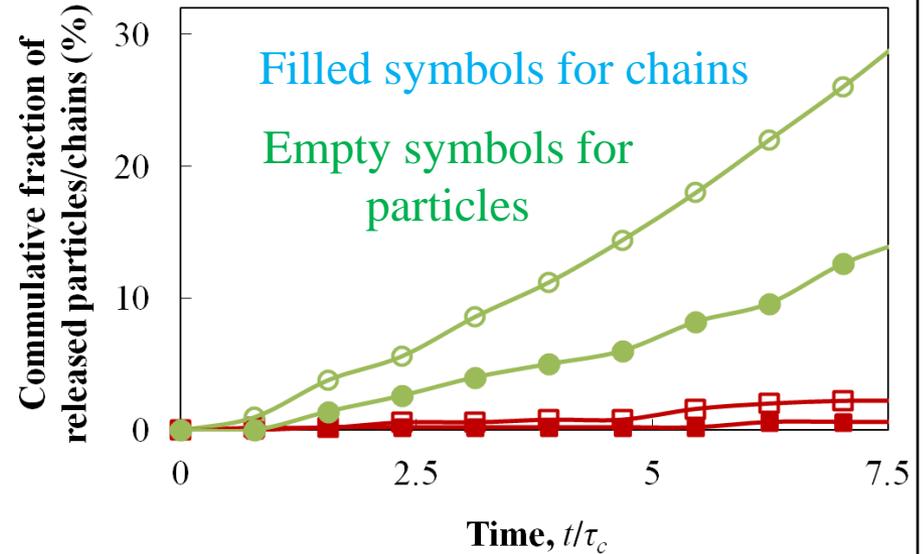
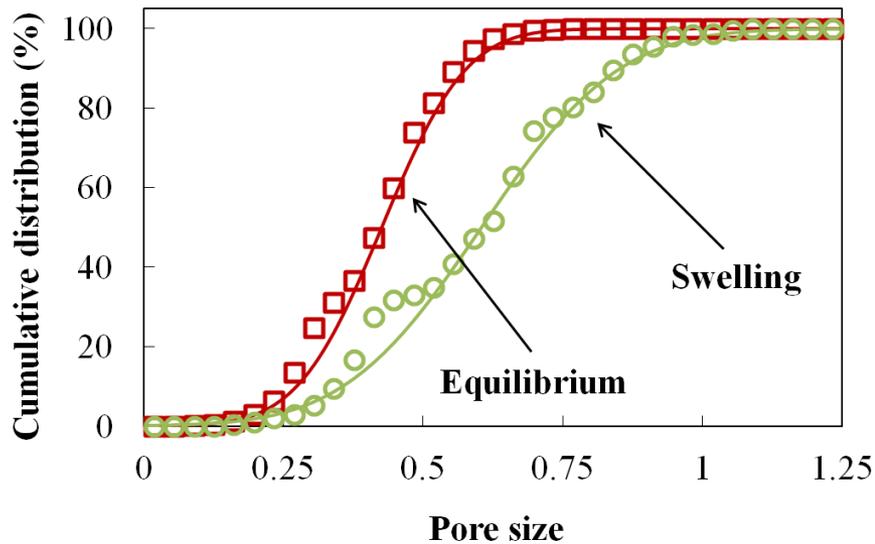
Capsule with nanoparticles



- **Capsule swelling leads to increase in network average pore size**
 - ☞ Gives rise to diffusion-controlled release

Release From Swelling Capsules

Pore size = Radius of the largest circle contained in a triangle formed by three filaments



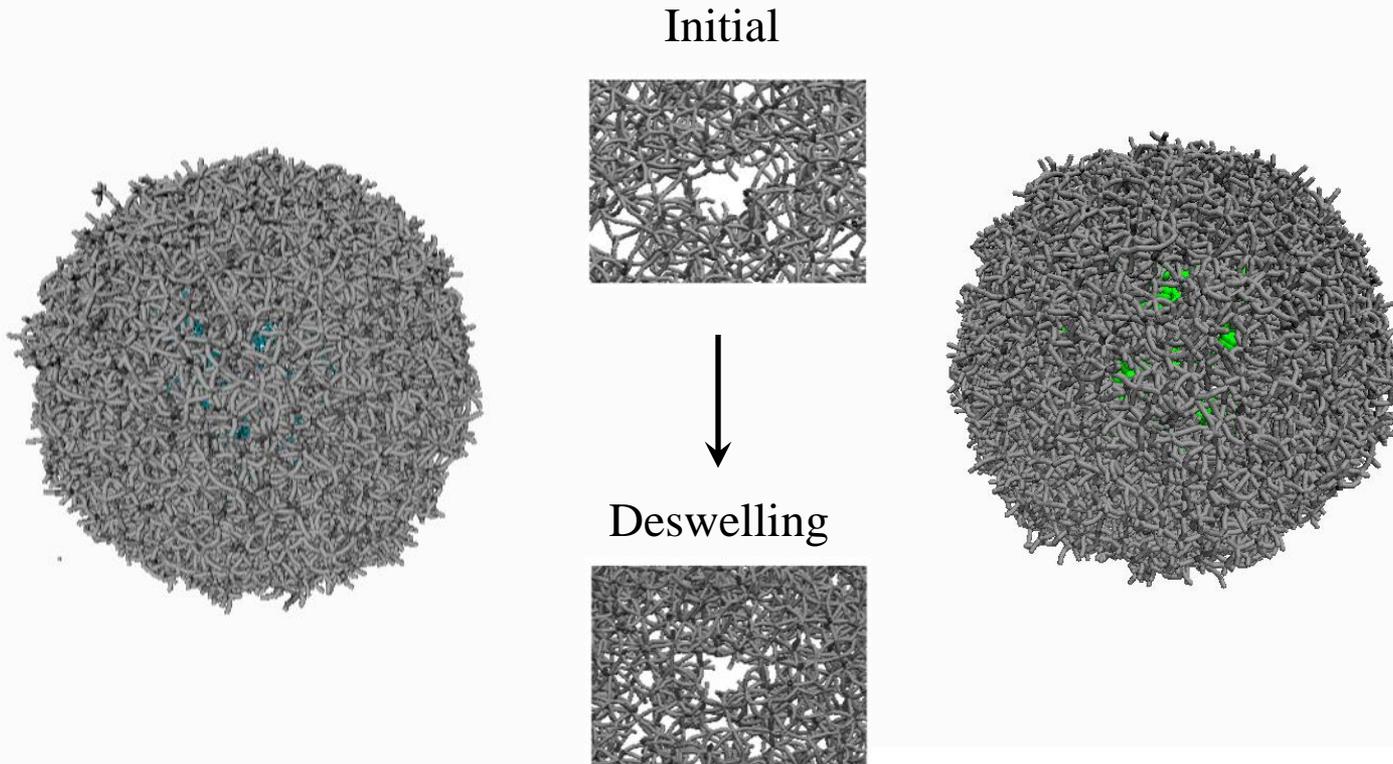
□ Steady release during swelling

- ☞ Defined by diffusion through capsule shell including entropic barrier to enter membrane
 - Polymer chains release slower than rigid nanoparticles
- ☞ Agrees well with 1D diffusion through spherical shell
 - Constant release rate

Release From Deswelling Capsules

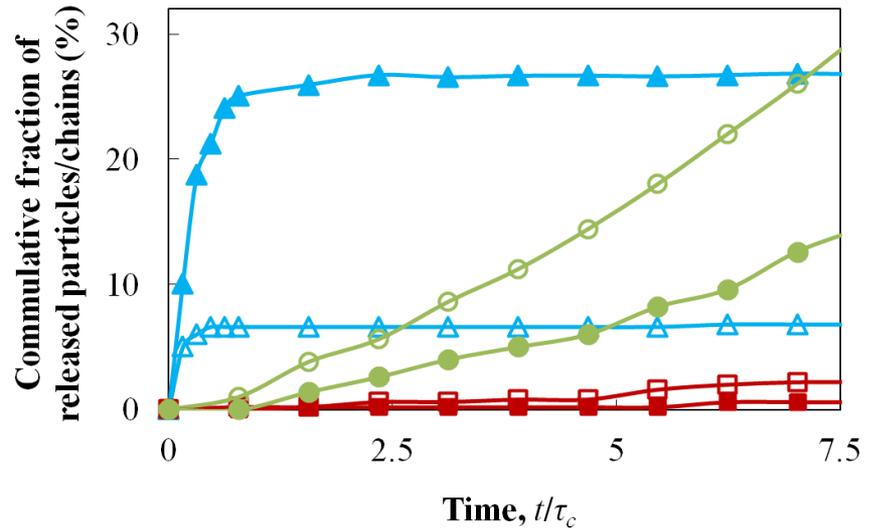
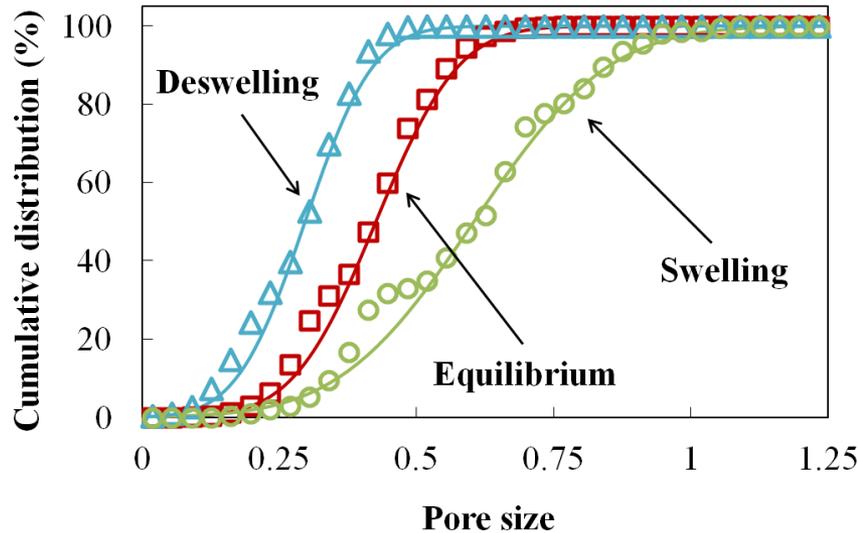
Capsule with macromolecules

Capsule with nanoparticles



- Capsule deswelling leads to squeezing flow from capsule interior
 - ☞ Results in rapid hydrodynamic release

Release From Deswelling Capsules



❑ Rapid discharge of solutes followed by no release

☞ Driven by fluid flow from inside capsule through membrane

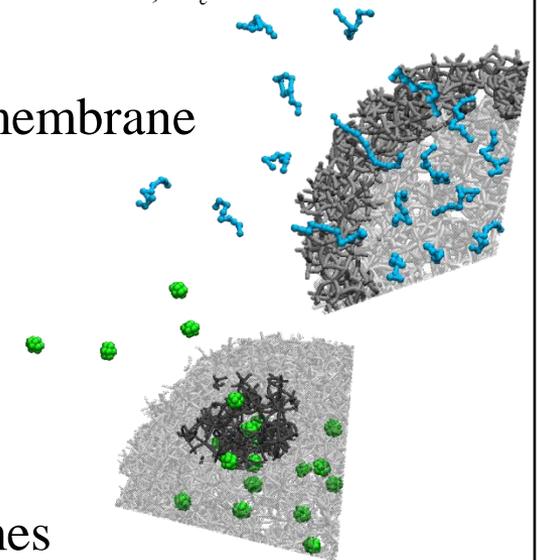
❑ Polymer chains release faster than nanoparticles

☞ Polymers unfold and reptate through membrane

❑ Precise control of release amount

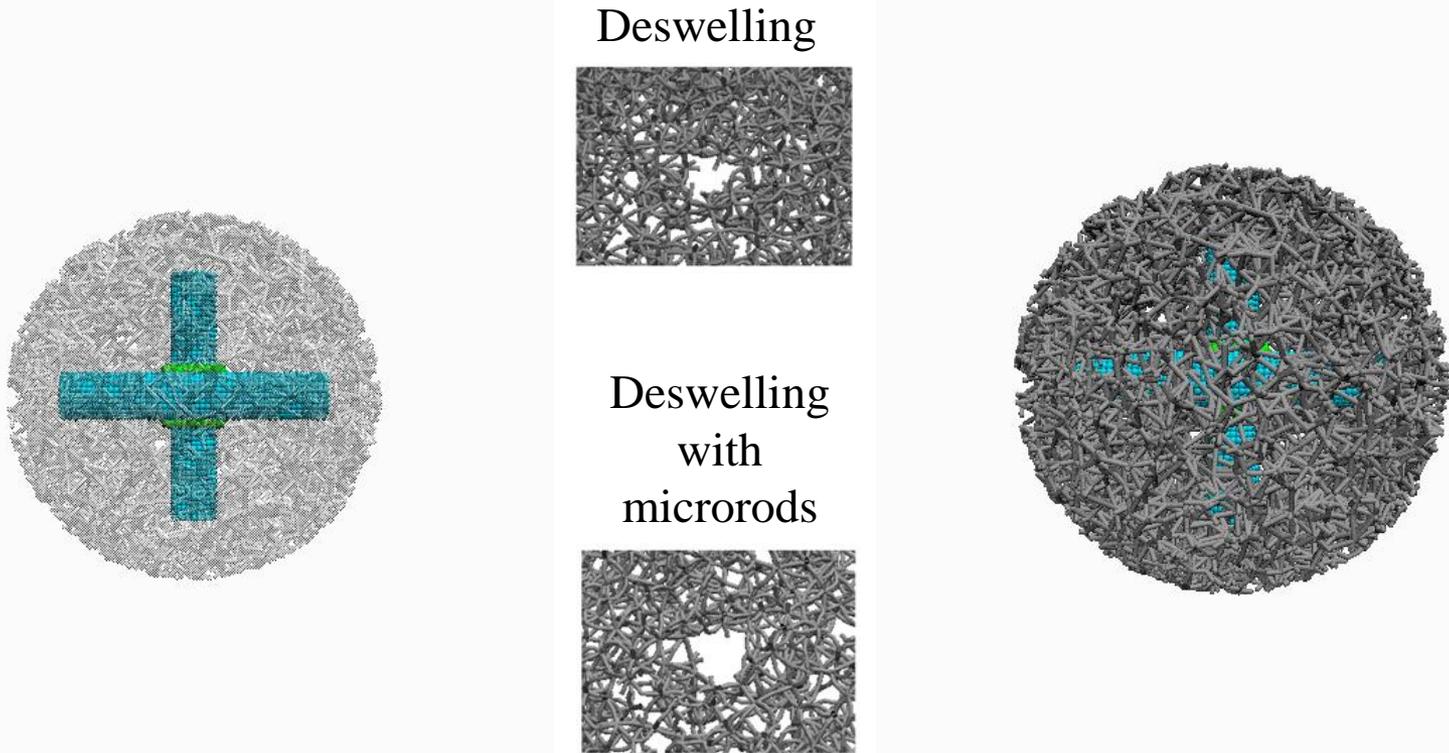
☞ Defined by volume of released fluid

☞ Release is limited by network mesh size at long times



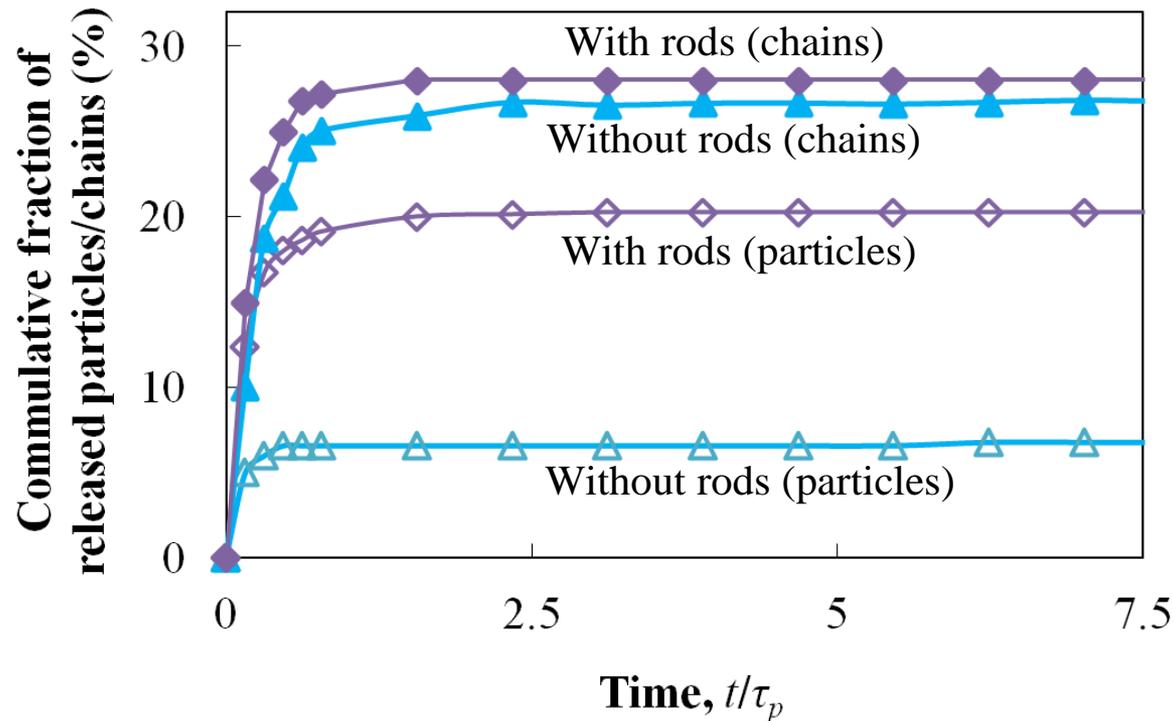
Prevent Membrane Sealing

- ❑ **Introduce long rigid microrods in capsule interior**



- ❑ **Rods stretch membrane of deswelling capsule and prevent pore closing**
 - ☞ Enhances flow driven release

Release From Deswelling Capsules With Rods



□ Inclusion of rigid microrods enhances nanoparticle release

- ☞ Membrane stretches due to interaction with rods
 - Mitigates rapid closure of membrane local “defects”
- ☞ Chain release remains unaffected
 - Release of linear macromolecules is not limited by pore sealing

Scaling Analysis

❑ Estimate rate of release during deswelling

☞ Compare relative strength of advective and diffusive transport

$$\text{Pe} = \frac{\text{Rate of advective transport}}{\text{Rate of diffusive transport}} = \frac{ub}{D_{pn}}$$

b = Capsule wall thickness

$$u = \text{Average discharge velocity} = \frac{\text{Rate of change of volume}}{\text{Surface area}} = \frac{\Delta V / \tau_c}{S}$$

D_{pn} = Effective diffusion coefficient through capsule membrane

$$\tau_c = \frac{R^2}{D_0} = \text{Time scale of capsule deswelling} \quad R = \text{Capsule outer radius at equilibrium}$$

❑ Good agreement with simulations of polymer chain release

☞ Model $\text{Pe} = 450$, simulations $\text{Pe} = 420$,

❑ Qualitative agreement for nanoparticles

☞ Particle filtering due to decrease in membrane pore size

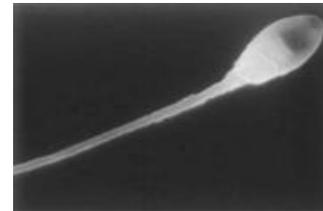
☞ Good agreement with rods: model $\text{Pe} = 120$, simulations $\text{Pe} = 90$

☞ Rods prevent particle filtering

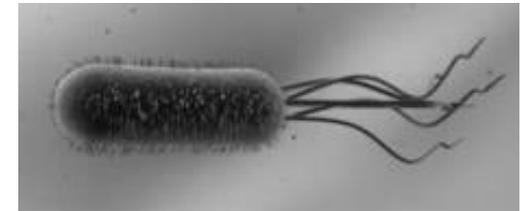
● Microswimmers

- ❑ **Microswimmers – autonomous micro-scale vehicles that can self-propel in a fluid**
- ❑ **Numerous applications in nanotechnology, MEMS research, and medicine**
 - ☞ Drug delivery
 - ☞ Lab-on-a-chip applications
 - ☞ Micro-fabrication
- ❑ **Complex design of microswimmers limits their utilization**
- ❑ **Goal: design a simple and controllable microswimmer**
 - ☞ Utilize responsive hydrogel to drive microswimmer

Natural microswimmers

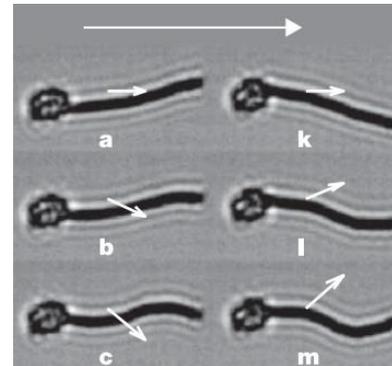


Spermatozoid

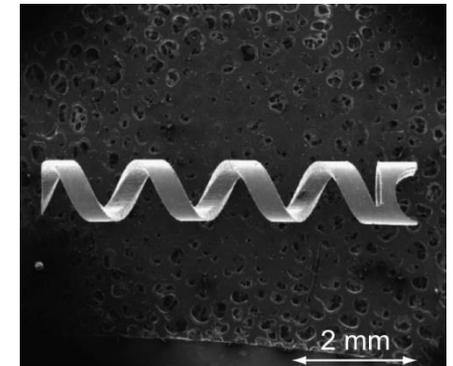


E.Coli

Artificial microswimmers



Dreyfus, *Nature*, 2005



Tiantian, *HAL*, 2014

● Bi-layered Gel

□ Gel modeled as random network of interconnected elastic filaments (springs)

- ☞ Filaments composed of stretching and bending springs connecting DPD beads
- ☞ Networks is immersed in DPD solvent

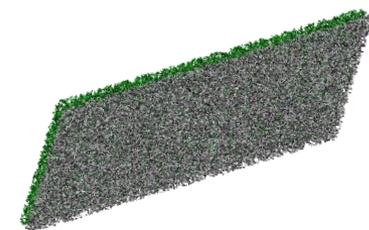
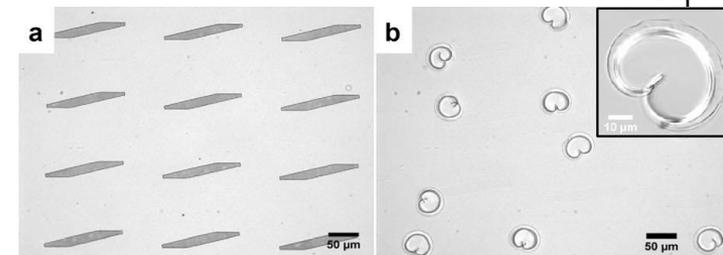
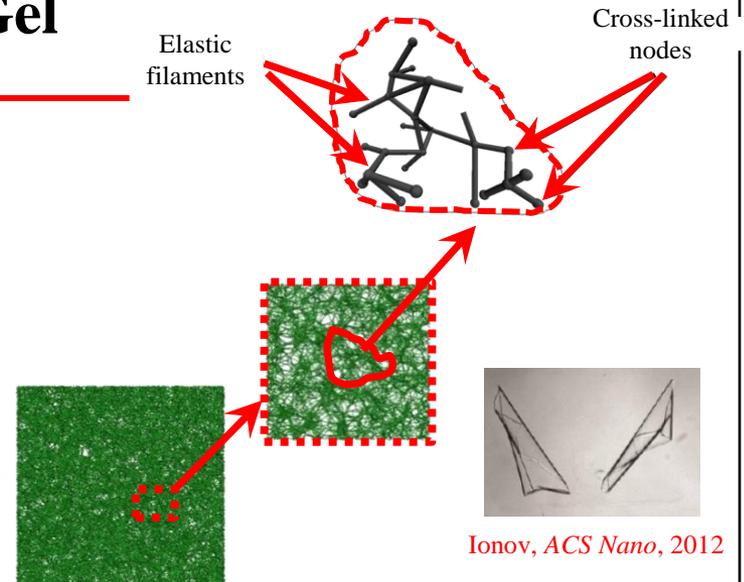
□ Gel volume transition is modeled by varying equilibrium length of network filaments

- ☞ Accounts for internal stresses that cause network to shrink or expand

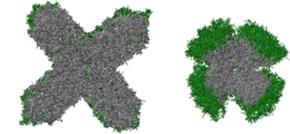


□ Swelling bi-layered sheets undergo bending

- ☞ Sheets have identical material properties
- ☞ One layer expands in response to stimulus
- ☞ Second layer is passive



● Gel Microswimmer



□ Bi-layered gel deforms upon periodic stimulus application

- ☞ Mismatch between stresses at gel interface
- ☞ Internal bending moment in network develops

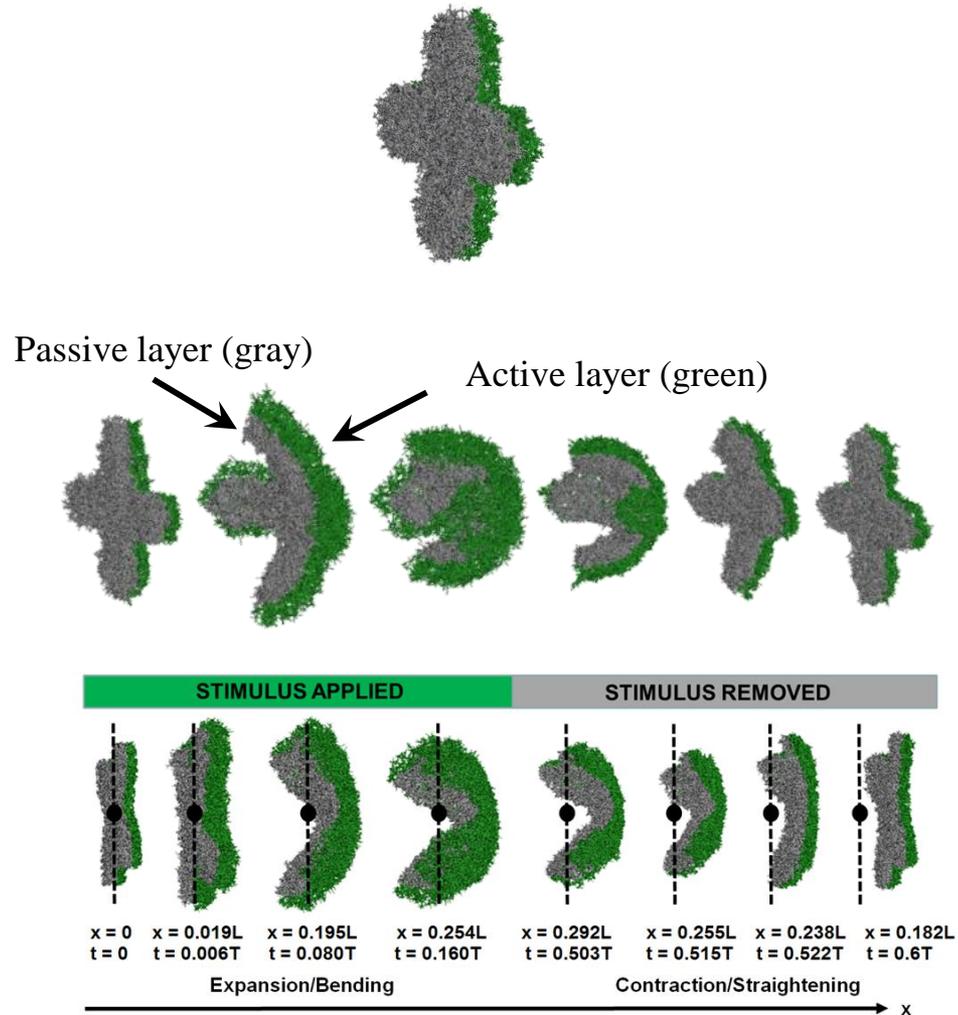
□ X-shaped geometry

- ☞ Large arms maximize bending

□ Swimming process 4 steps:

- ☞ Expansion
- ☞ Bending
- ☞ Contraction
- ☞ Straightening

□ Each periodic application of stimulus produces net forward displacement



Swimming Cycle

Swimmer trajectory

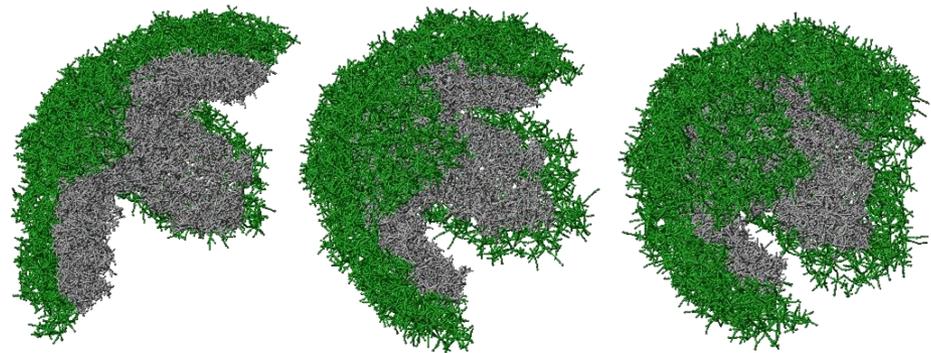
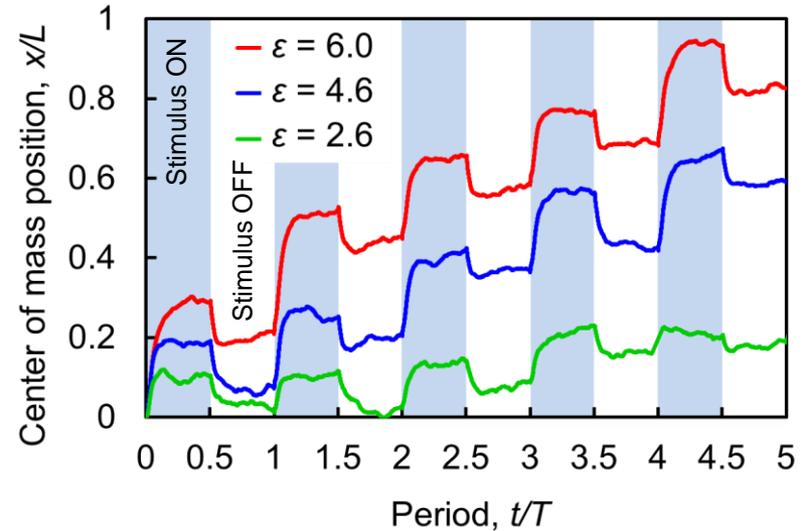
- Large forward displacement during expansion and bending phases
- Small backward displacement during contraction and straightening phases

Swimmer speed similar to that of *E.coli*

- ~ 0.2 body-lengths/period

Larger ε leads to better swimming performance

$$\varepsilon = \frac{\text{expanded network volume}}{\text{contracted network volume}}$$



$\varepsilon = 2.6$

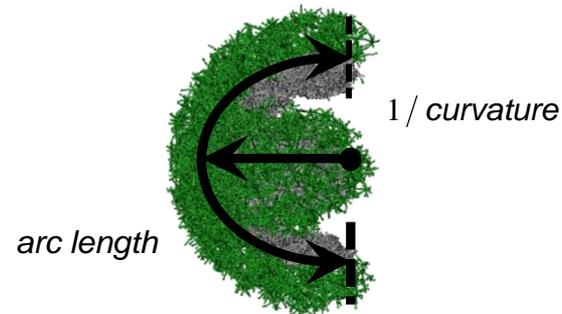
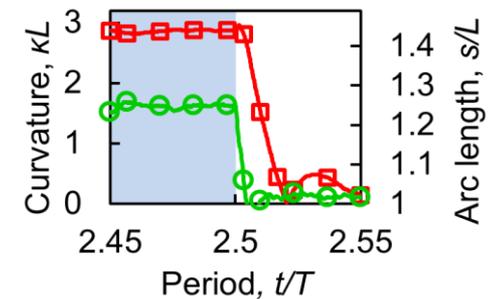
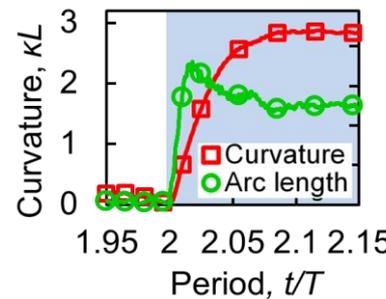
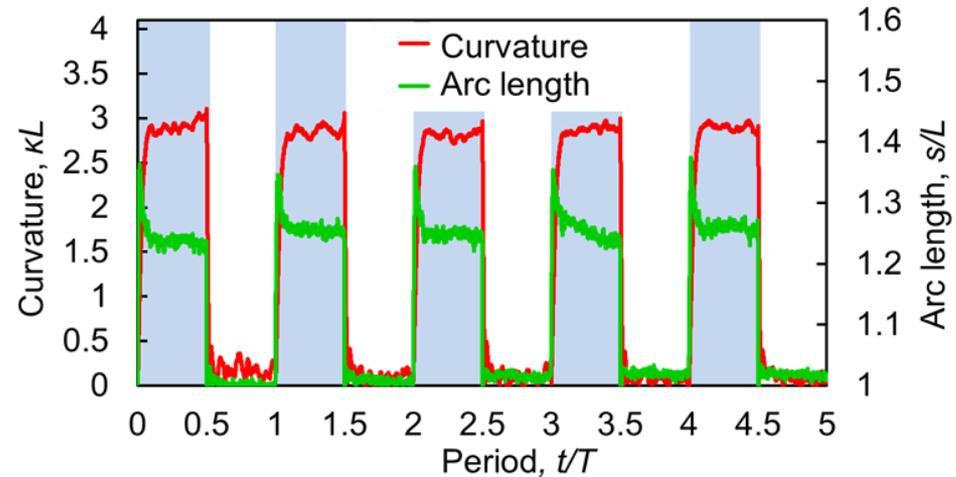
$\varepsilon = 4.6$

$\varepsilon = 6.0$

Geometric configurations at $t/T = 0.5$

Why swimmer swims at low Reynolds number

- ❑ Need to create time irreversible motion
- ❑ Characterize swimmer kinematics
 - ☞ Arc length (extension)
 - ☞ Curvature (bending)
- ❑ Hydrodynamic drag for bending is larger than drag for extension
 - ☞ Leads to time delay between changes in arc length and curvature
- ❑ Swimmer propels due to timescale difference between bending/expansion



● Optimizing Swimming Performance

□ Swimming performance defined by

- ☞ Swelling ratio (ϵ)
- ☞ Relative elasticity (thickness) of sheets (R)

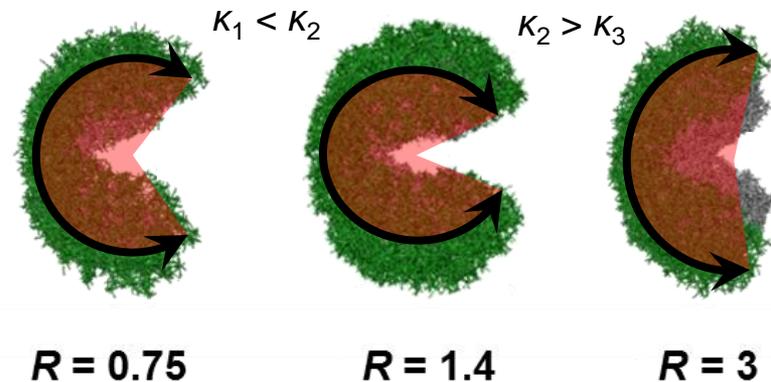
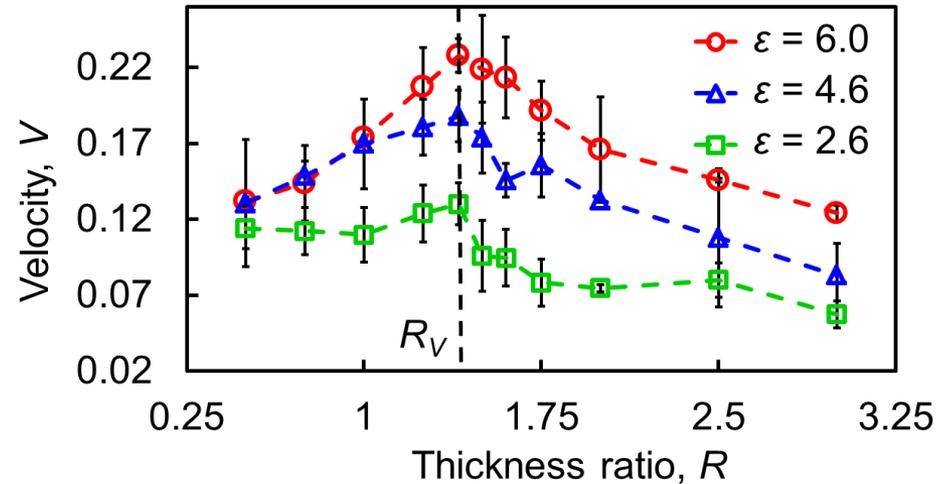
□ Larger swelling ratios lead to faster propulsion

□ Gel thickness ratio defines swimmer bending

- ☞ Passive layers that are either too thick or too thin hinder bending

□ Best swimming performance occurs for a thickness ratio of 1.4

$$R = \frac{\text{passive layer thickness}}{\text{active layer thickness}}$$



Scaling Analysis

- Increasing R leads to a monotonic decrease in swimmer's extension

$L_s/L \sim (\varepsilon^{1/3} + R)/(1 + R)$, (L_s is swollen length)

- For low/high R values curvature decreases

Leads to poor swimming performance

$$\kappa L \sim \frac{6R(1 - \varepsilon^{-1/3})}{\varepsilon^{1/3}(1 + R\varepsilon^{-1/3})^3} \left(\frac{L}{d_R} \right)$$

where κ is the curvature

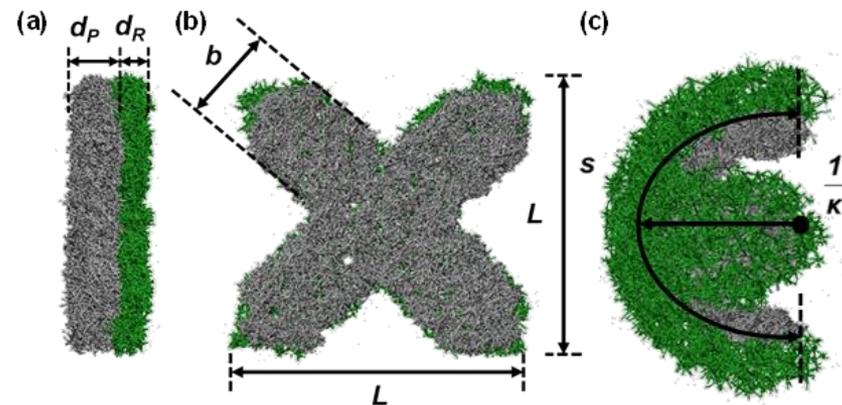
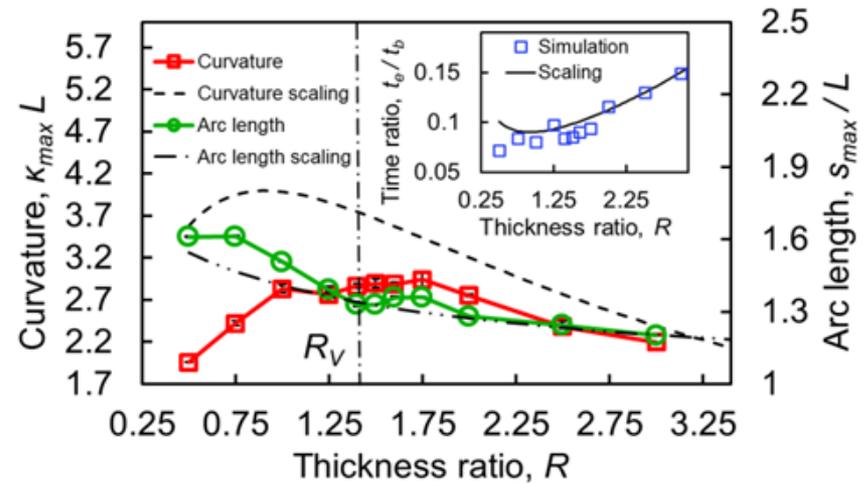
- The ratio of extensional and bending time scales

$$\frac{t_e}{t_b} \sim \frac{C_e}{C_b} \frac{8(1 + R\varepsilon^{-1/3})^3}{3R\varepsilon^{-1/3}} \left(\frac{d_R}{L} \right)^2$$

$C_e/C_b \sim 0.5$ is ratio of drag coefficients for extension and bending

Theory predicts t_e/t_b minimum at $R \sim 1$

- Theory supports computational models



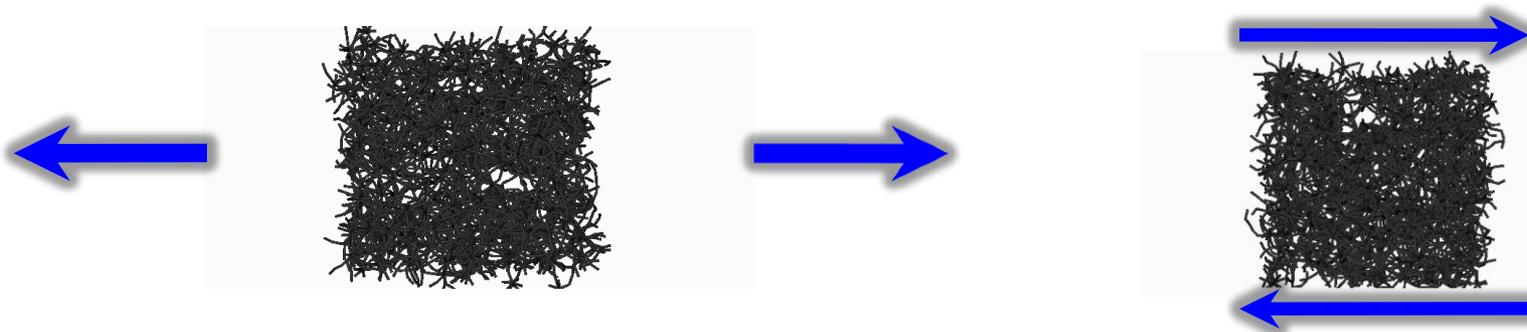
Summary: Polymer Network

❑ Developed mesoscale model for random polymer networks

- ☞ Validated by simulating transport properties and swelling kinetics
 - ☞ Good agreement with theory and experiment

❑ Examined transport in mechanically deformed (anisotropic) networks

- ☞ Stretching enhances permeation and diffusion in direction of deformation
- ☞ Total diffusivity remains unchanged under deformation
- ☞ Permeability defined by internal network orientation



Masoud and Alexeev, Permeability and diffusion through mechanically deformed random polymer networks *Macromolecules* 43, 10117 (2010)

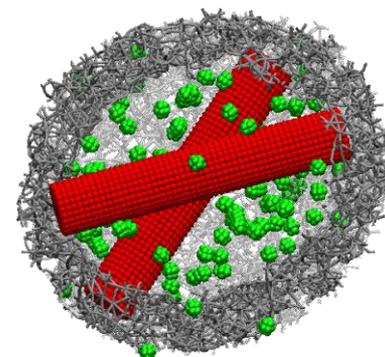
Summary: Microgel Capsules

□ Responsive capsules enable control over release dynamics

- ☞ Swollen microgel capsules: slow diffusive release
- ☞ Deswelling capsules: fast hydrodynamic release
 - ☞ Precise control over released amount
 - ☞ Pulsatile release with repeating stimuli

□ Applications

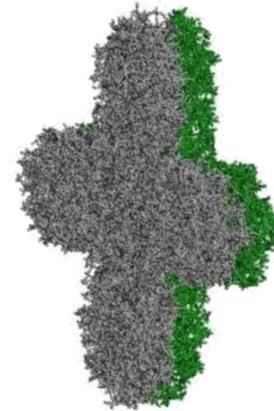
- ☞ Targeted drug delivery, in-vivo sampling, viruses removal, detoxification



Masoud and Alexeev, Controlled release of nanoparticles and macromolecules from responsive microgel capsules **ACS Nano** 6, 212, 2012

● Summary: Gel Swimmer

- ❑ **Designed a simple self propelling hydrogel microswimmer**
- ❑ **Examined effects of swelling ratio**
 - ☞ Larger ratios lead to faster swimming
- ❑ **Developed an approach for optimizing swimming performance**
 - ☞ $R \sim 1.4$ improves swimming speeds
- ❑ **Explained fundamentals behind time irreversible motion**
 - ☞ Time delay between expansion and bending



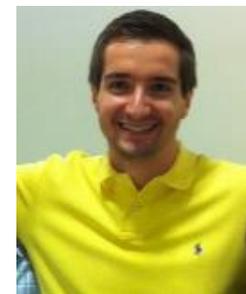
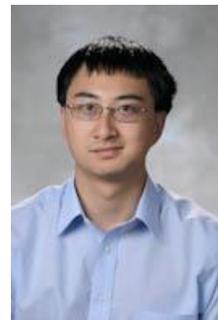
Nikolov, Yeh, Alexeev, Self-propelled microswimmer actuated by stimuli-sensitive bilayered hydrogel. **ACS Macro Letters** 4, 84, 2015.

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