



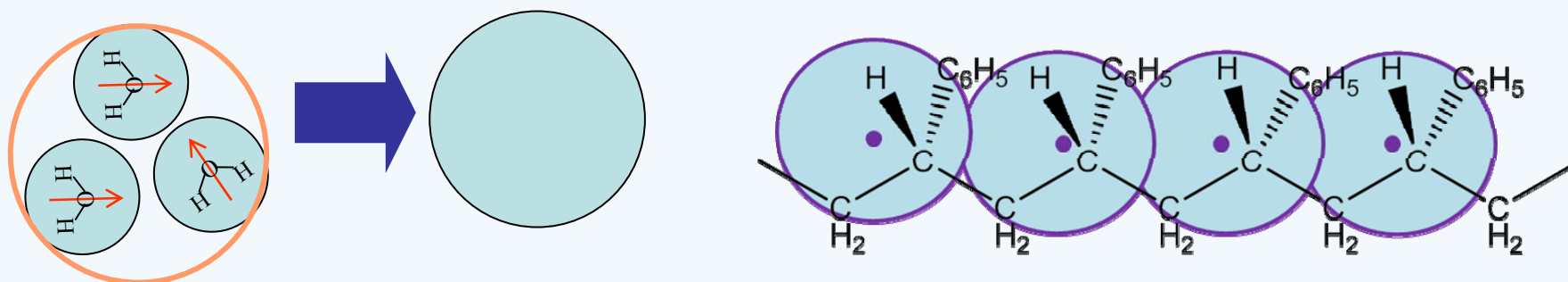
The Application of Dissipative Particle Dynamics Simulation Technique to Complex Fluids

Hongxia Guo

**State-Key Lab. of Polymer Physics and Chemistry
Institute of Chemistry, Chinese Academy of Sciences**

I. Introduction

➤ **DPD** : a particle-based mesoscopic simulation technique for complex fluids



- individual particle represents a cluster of atoms or molecules
- interacts via soft potentials, and is subjected to dissipative and random forces (DPD thermostat)
- ∴ **DPD simulation a valuable approach over MD and MC**
- DPD thermostat conserves both global and local momentums, is Galilean invariance, avoids profile biasing of NEMD simulation
- ∴ **DPD thermostat an ideal thermostat in (NE)MD**



- ## II. Application of DPD simulations—1.ternary polymer blends
- 2.nanoparticles filled polymer blend
 - 3. LCs or its related complex systems
 - 4. lipid

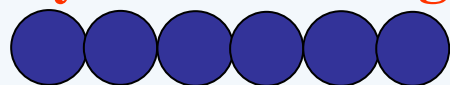


II. Application—1.ternary polymer blends

Interfacial properties and phase transitions in ternary symmetric homopolymerecopolymer blends: DPD

II. Application—1.ternary polymer blends

Polymer blending: an assembly of the interfaces



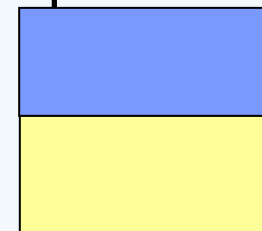
homopolymer A



homopolymer B

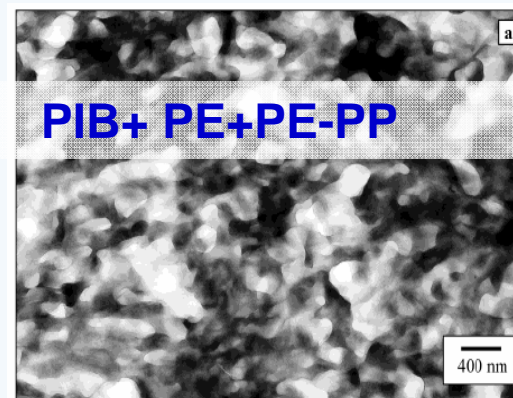
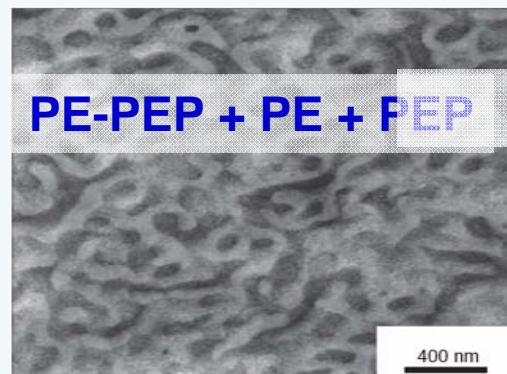
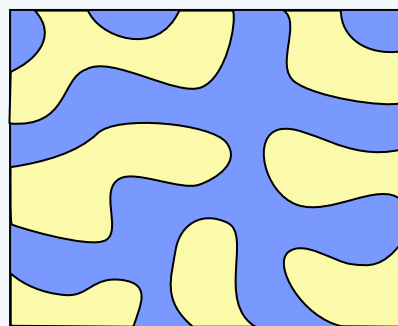
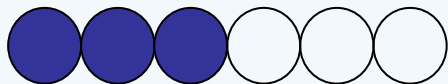


macrophase separation



Poor mechanical performance

∴ **compatibilizer-copolymer**

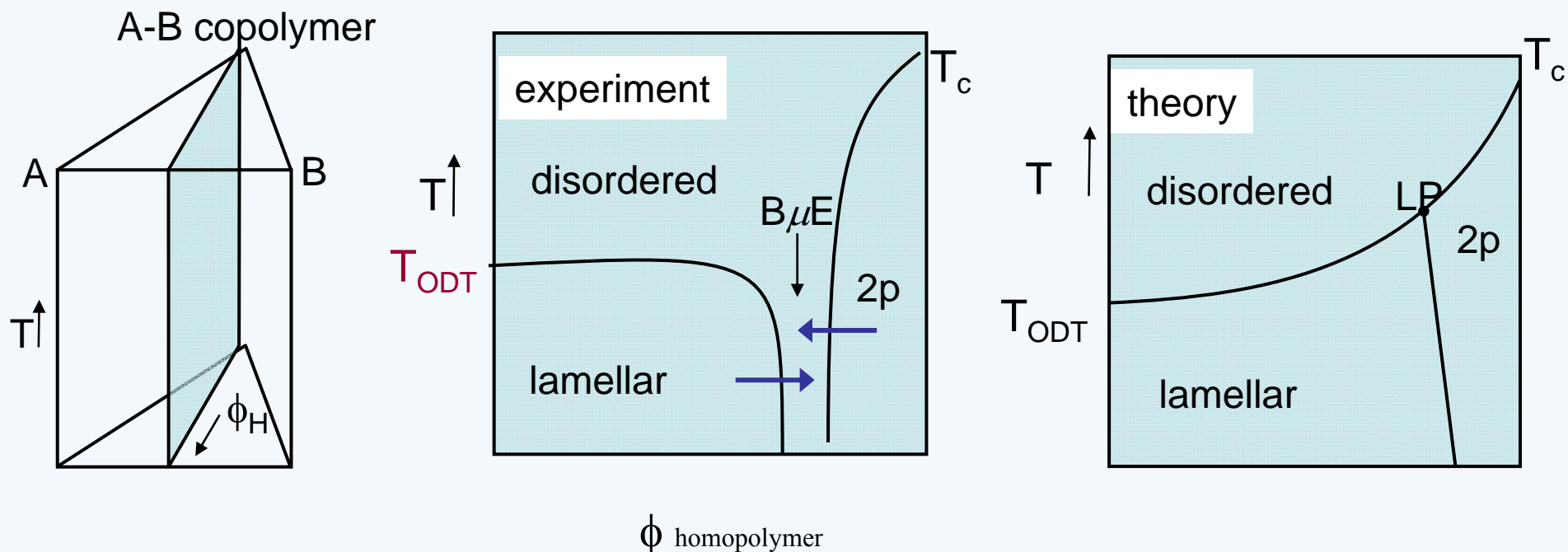


Polymeric microemulsions

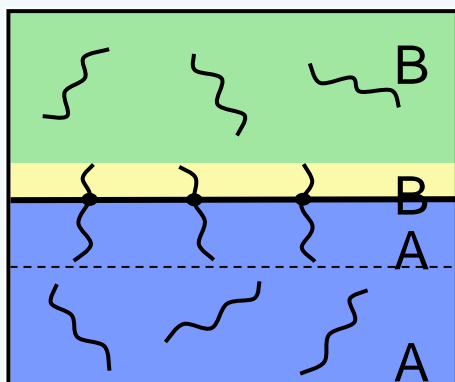
Improved mechanical/transport property

Many Applications: elastomer, porous membranes

II. Application—1.ternary polymer blends



symmetric ternary blend



our interest:

- interfacial properties of such blends (especially lamellar phase)
- their relevance to the phase transitions



II. Application—1.ternary polymer blends

Why DPD not MD:

- long-time and metastability in ternary polymer blend
- their phase behavior and interfacial properties within the mesoscopic spatio-temporal scale
- DPD: soft repulsive potential and a momentum conserving thermostat

Interfacial properties:

Interfacial tension γ

bending modulus k

interlayer compressibility modulus B

Measurement : the undulation spectrum requires simulations on a relatively large system, as this method is only applicable in the “long-wavelength” limit!!!.

II. Application—1.ternary polymer blends

- Elasticity of an amphiphilic monolayer between coexisting phases
the undulation spectrum: these interfaces display **long-wavelength fluctuations** = single smoothly undulating surfaces

Helfrich's curvature model:

$$f = \int dA \left[\gamma + \frac{k}{2} (c_1 + c_2 - c_0)^2 + k' c_1 c_2 \right] \longrightarrow f = \int_A dx dy \left[\frac{\gamma}{2} \left(\frac{\partial h}{\partial x} + \frac{\partial h}{\partial y} \right)^2 + \frac{k}{2} \left(\frac{\partial^2 h}{\partial x^2} + \frac{\partial^2 h}{\partial y^2} \right)^2 \right]$$

$h(x,y)$: a height fluctuation function, the displacement of the interface from its mean position

a Fourier transform $f(\hat{h}_q^0) = \frac{\gamma}{2} q^2 \hat{h}_q^0 + \frac{k}{2} q^4 \hat{h}_q^0$ } **spectral intensities of each undulation mode**

equipartition theorem $\langle f(\hat{h}_q^0) \rangle = \frac{1}{A} \frac{K_B T}{2}$ } $\langle |\hat{h}_q^0|^2 \rangle = \frac{1}{A} \frac{K_B T}{(\gamma q^2 + k q^4)}$

fitting undulation spectrum **at small q**
interfacial tension γ
bending modulus k

$$\frac{1}{q^2 S(q)} \sim \gamma + k q^2$$

interfacial structure factor

$$S(q) = \langle |\hat{h}_q^0|^2 \rangle A$$

II. Application—1.ternary polymer blends

➤ Elasticity of the lamellar phase

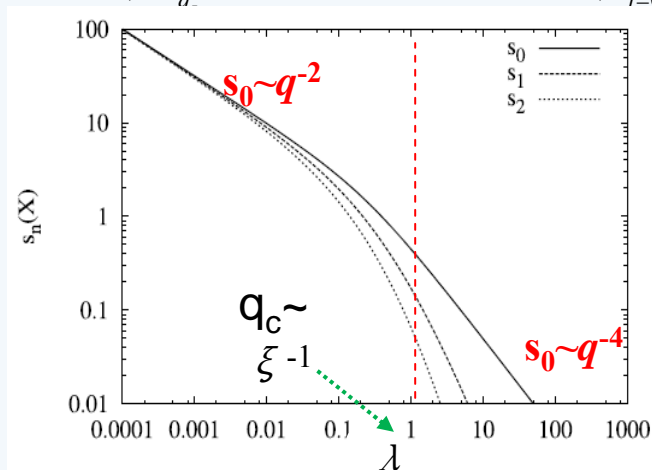
Discrete Harmonic : lamellae as a discrete set of two-dimensional fluctuating layers, stacked in the z direction, the energy cost due to local layer deformations and local deviations from the average interlayer distance

$$f = \int dA \left[\gamma + \frac{K_c}{2} (c_1 + c_2 - c_0)^2 + k' c_1 c_2 \right] \longrightarrow f = \sum_{n=0}^{N-1} \int_A dx dy \left\{ \frac{K_c}{2} \left(\frac{\partial^2 h_n}{\partial x^2} + \frac{\partial^2 h_n}{\partial y^2} \right) + \frac{B}{2} (h_n - h_{n+1})^2 \right\}$$

bending modulus K_c and compressibility modulus B

trans-bilayer structure factor:

$$s_n(q_{\perp}) = \frac{1}{N^2} \sum_a e^{iq_z n d} \langle |h(q_{\perp}, q_z)|^2 \rangle = \frac{1}{N} \sum_{i=0}^{N-1} \langle h_j(q_{\perp}) \cdot h_{n+j}(q_{\perp})^* \rangle$$



Autocorrelation fluctuation spectra

$$N \rightarrow \infty$$

$$s_0(q_{\perp}) = \frac{L_x L_y k_B T}{K_c q_{\perp}^4} \left[1 + \frac{4}{\lambda} \right]^{-1/2}$$

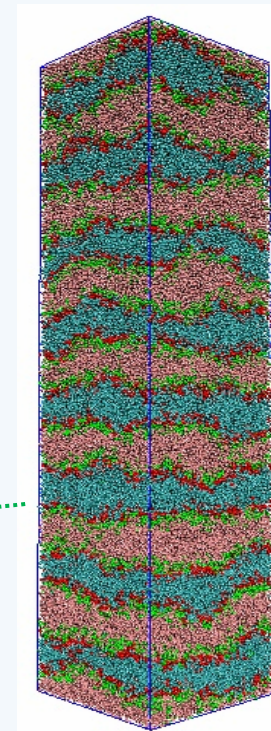
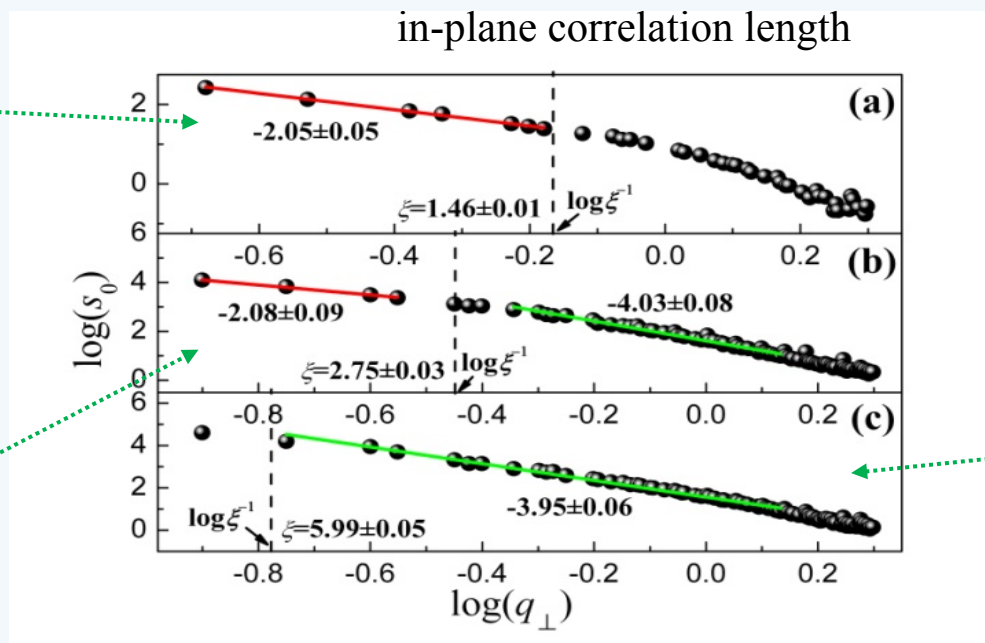
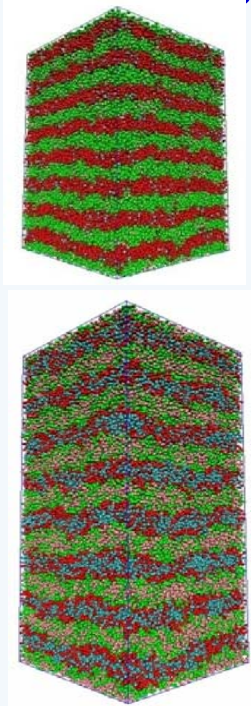
$$s_n(q_{\perp}) = s_0(q_{\perp}) \left[1 + \frac{\lambda}{2} - \frac{1}{2} \sqrt{\lambda(\lambda + 4)} \right]^n$$

$$\lambda = (\xi q_{\perp})^4 \quad \text{in-plane correlation length}$$

little work has been done about the undulations and thickness fluctuations in the lamellar phase

II. Application—1.ternary polymer blends

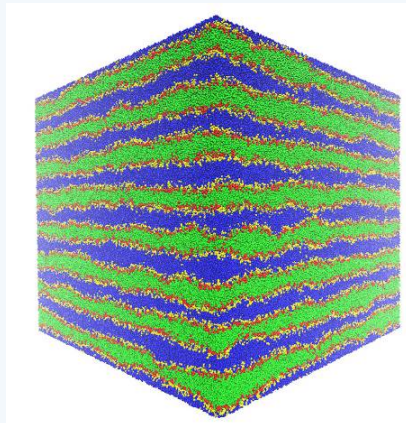
➤ Elasticity of lamellar phases



Autocorrelation fluctuation spectra s_0

(a) $\Phi_H=0\%$, $L_x=L_y=30r_c$, (b) $\Phi_H=55\%$, $L_x=L_y=50r_c$, (c) $\Phi_H=75\%$, $L_x=L_y=50r_c$.

Simulation boxes containing as many as eight lamellae the simulation results can be well compared to the above mentioned continuum theory for stacked membranes.



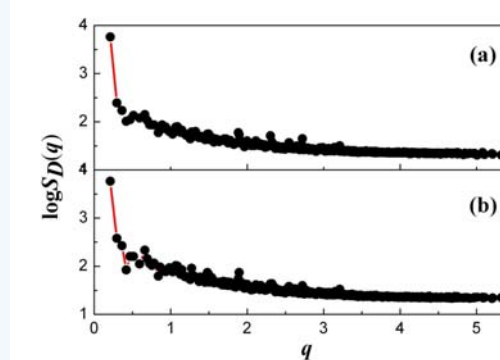
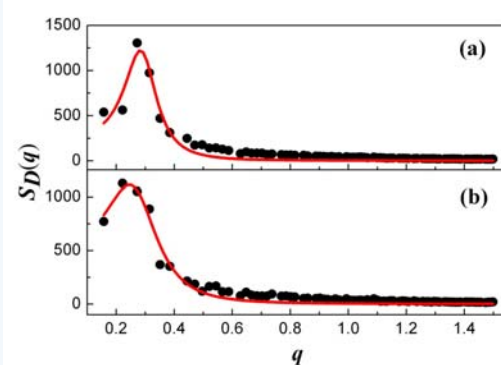
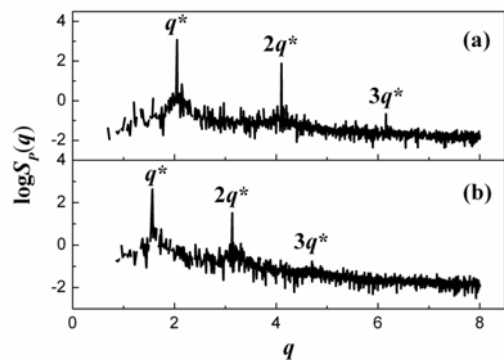
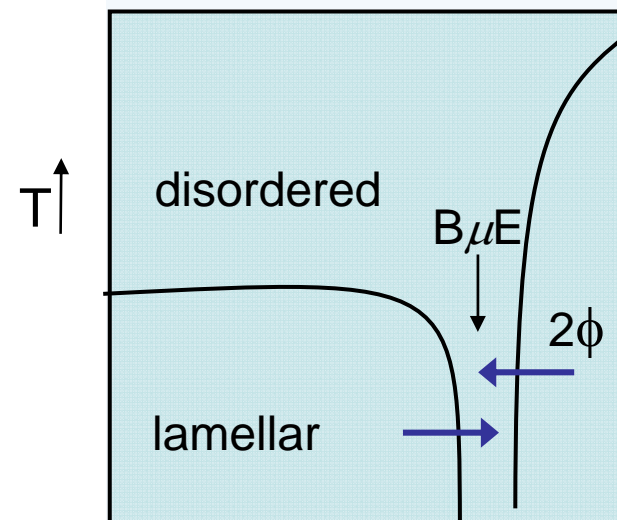
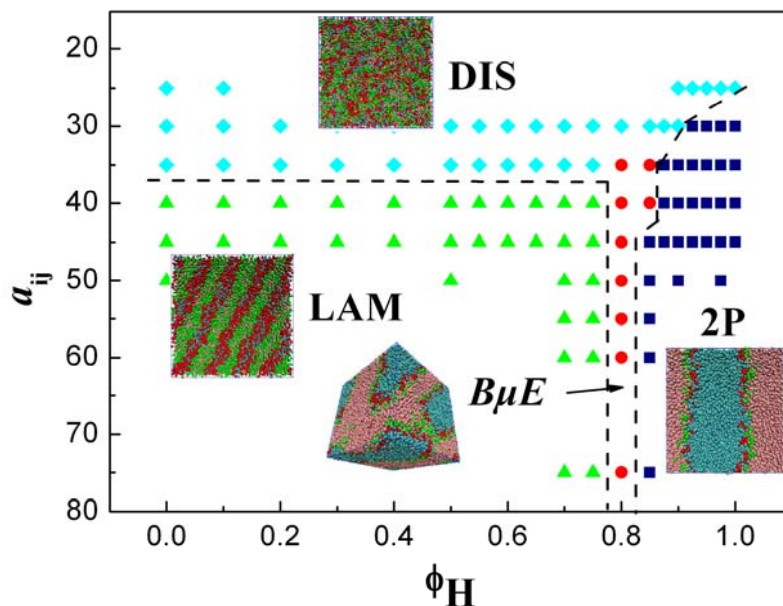
144*144*160

11,059,200 beads! !



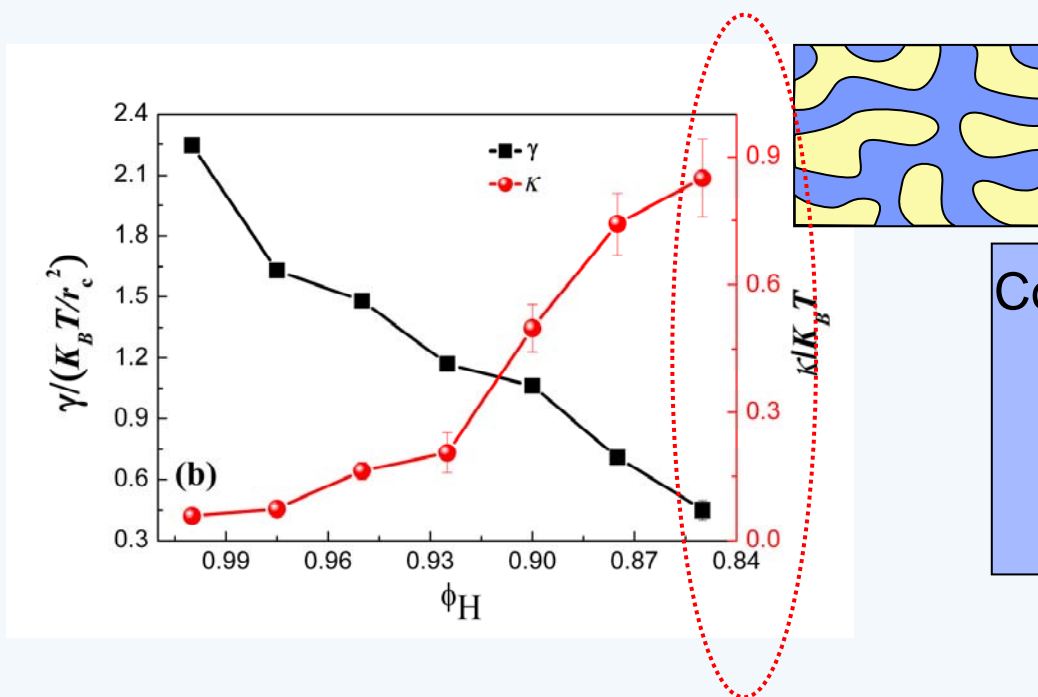
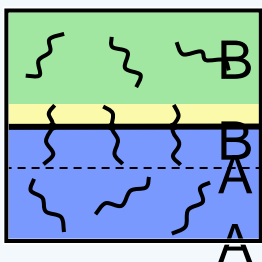
II. Application—1.ternary polymer blends

LAM: lamellar
 2P : macro-phase
 separated phase
 DIS: disorder
 BuE: bicontinuous
 microemulsion



II. Application—1.ternary polymer blends

● Interfacial properties of 2P phase



Conditions for $B_{\mu}E$

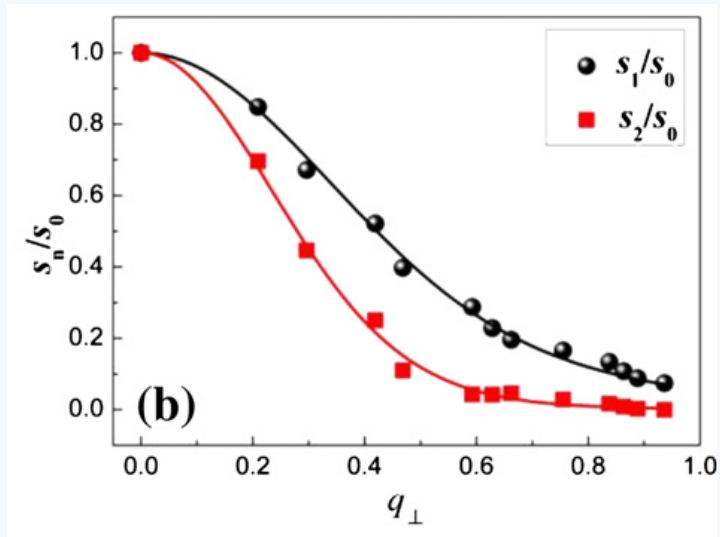
$$\gamma = 0$$

$$K = K_B T$$

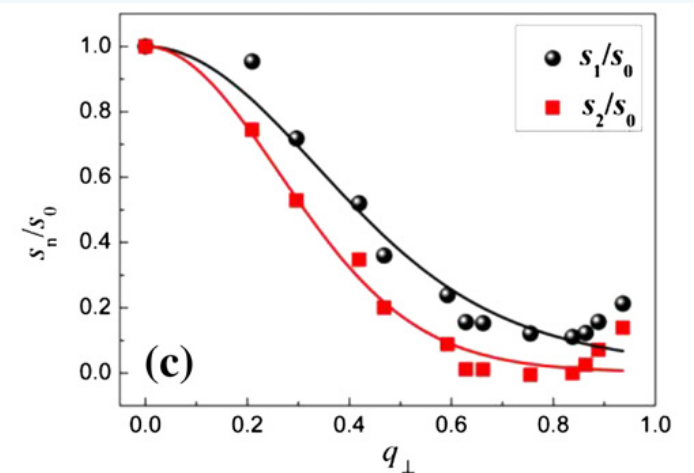
- γ : $B_{\mu}E$ structure to accommodate an extensive amount of copolymer-laden internal interfaces
- K : thermal fluctuations to stabilize $B_{\mu}E$ structure

II. Application—1.ternary polymer blends

● Interfacial properties of LAM phase



LAM stack of 8 bilayers



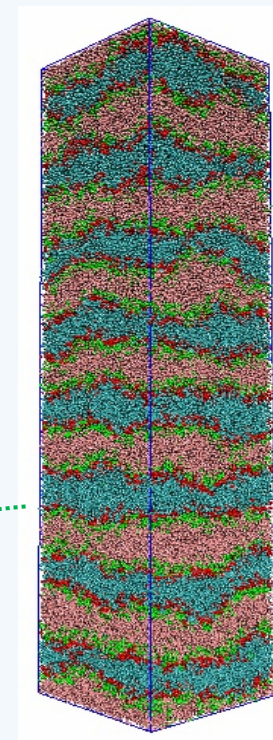
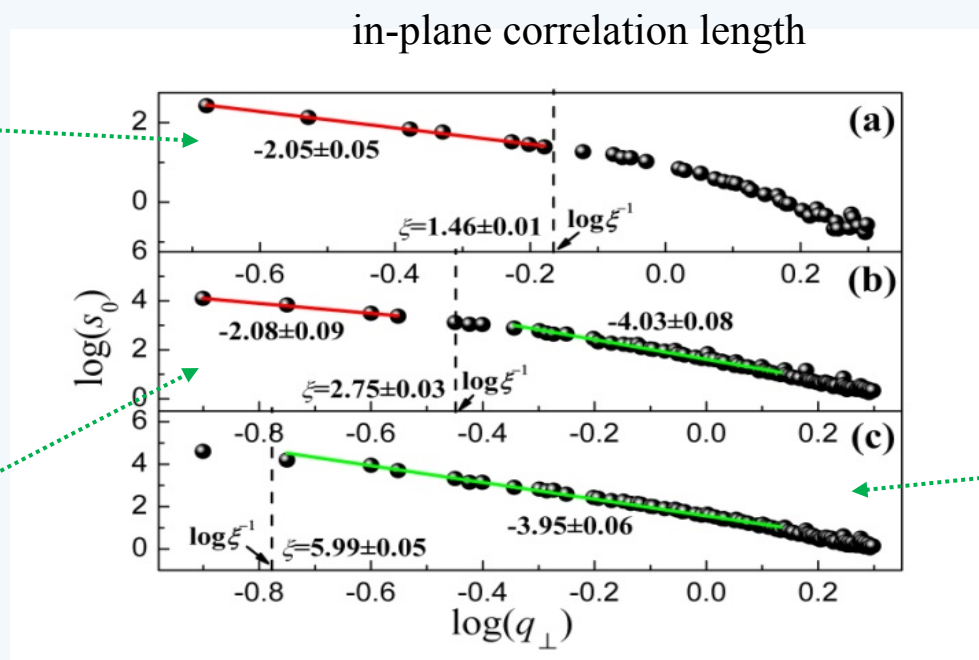
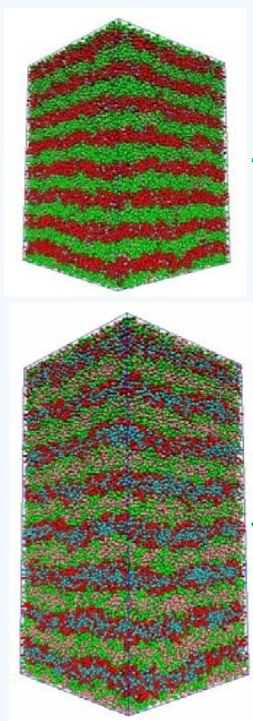
LAM stack of 4 bilayers

Note: DH theory for an infinitely thick stack of bilayer

an LAM stack of 8 bilayers is sufficiently large to reproduce the behavior described by continuum DH theory!!!

II. Application—1.ternary polymer blends

● Interfacial properties of LAM phase

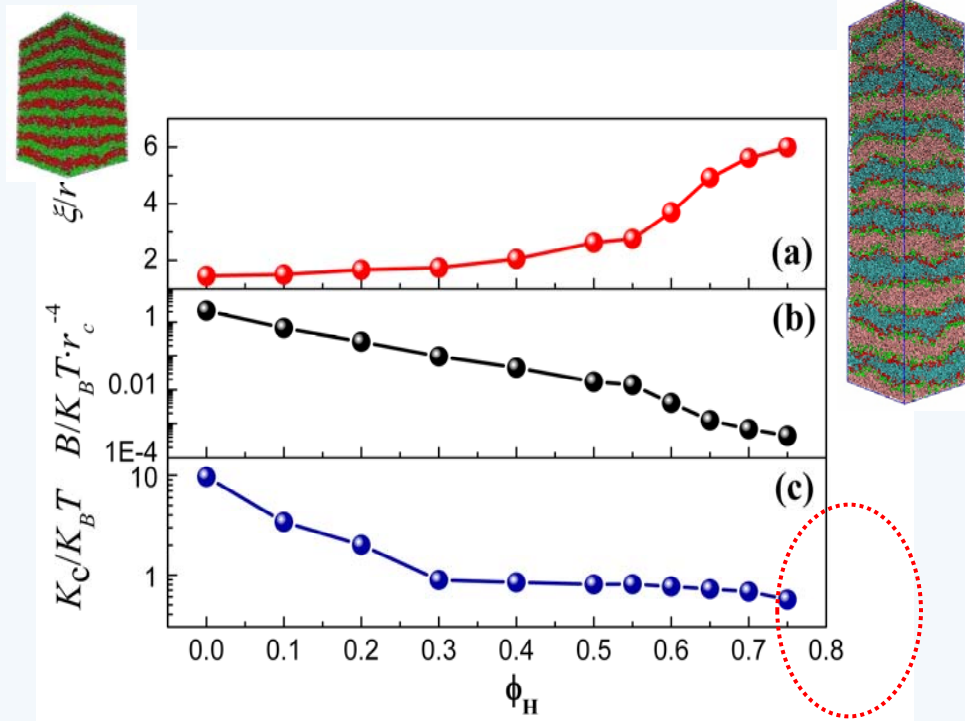


Autocorrelation fluctuation spectra s_0

(a) $\Phi_H=0\%$, $L_x=L_y=30r_c$, (b) $\Phi_H=55\%$, $L_x=L_y=50r_c$, (c) $\Phi_H=75\%$, $L_x=L_y=50r_c$.

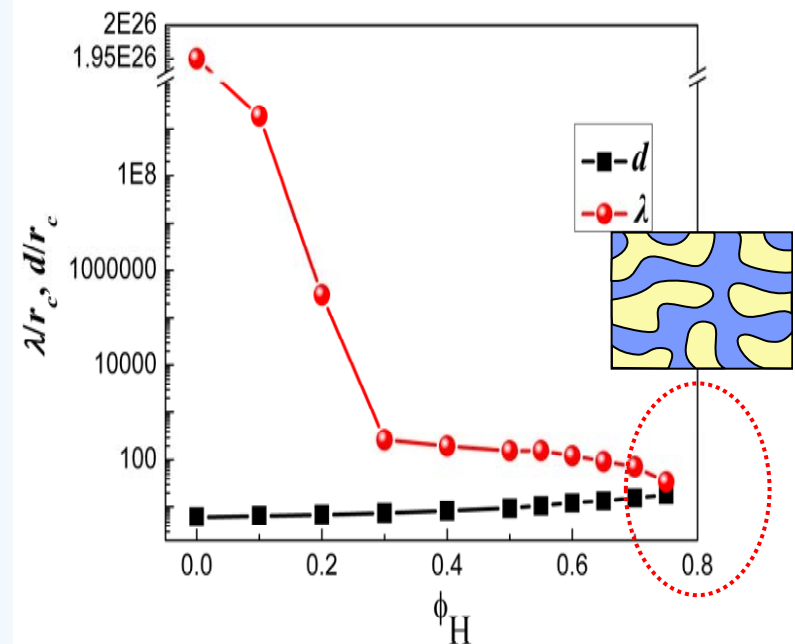
II. Application—1.ternary polymer blends

● Interfacial properties of LAM phase



in-plane correlation length ξ
compressibility modulus B
bending modulus Kc
as function of Φ_H

$$\lambda = a \exp\left(\frac{2\pi K_c}{K_B T}\right)$$



the persistence length of the
interface and the lamellar spacing

$$\text{LAM- } B_{\mu E}: Kc < K_B T, \lambda < d$$



II. Application—1.ternary polymer blends

- DPD: a simple ternary symmetric system, A₄B₄/A₂/B₂
- For the 2P systems:
addition of diblock copolymers, γ reduces but K increases;
phase transition from 2P to B_μE: $\gamma = 0$ and $K = K_B T$
- an LAM stack of 8 bilayers is sufficiently large to re-produce the behavior described by continuum DH theory==feasibility of DH theory to calculate K_c and B of the lamellar stack from the simulation data
- For the LAM systems:
with the addition of homopolymers, ξ increases but B and K_c reduce;
phase transition from LAM to B_μE : $K_c < K_B T$, $\lambda < d$

fundamental understanding of the interfacial properties and their relevance to the phase transitions in ternary symmetric blends!!



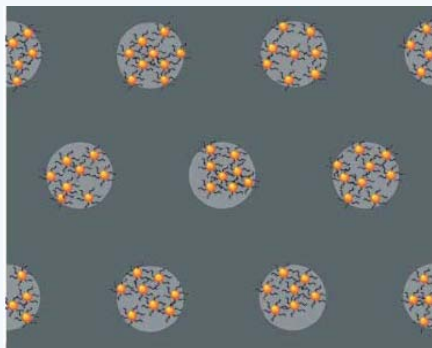
II. Application—2. nanoparticles filled polymer blends

- The effect of Janus nanospheres on the phase separation of polymer blends
- The effect of Janus nanoparticles with various shapes and different dividing surface designs on the ordering and compatibilizing performance in immiscible polymer blends

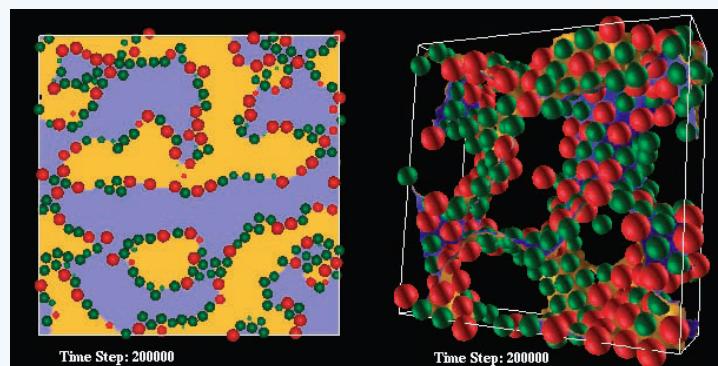
II. Application—2. nanoparticles filled polymer blends

Nano-
particle

unique structural, mechanical, optical, electronic, magnetic properties



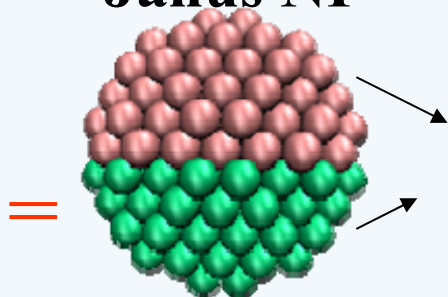
Self-directed self-assembly
Nature 2005, 434, 55



A Route to Fluid-Bicontinuous Gels
Science, 2005, 30,2198



Janus NP

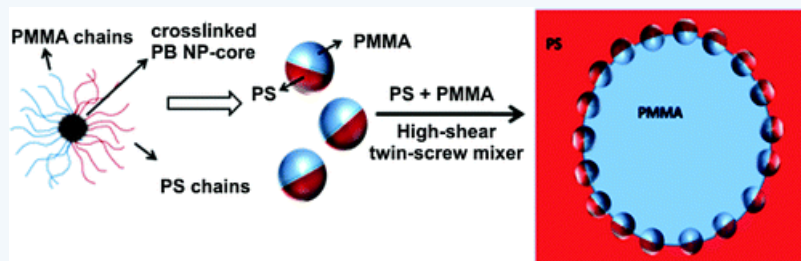


two compartments of
different chemical makeup or surface properties
i.e., hydrophilic hydrophobic

Roman God

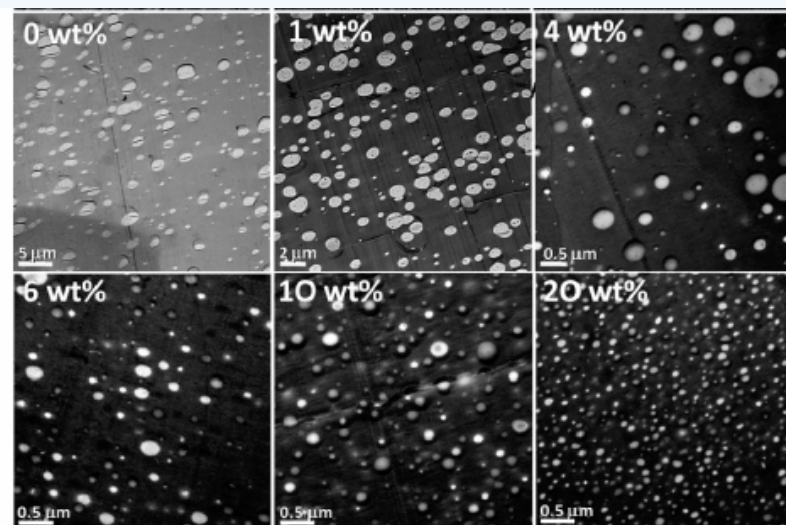
Adv. Mater., 2006, 18, 1152; 2010,22,1060;.....

a novel type of efficient stabilizers in polymer alloys



**combination of amphiphilicity
with the particle character**

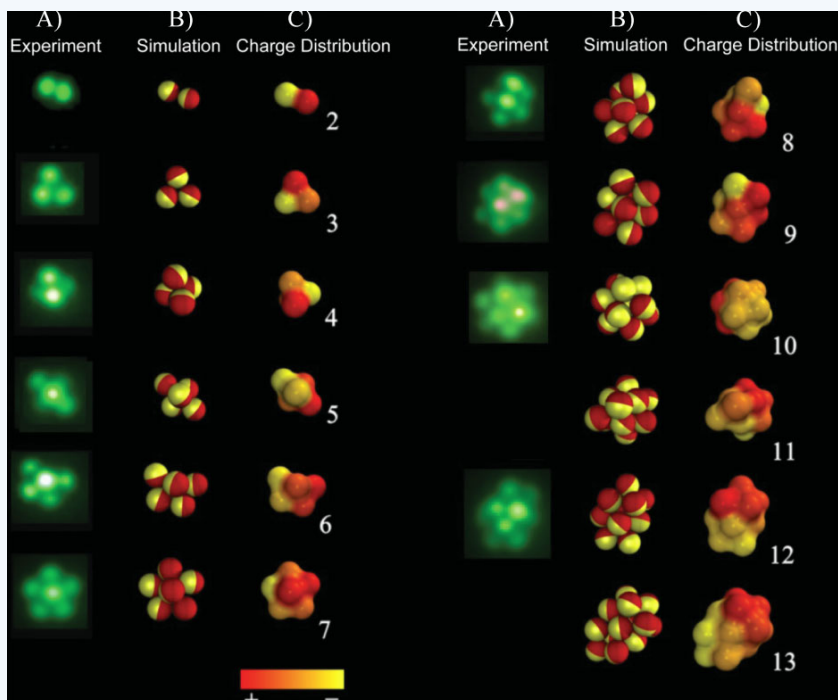
Or Janus NP as “solid surfactants” to stabilize emulsions and foams



**SEM images obtained for blends at a PS/PMMA ratio
of 6/4 with Different NP content**

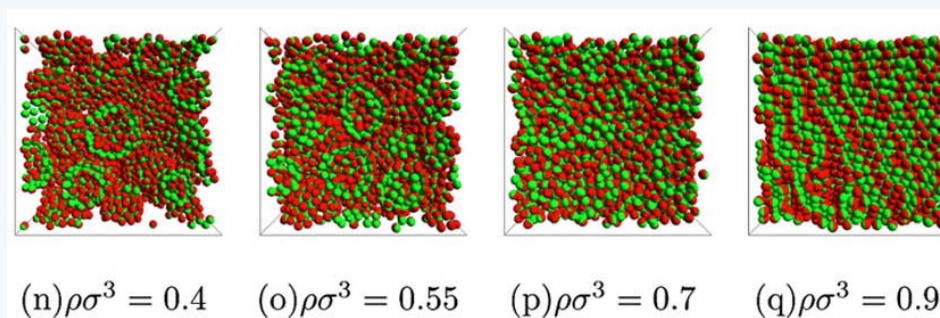
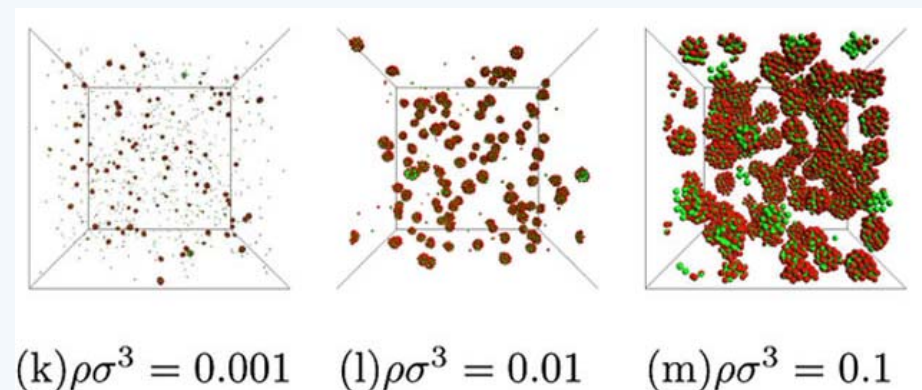
**Janus nanospheres are
superior to block
copolymers in the
emulsification of the
polymer blends**

Janus NPs : a unique type of building blocks for directional self-assembly of superstructures



Janus nanospheres in water
complex clusters with various sizes and shapes

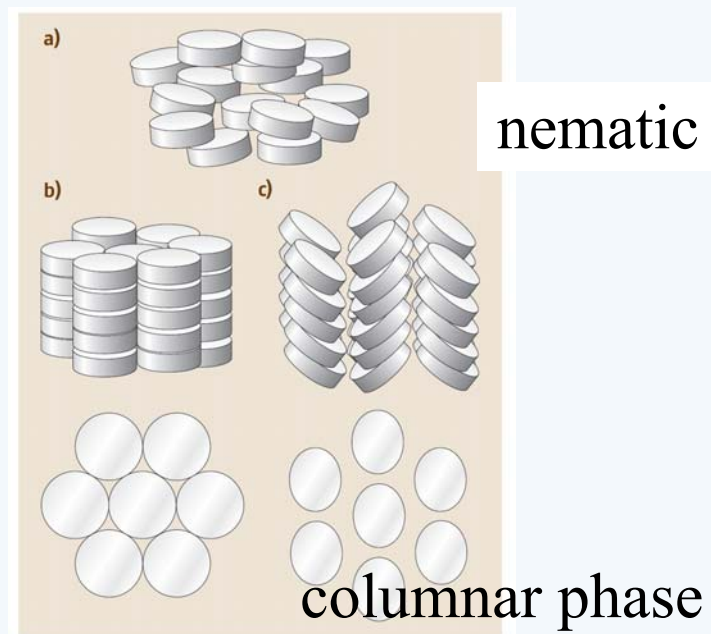
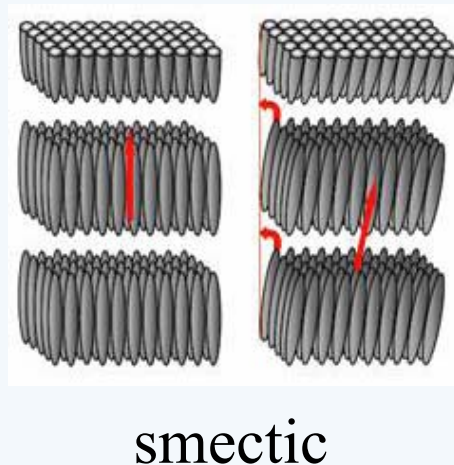
Adv. Mater., 2010, 22, 1060



Janus nanospheres in the bulk state

Phys. Rev. Lett., 2009, 103, 237801

the shape of Janus particles: an important design parameter to create self-assembled superstructures



smectic and columnar phases: internal phase transitions, 2D ordering with a hexagonal or rectangular symmetry within each smectic layer or in the plane orthogonal to columns

**non-spherical JNs: Janus character+orientational and positional ordering
complex superstructures and richer mesophases!!!**



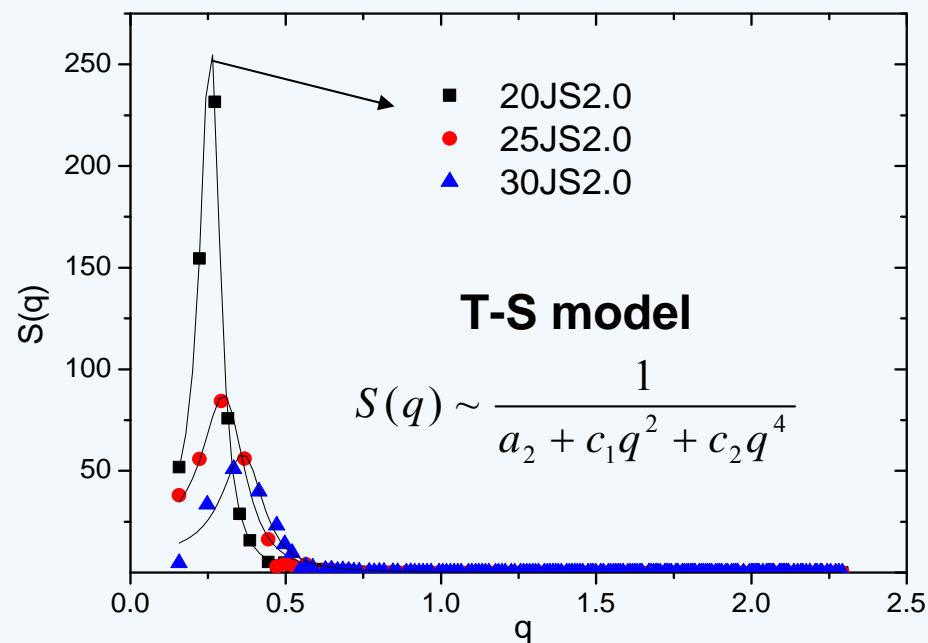
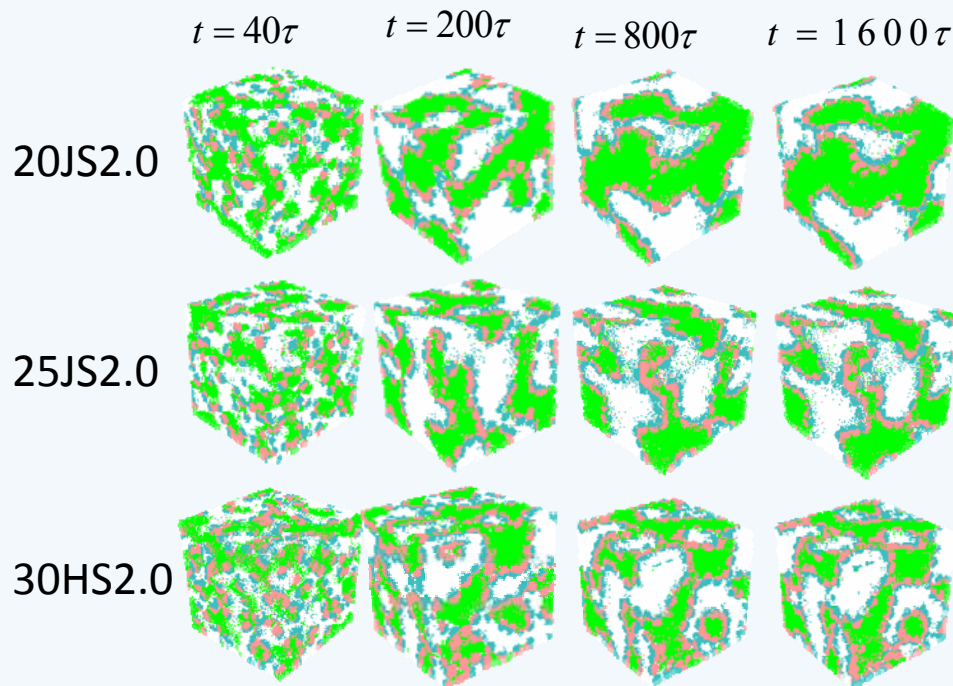
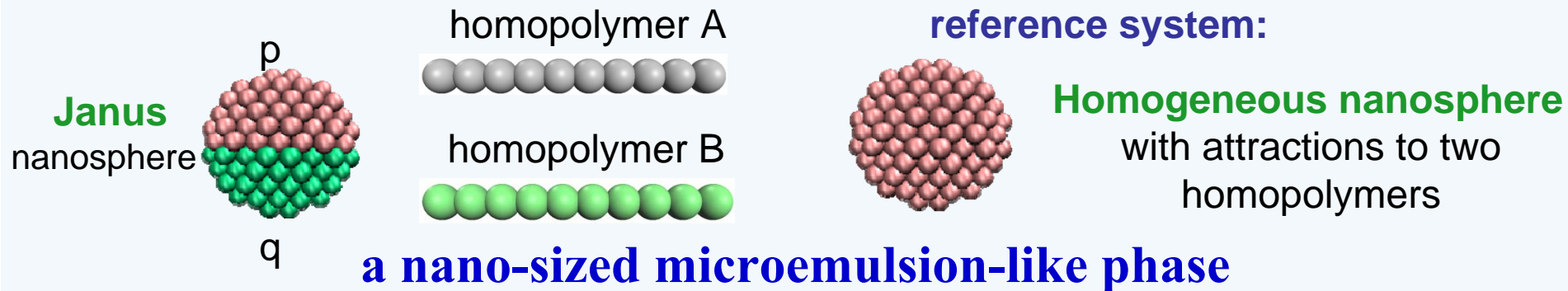
II. Application—2.nanoparticles filled polymer blends

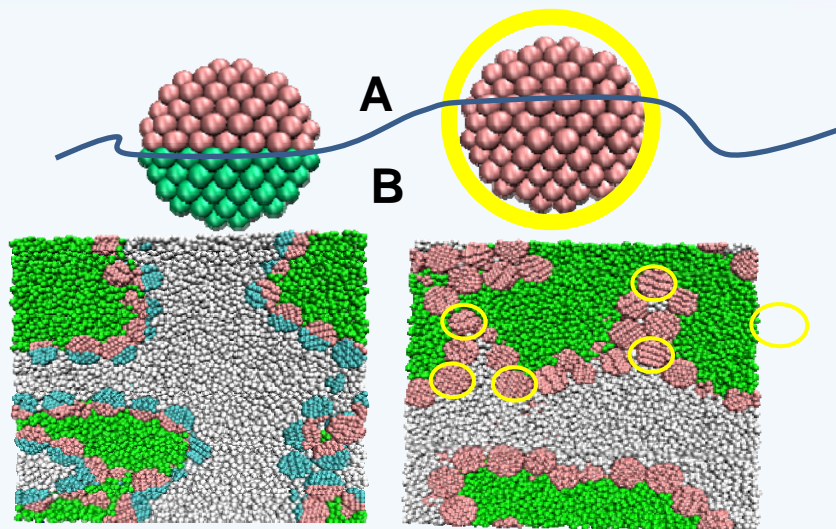
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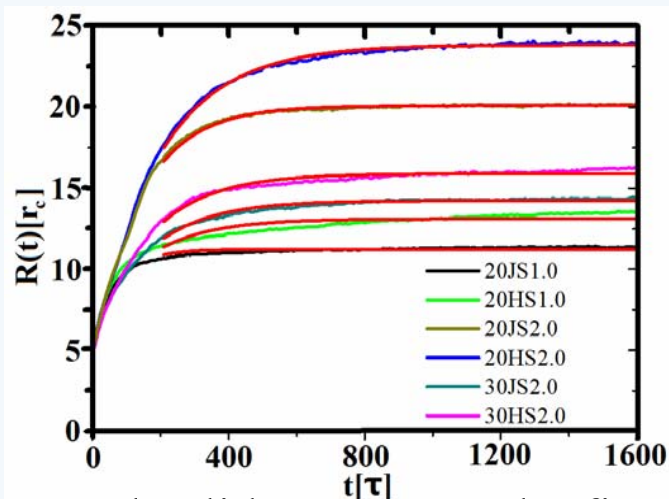
► The effect of Janus nanospheres on the phase separation of polymer blends





● equatorial adsorption

● low desorption probability



20HS1.0

20JS1.0

20HS2.0

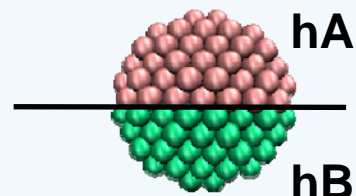
20JS2.0

30HS2.0

30JS2.0

Note: red solid curves are the fits to $R(t)$
using eqn (3) for $t > 200s$

Slowing-down kinetics :



Assumption:

1. adsorbed equatorially
2. no desorption

$$\text{Model H} \quad \frac{dR(t)}{dt} = \frac{\gamma}{\eta}$$

$$R(t) = \frac{\alpha R_{NS}}{\phi_{NS}} - \left(\frac{\alpha R_{NS}}{\phi_{NS}} - R_0 \right) e^{-t/\tau} \quad (1)$$

$$\tau = (\alpha \eta R_{NS}) / \gamma_{AB} \phi_{NS}$$

$$t \rightarrow \infty \quad R_{sat} \sim \alpha \frac{R_{NS}}{\phi_{NS}} \quad (2)$$

$$R(t) = R_{sat} - (R_{sat} - R_0) e^{-t/\tau} \quad (3)$$

● Dynamical scaling

self-similar behavior

at late stage of SD in symmetric binary fluids



A scaling of the structure factor

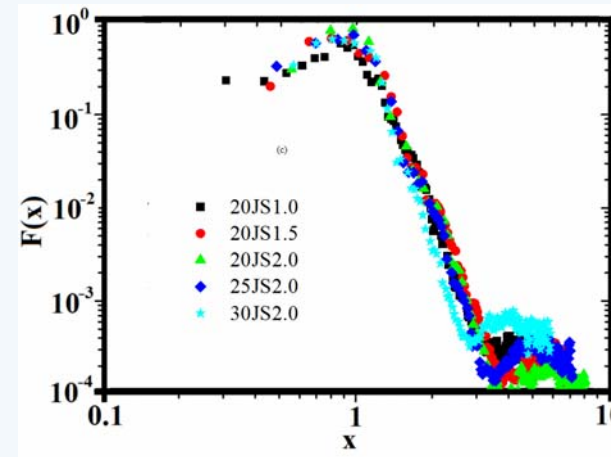
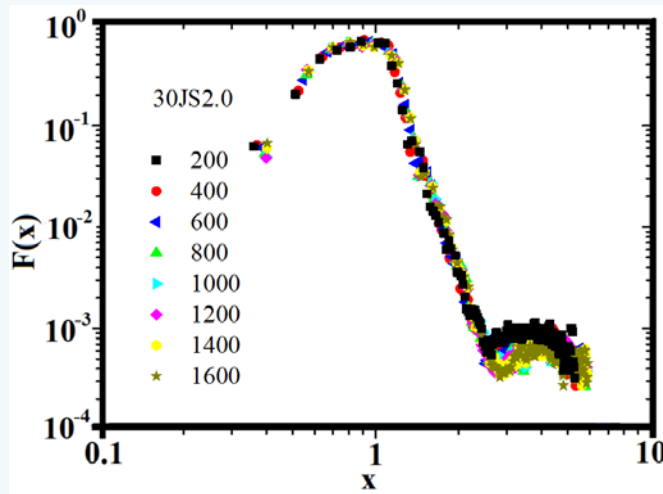
$$S(q,t) = [R(t)]^3 F(x)$$

$$x = q / q_2^{1/2}$$

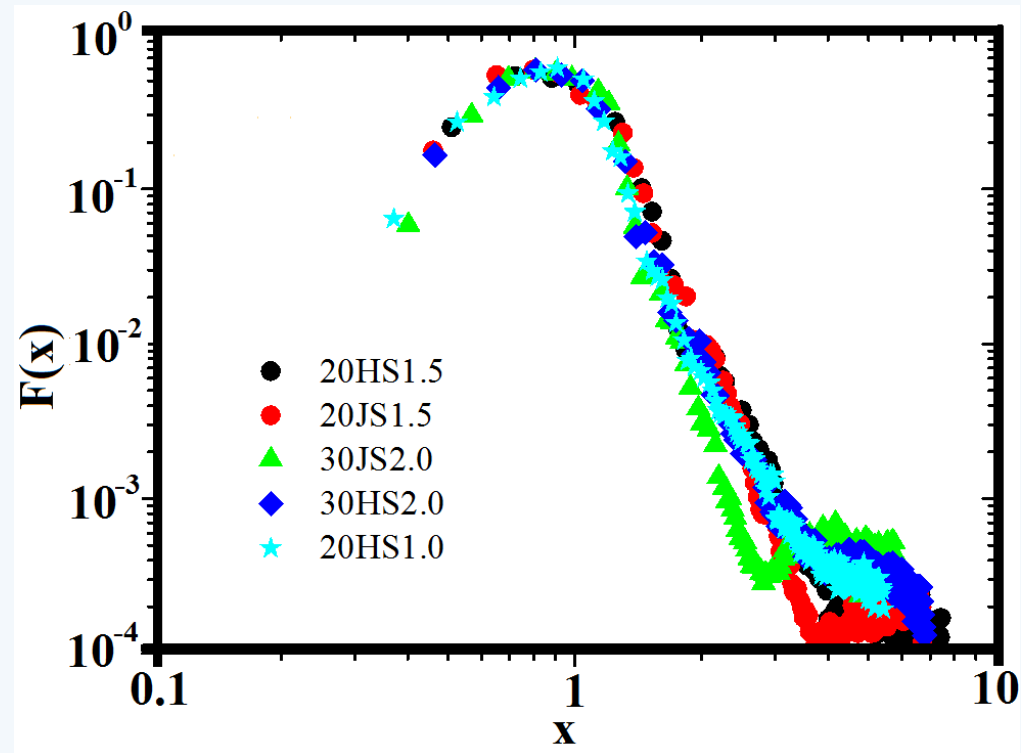
q_2 : second moment of structure factor $S(q,t)$

the time-independent scaling function

$$F(x) \equiv \frac{L}{\pi} q_2^{3/2} S(q,t) / \sum_{q=0}^{q=q_c} q^2 S(q,t)$$



a single characteristic length $R(t)$ can be used to characterize the global morphology evolution and the pattern grows with dynamical self-similarity



domain growth in the late stage obeys the same mechanism, although the volume fraction, radius, and composition of surface-active nanospheres could alter the growth rate of the phase-separated domains

● **crossover scaling**

for slower late-time phase separation dynamics

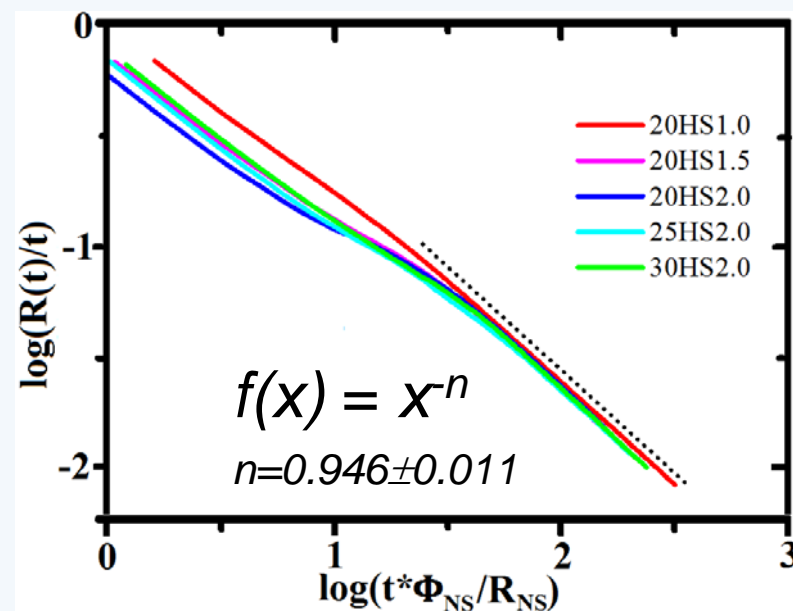
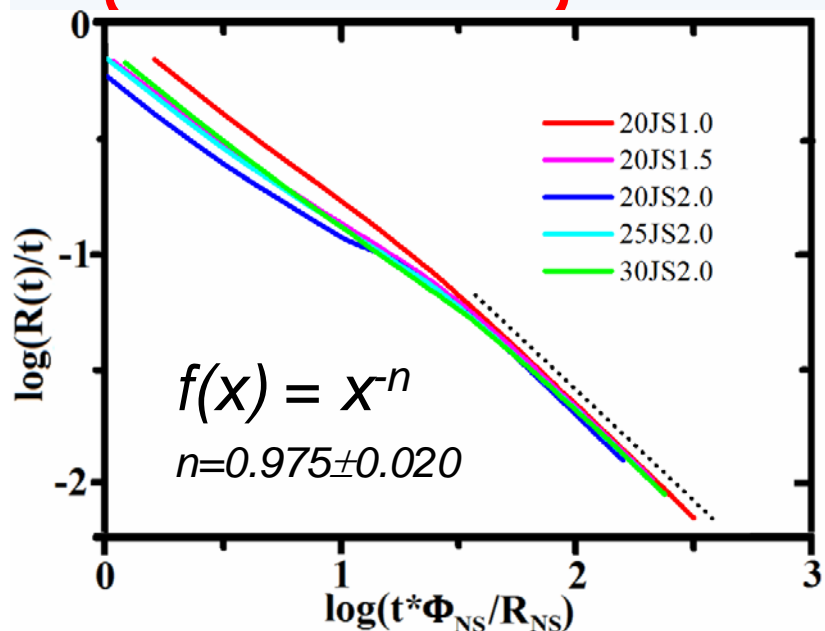
$$R(t) = t f \left(t \phi_{NS} / R_{NS} \right)$$

linking domain growth in pure binary polymers to that in the presence of these surface-active nanospheres

$f(x)$ is the crossover scaling function

$x = t \phi_{NS} / R_{NS}$ is the scaling variable

$x \rightarrow \infty \quad f(x) \rightarrow x^{-1}$



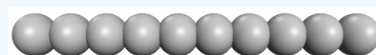


- Due to the inherent equatorial adsorption and low desorption probability Janus nanospheres significantly impede domain growth and at a later stage the average domain size approaches saturation and the growth exponent n decays to near-zero.
- In the later-stage of the phase separation process there exists a dynamical self-similarity in the ternary systems that undergo microphase separation and the domain growth follows a crossover scaling form.

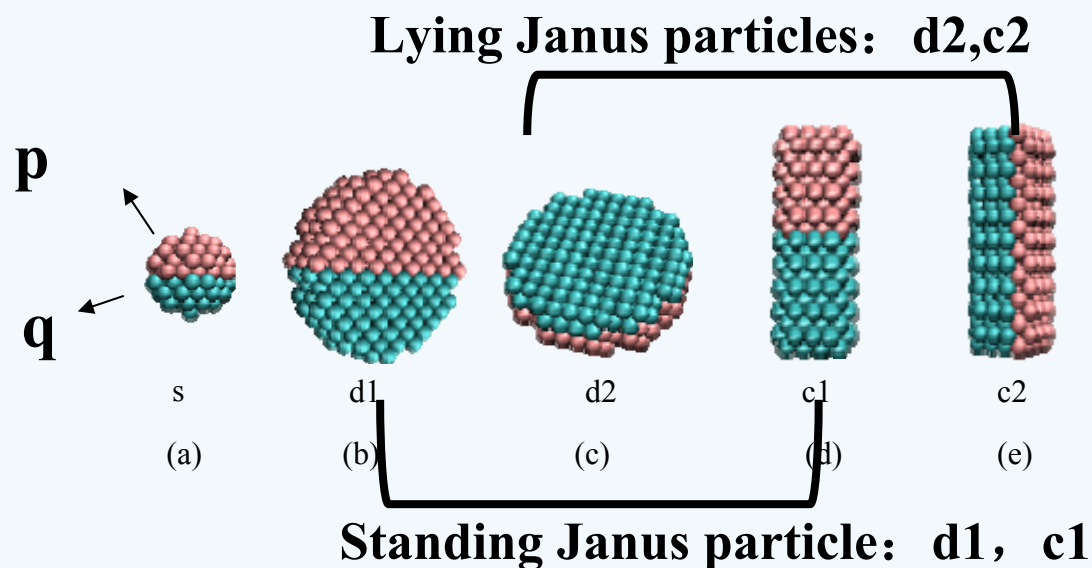
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➤ ordering and compatibilizing performance of Janus nanoparticles with various shapes and different dividing surface designs in immiscible polymer blends

Homopolymer A



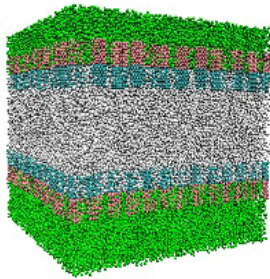
Homopolymer B



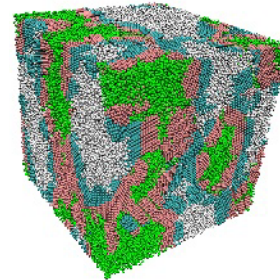
JNs : equal volume fraction
and equal net areas

aspect ratios : 2.8 and 0.25

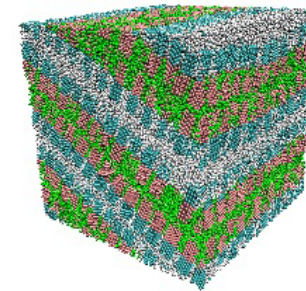
II. Application—2. nanoparticles filled polymer blends



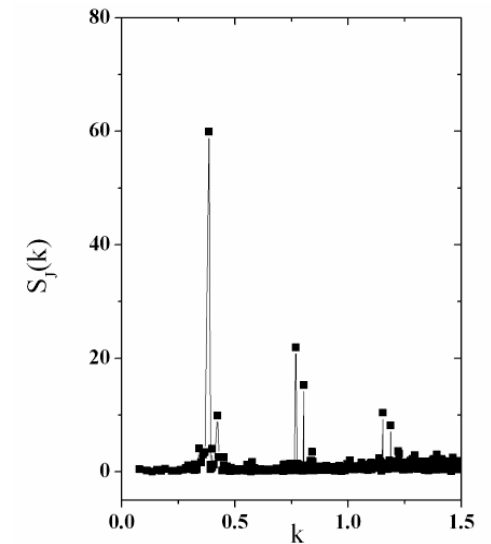
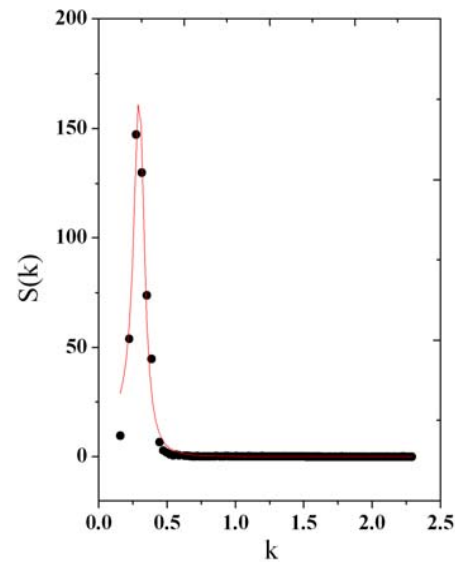
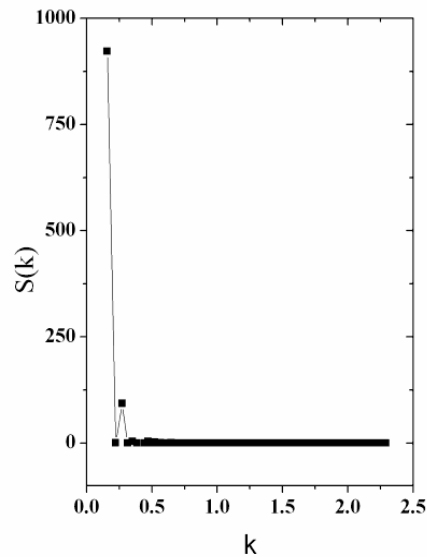
(a) macrophase-separated (2P) state



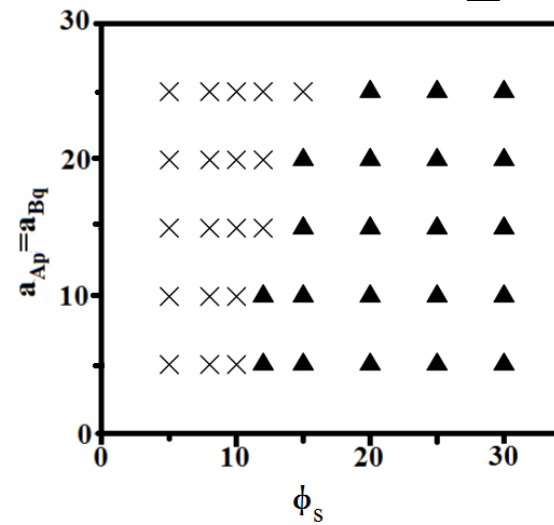
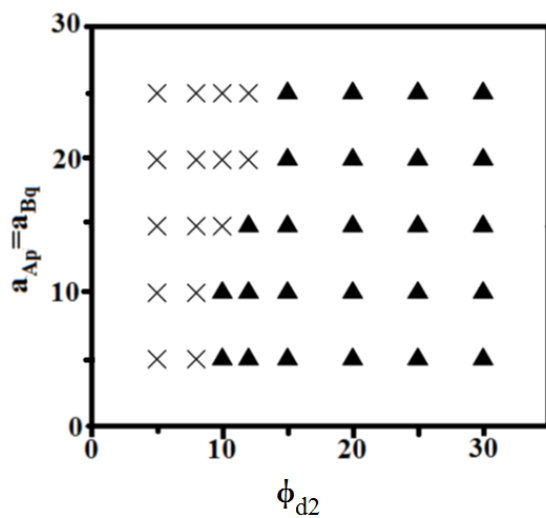
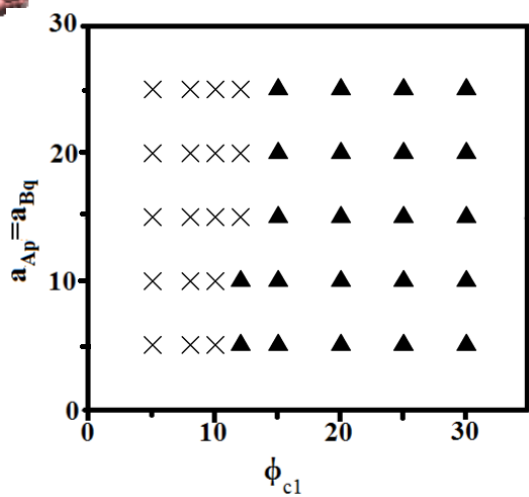
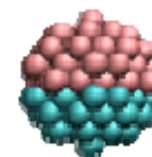
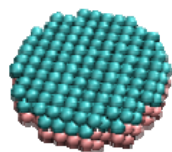
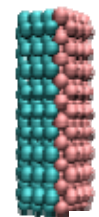
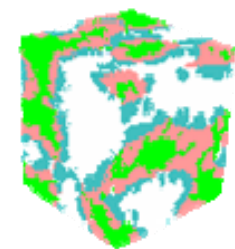
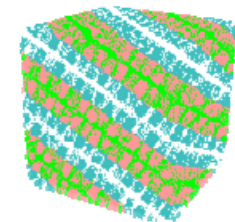
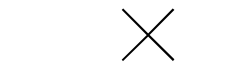
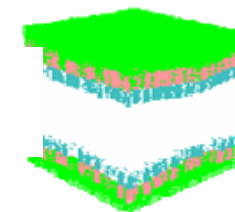
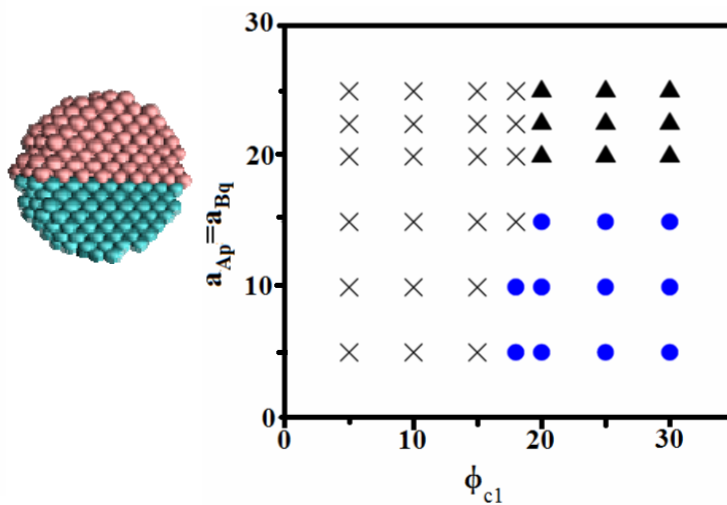
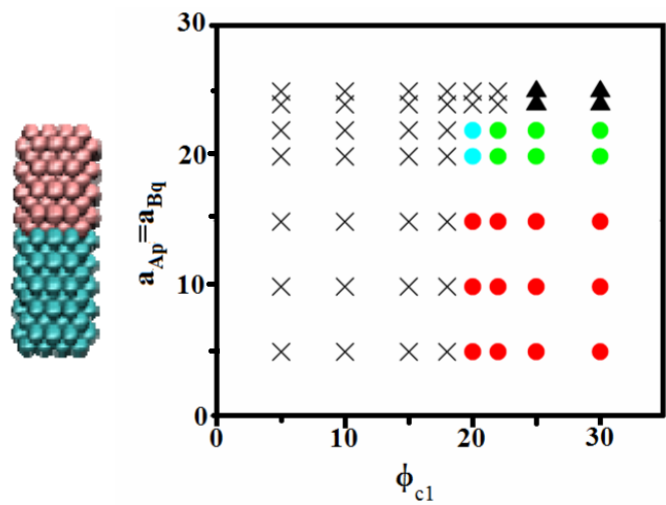
(b) bicontinuous microemulsion-like ($B \mu E$) phase

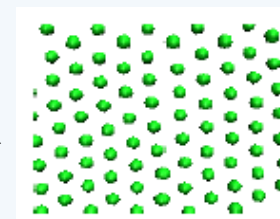
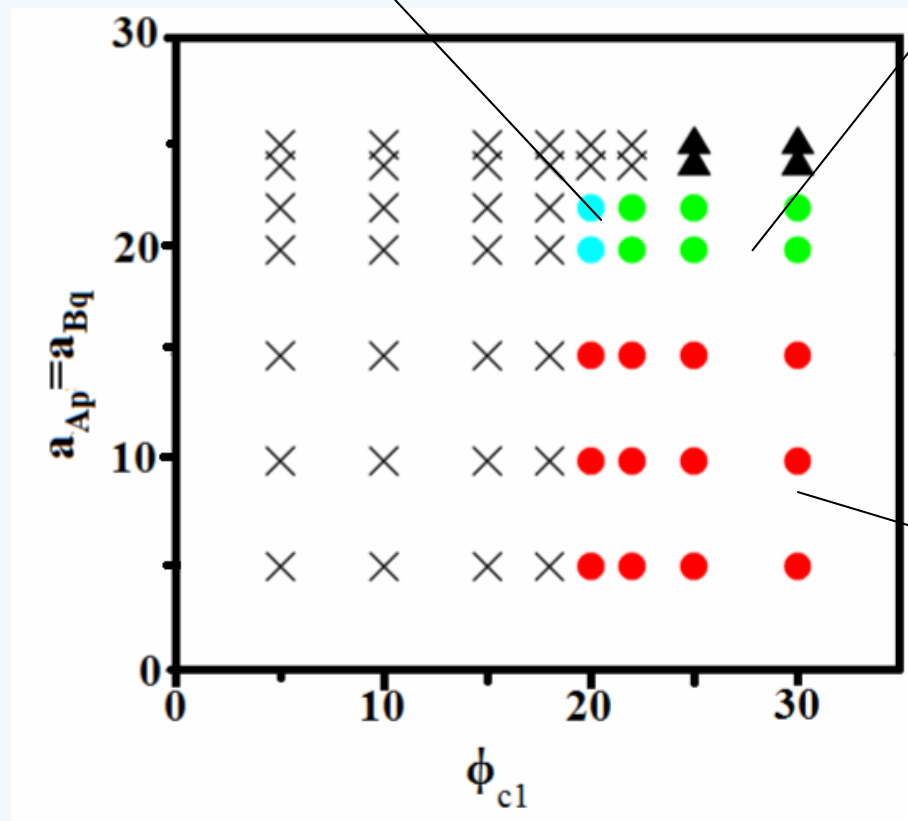
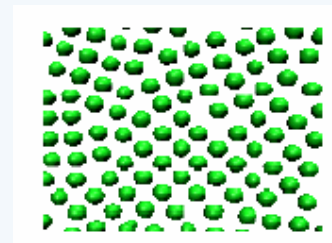
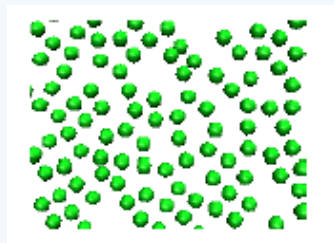
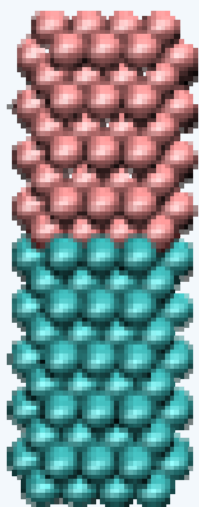


(c) lamellar (LAM) phase



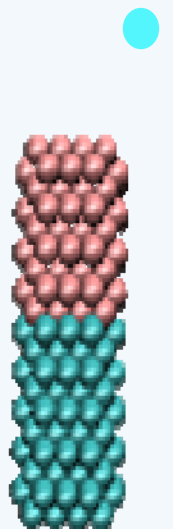
phase diagram



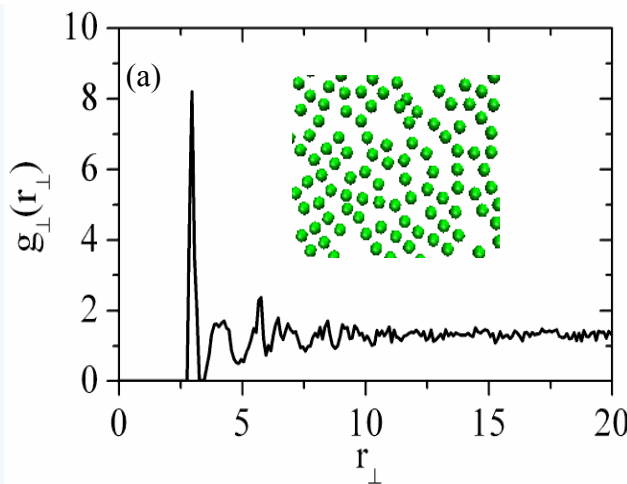




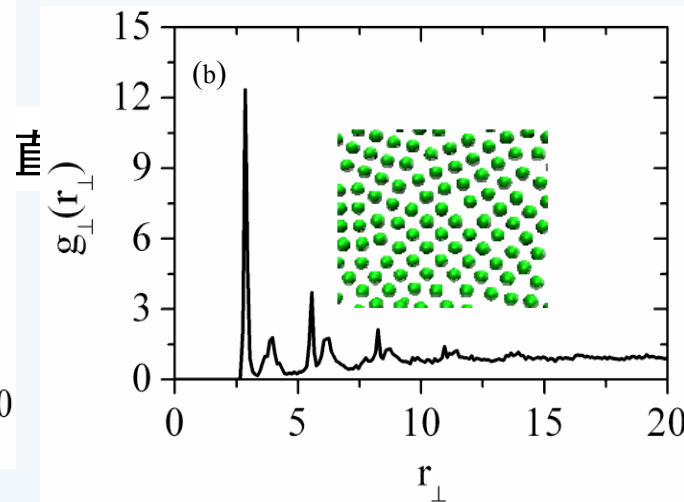
2d-ordering within the layers of the lamellar phase



$1:\sqrt{3}:2$

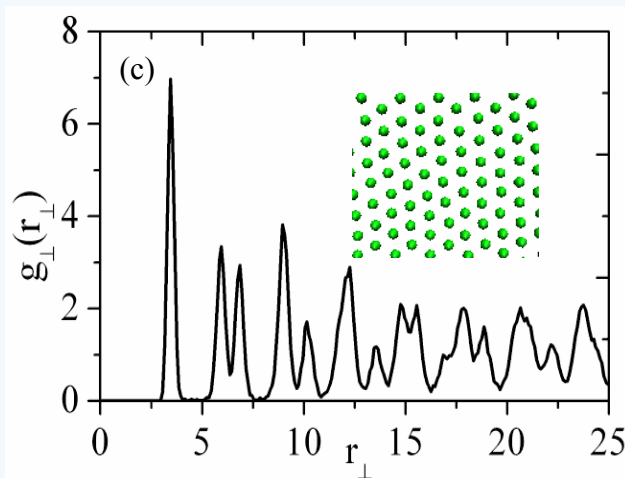


c1-20-20

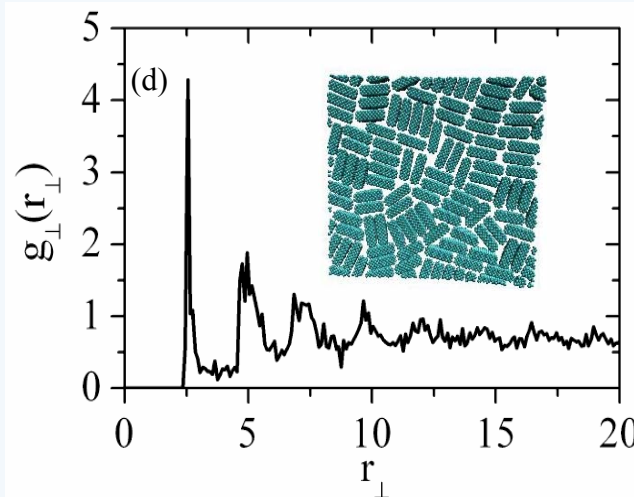


c1-25-20

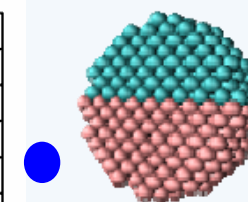
$1:\sqrt{2}:2$



c1-25-15



d1-25-10

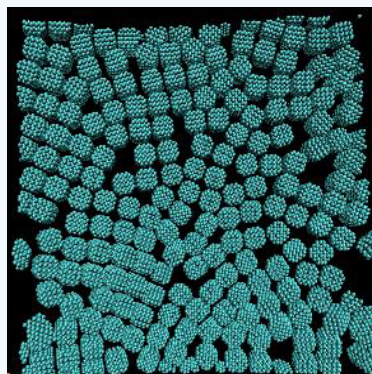


$1:2:3$

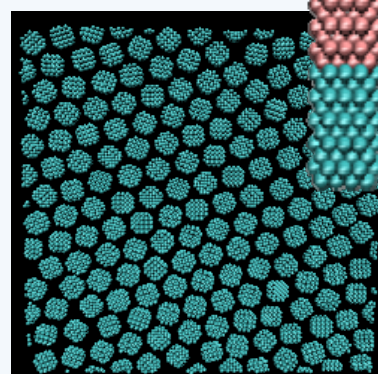
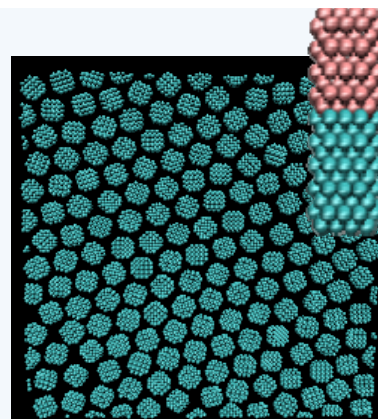
2D radial distribution function for the mass center of particles



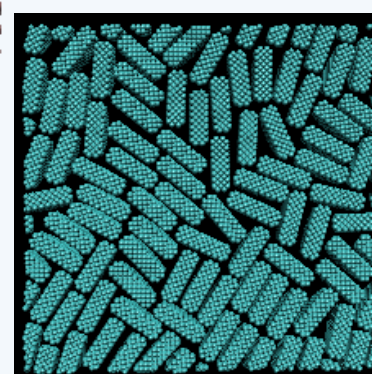
local 2d-ordering at the interface layers of the 2P phase



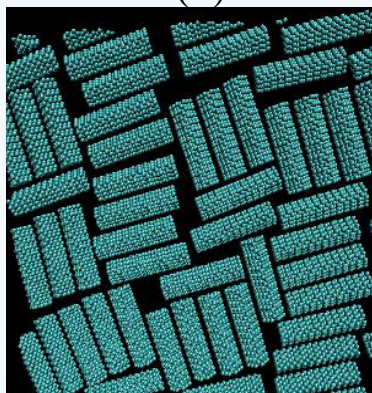
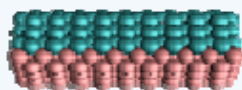
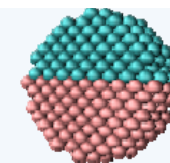
(a)



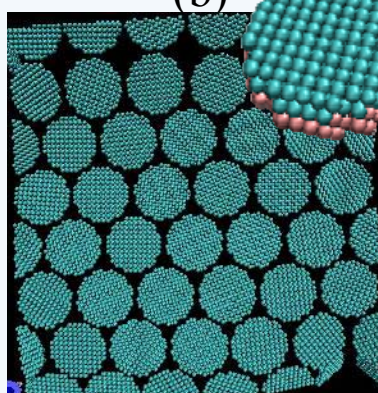
(b)



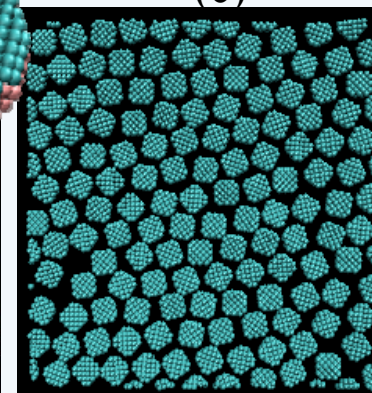
(c)



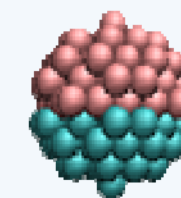
(d)



(e)

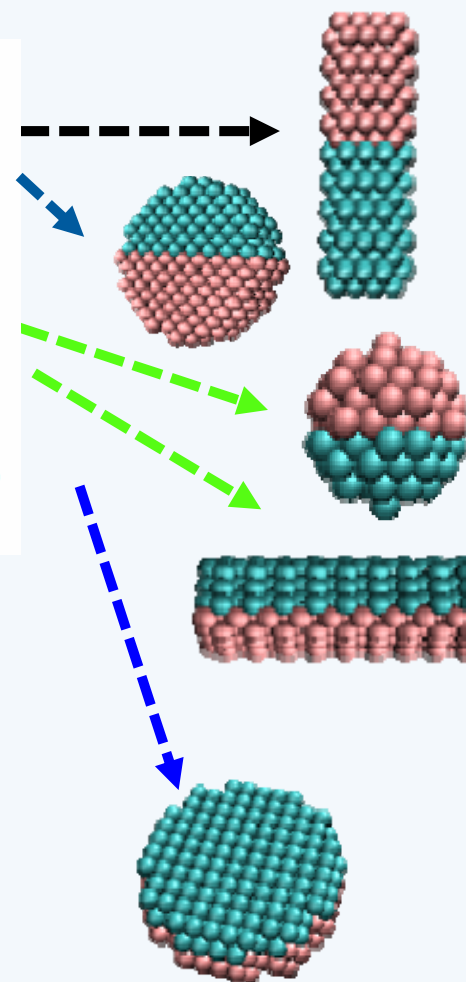
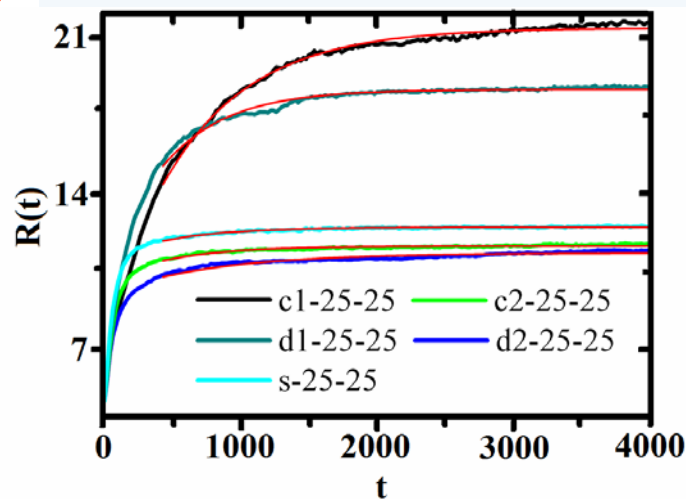
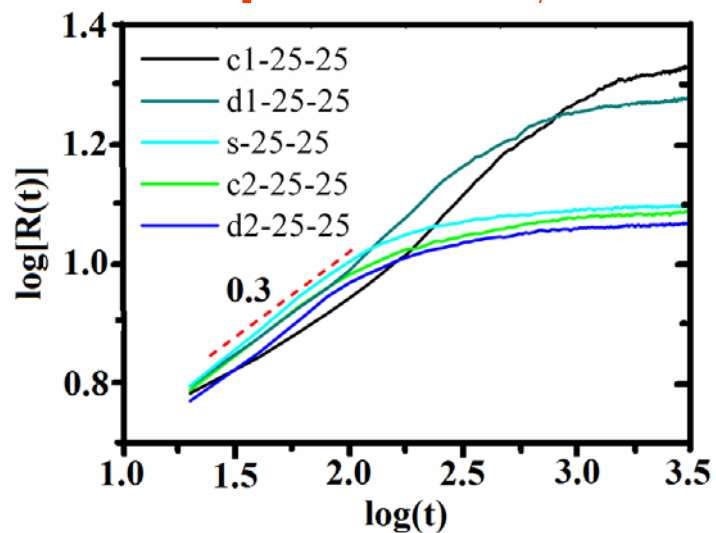


(f)



(a) c1-18-20, (b) c1-15-10, (c) d1-10-10,
(d) c2-5-10, (e) d2-5-10 and (f) s-6-10.

Phase separation dynamics



the time-dependent average domain size

at late times

$$R(t)_{d2} < R(t)_{c2} < R(t)_s < R(t)_{d1} < R(t)_{c1}$$

System type [⊖]	c1-25-25 [⊖]	c2-25-25 [⊖]	d1-25-25 [⊖]	d2-25-25 [⊖]	s-25-25 [⊖]
JN type [⊖]	c1 [⊖]	c2 [⊖]	d1 [⊖]	d2 [⊖]	s [⊖]
JN number [⊖]	570 [⊖]	570 [⊖]	398 [⊖]	398 [⊖]	1193 [⊖]
Total dividing surface area of JNs, $A_{tot}(r_c^2)$ [⊖]	2422.5 [⊖]	8800.8 [⊖]	3458.6 [⊖]	10853.5 [⊖]	8136.3 [⊖]
$R_{sat}(r_c)$ [⊖]	21.72 [⊖]	12.21 [⊖]	18.87 [⊖]	11.69 [⊖]	12.63 [⊖]

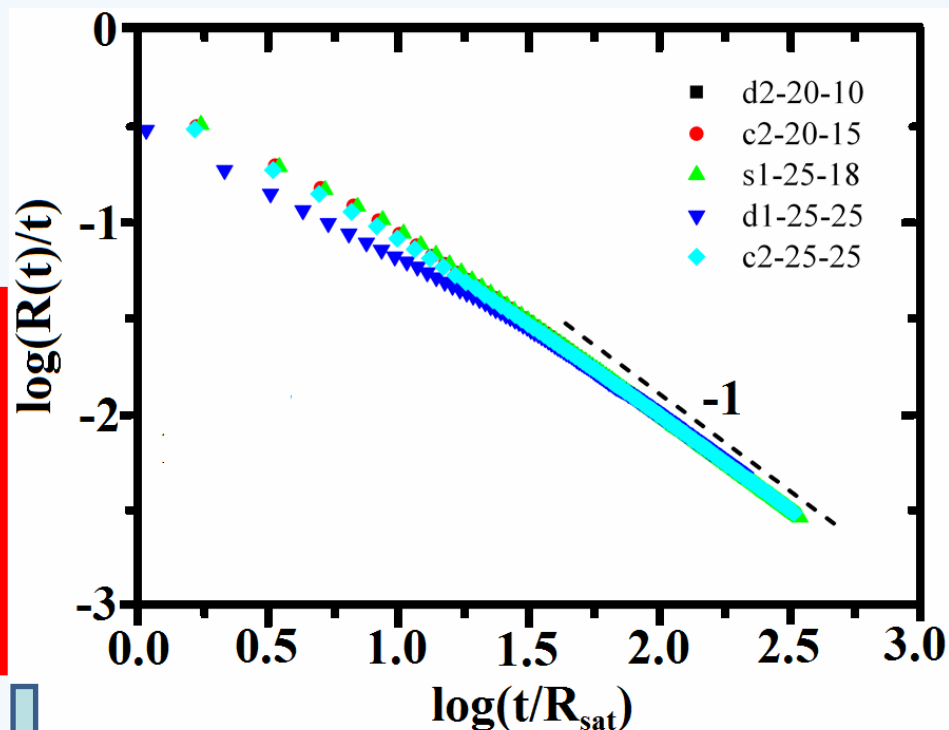
$$A_{tot}^{d2} > A_{tot}^{c2} > A_{tot}^s > A_{tot}^{d1} > A_{tot}^{c1}$$

crossover scaling

$$R(t) \approx t f \left(t / R_{sat} \right)$$

$$x \rightarrow \infty \quad f(x) \rightarrow x^{-1}$$

The general mechanism
of JNs to stabilize immiscible
homopolymers



JNs : powerful compatibilizers, greatly reducing the interfacial tension and leading to the formation of bicontinuous microemulsion-like structures



- with the addition of 5%~30% JNs a **BuE** structure is observed in all ternary systems while the **LAM** phase is only formed in systems with “standing” anisotropic JNs;
- lamellar stacks of “standing” JNs assume **several different in-layer structures, which arise from the strong excluded-volume interactions and depend on the particle shape;**
- the addition of JN slows down the phase separation, compatibilizing performance is related to the total dividing surface area of JN which is determined by the particle shapes and dividing surface designs.
- domain growth dynamics at the late-time phase separation process follows a crossover scaling form regardless of the particle shapes and dividing surface designs.

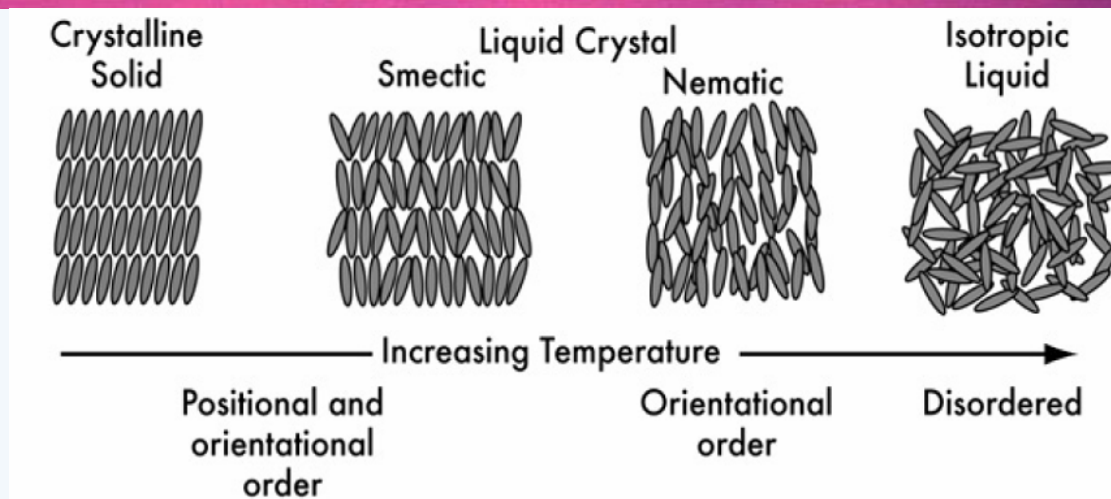
Rich lateral ordering behavior of JNs at the polymer blend interface has practical implications for the precise positioning of nanoparticles to fabricate nanostructured functional materials.



II. Application—3. LCs or its related complex systems

- The phase behavior, structure, and dynamics of rodlike mesogens with various flexibility
- The anchoring transitions driven by rod–coil amphiphiles at aqueous–liquid crystal
- the Phase Behavior of T-Shaped Ternary Amphiphiles Possessing Rodlike Mesogens

II. Application—3 rodlike mesogens with flexibility



generic single site models :

- hard particle models : ellipsoids and sphero-cylinders with hard core repulsions
- computational simplicity, valid for entropically driven lyotropic LC behavior but not for the thermotropic LCs
- soft particle models : Gay - Berne GB potential model
- soft ellipsoids with well-defined anisotropic attractive and repulsive interactions, a more realistic model

most LC molecules, LC polymer, exhibit some degree of flexibility
have large effect on the stability of particular LC phase!!



➤ effect of molecular flexibility on the LC phase behavior

Why DPD not MD:

- Mesophase formation in thermotropic LCs usually occurs in the mesoscopic length- and time-scales

1. rigid mesogen model



(Rod7、Rod6、Rod5)

+ Rattle constraining algorithm

(to keep the beads aligned and equidistant)

DPD model
for mesogen

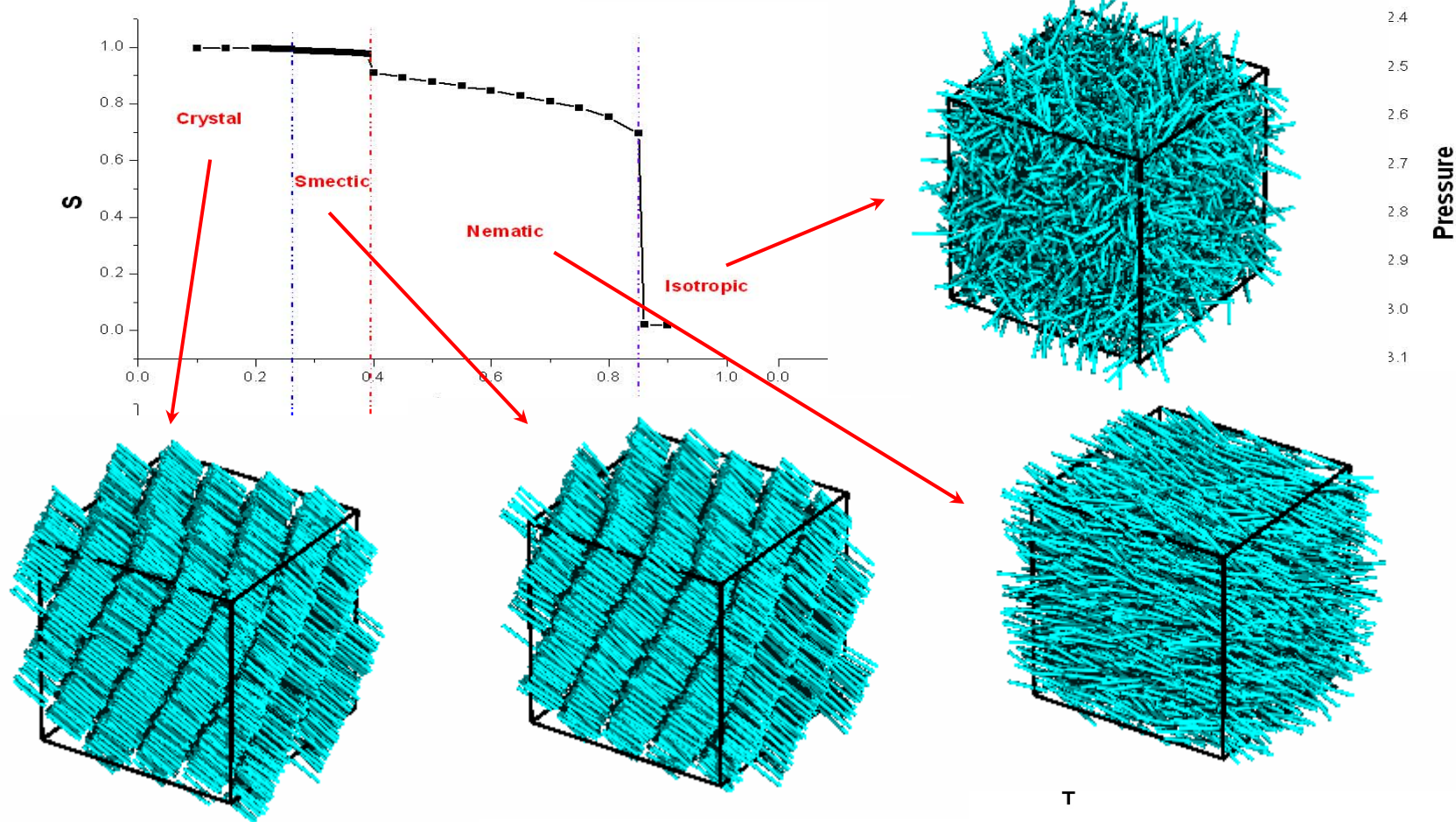
2. semirigid mesogen model: semirigid bead-spring model



+ bending potential

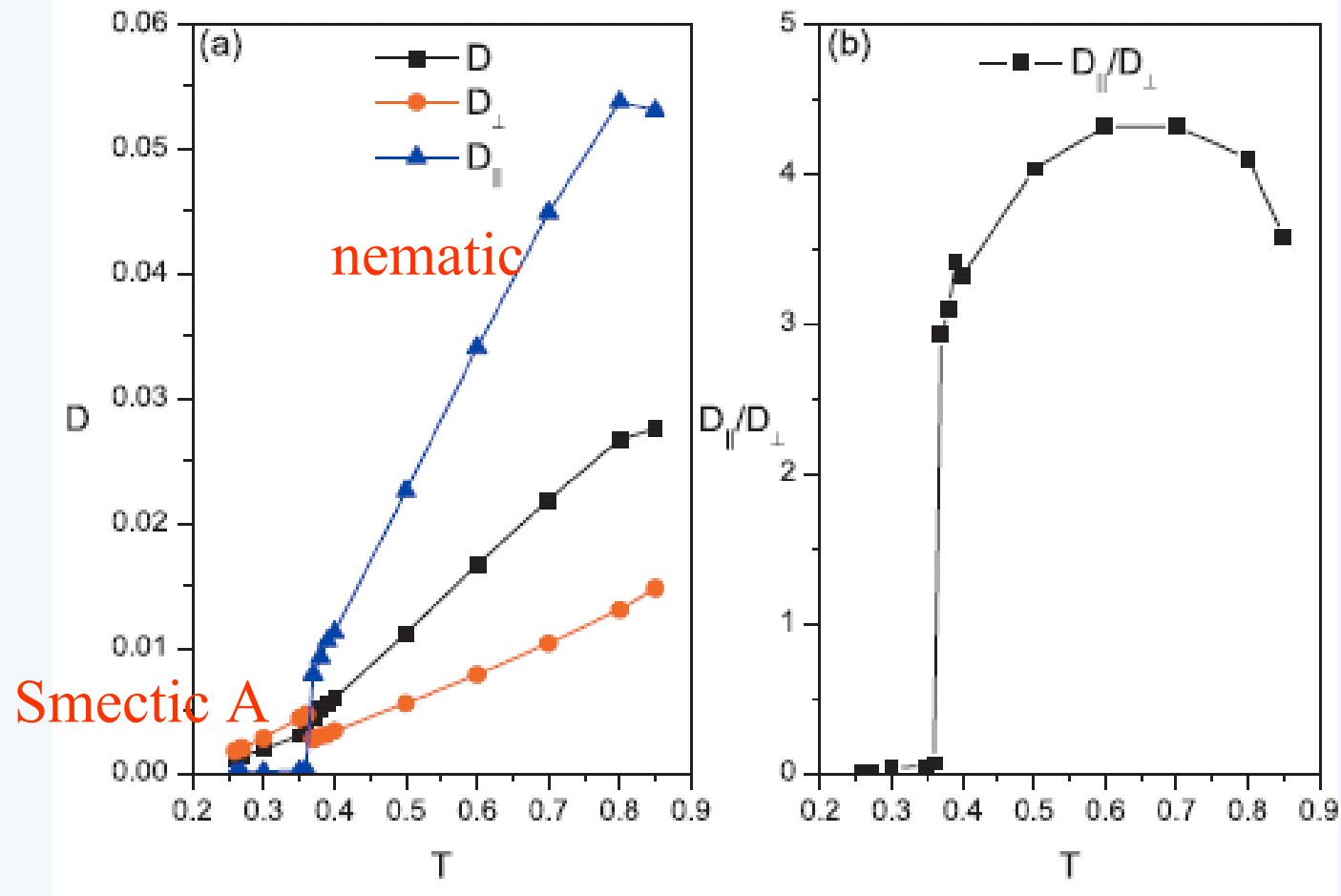


● Phase behavior of the rigid mesogen R₇



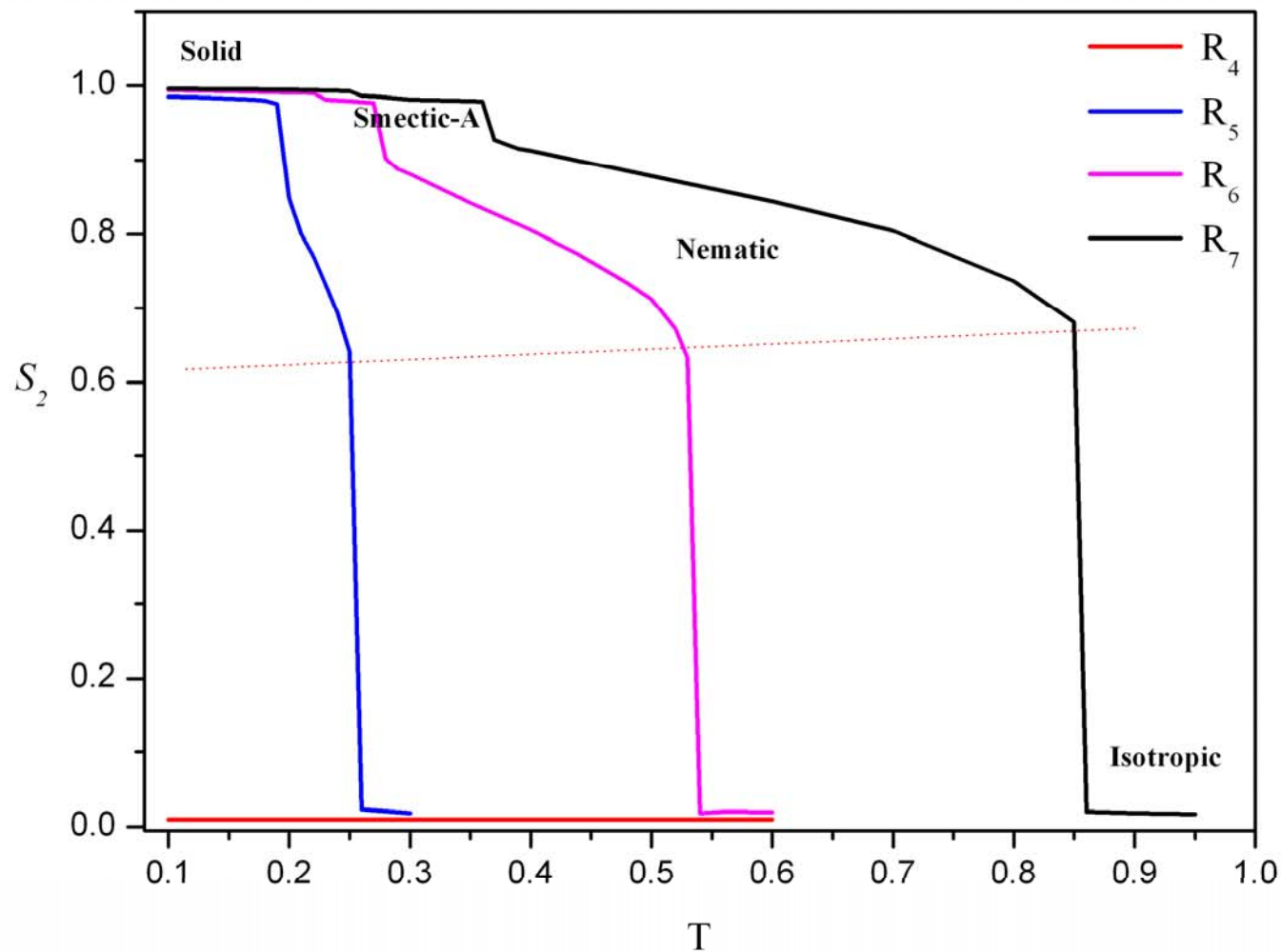


● The diffusion properties of R₇ in various LC phase



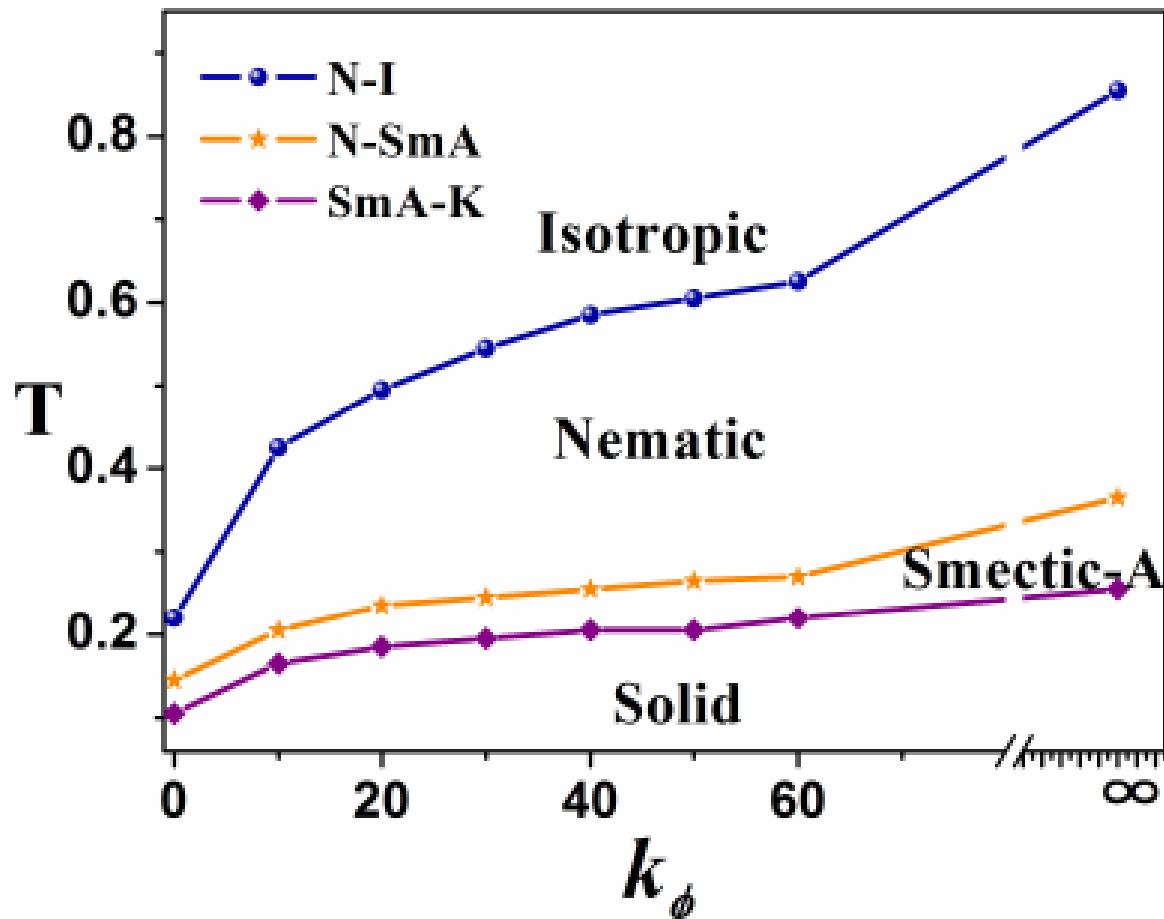


● The effect of rod length on the phase behavior





● The effect of molecular flexibility, k_{ϕ} on the phase behavior



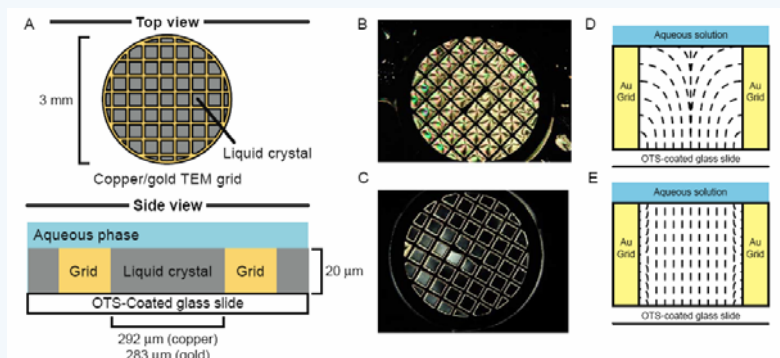


- Both the rigid model and the semirigid model capture the essential molecular features necessary for the formation of LC phases.
- The static and dynamic properties of LC phases have been efficiently reproduced.

Our extensive DPD simulations on these two models provide an important insight into the effect of molecular flexibility on thermodynamical, dynamical, and structural properties of the LC phases.

It is clear that dissipative particle dynamics is an extremely efficient mesoscale technique for numerical simulation of LC phases.

II. Application—3 rod-coil amphiphiles at aqueous-liquid crystal

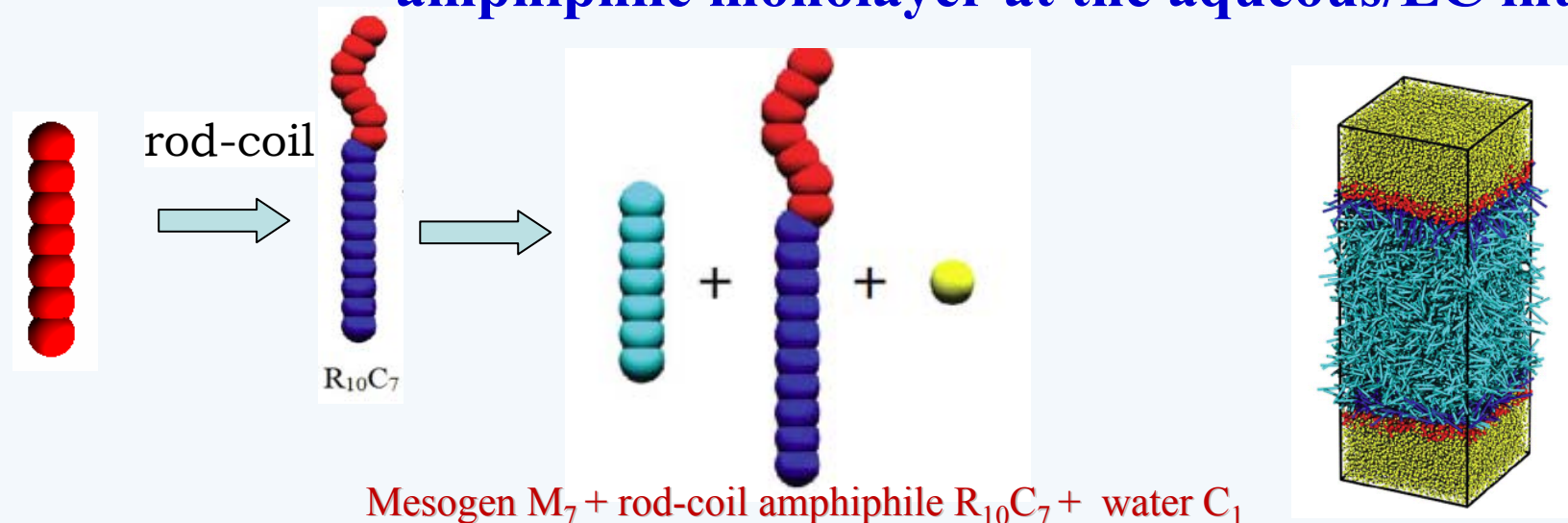


biological and medical sensor

anchoring transition of LCs at aqueous-LC interfaces provides a powerful tool for probing and amplifying the molecular events at the interfaces into visual optical imaging.

Nature **2007**, 6, 929-938

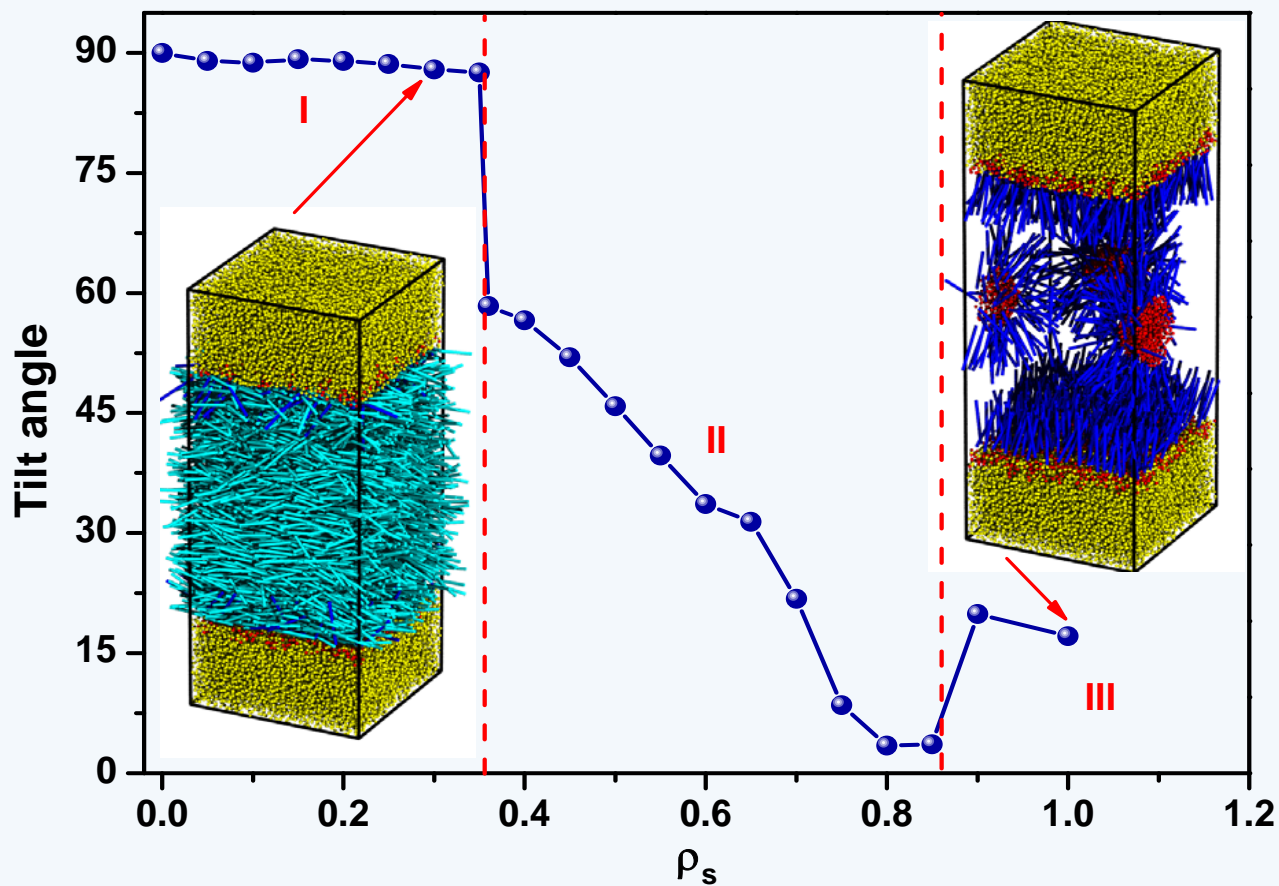
● anchoring transitions in the presence of amphiphile monolayer at the aqueous/LC interface



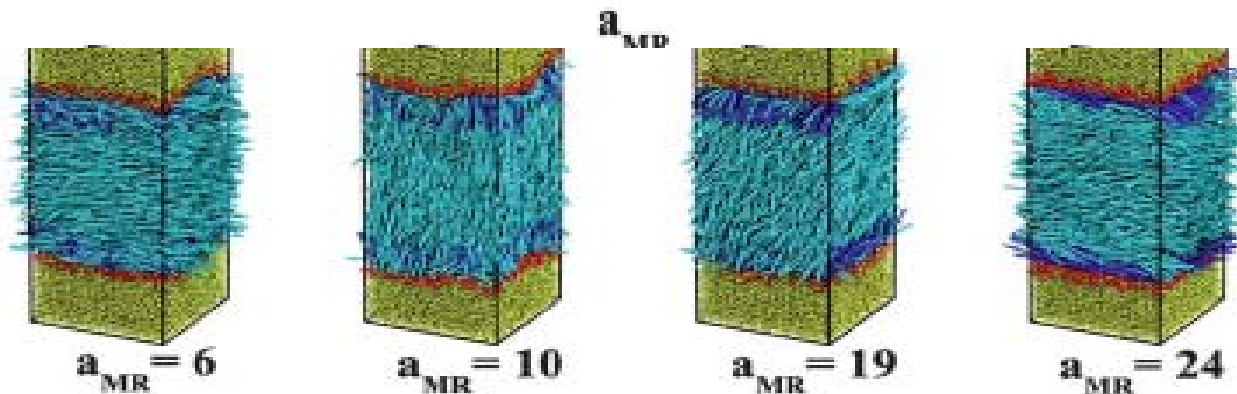
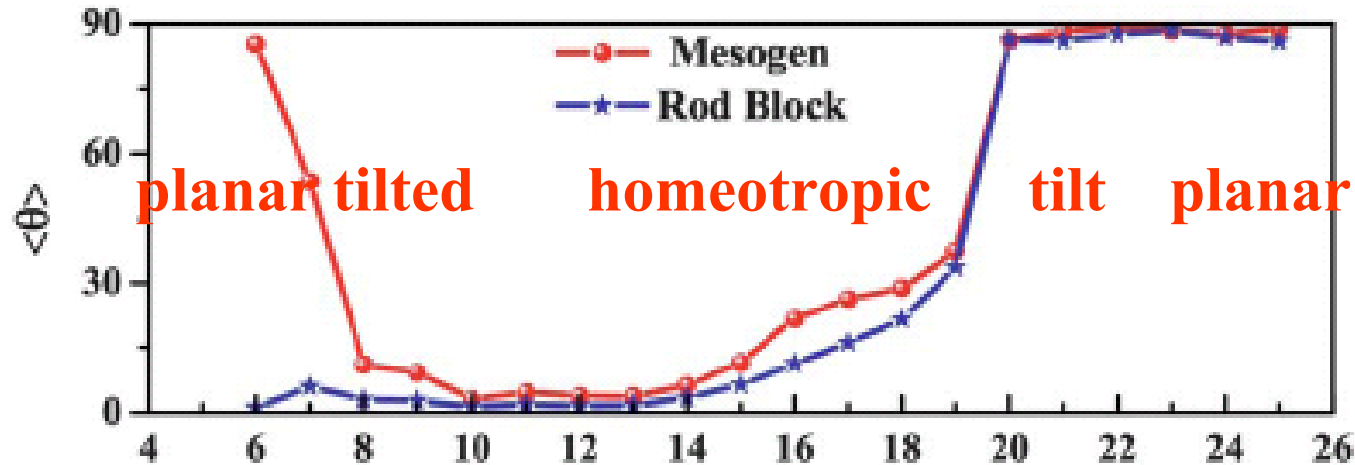
Soft Matter, 8, 5168-5174(2012)

➤ II. Application—3 rod-coil amphiphiles at aqueous-liquid crystal

Anchoring transition with rod-coil content



● Influence of the repulsion interaction parameter between the mesogens and the rod blocks of amphiphiles (aMR) on the anchoring behavior of liquid crystals

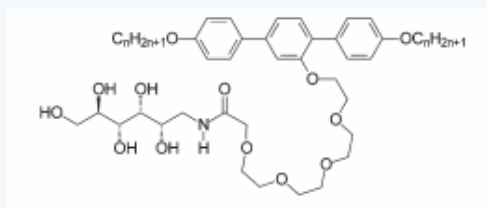


tuning a_{MR} , which is essentially equivalent to changing the chemical constitution of the amphiphiles or the mesogens, enriches the possibility of tailoring the anchoring behavior at aqueous-LC interfaces.

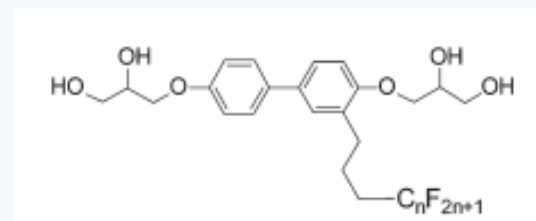
➤ II. Application—3 T-Shaped Ternary Amphiphiles Possessing Rodlike Mesogens

To fabricate the complex and highly ordered nanoscale or mesoscale structures, the practical strategy is to build molecules with the appropriate building blocks--“**bottom up**” design

T-shaped ternary liquid crystals (TLCs)



bolaamphiphile



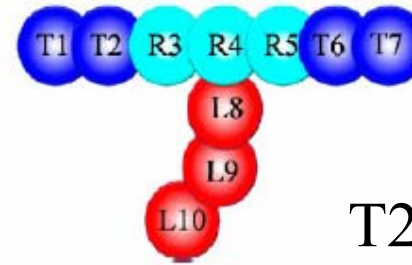
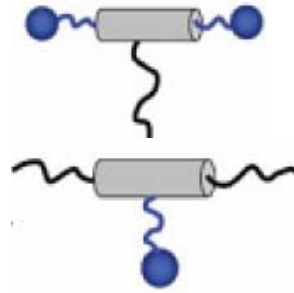
Facial amphiphile

Our interest

to understand how controlling the sizes of both terminal and lateral substituents affects the complex interplay of the entropic and enthalpic incompatibility of the system and in turn tunes the morphology formation of complex phases



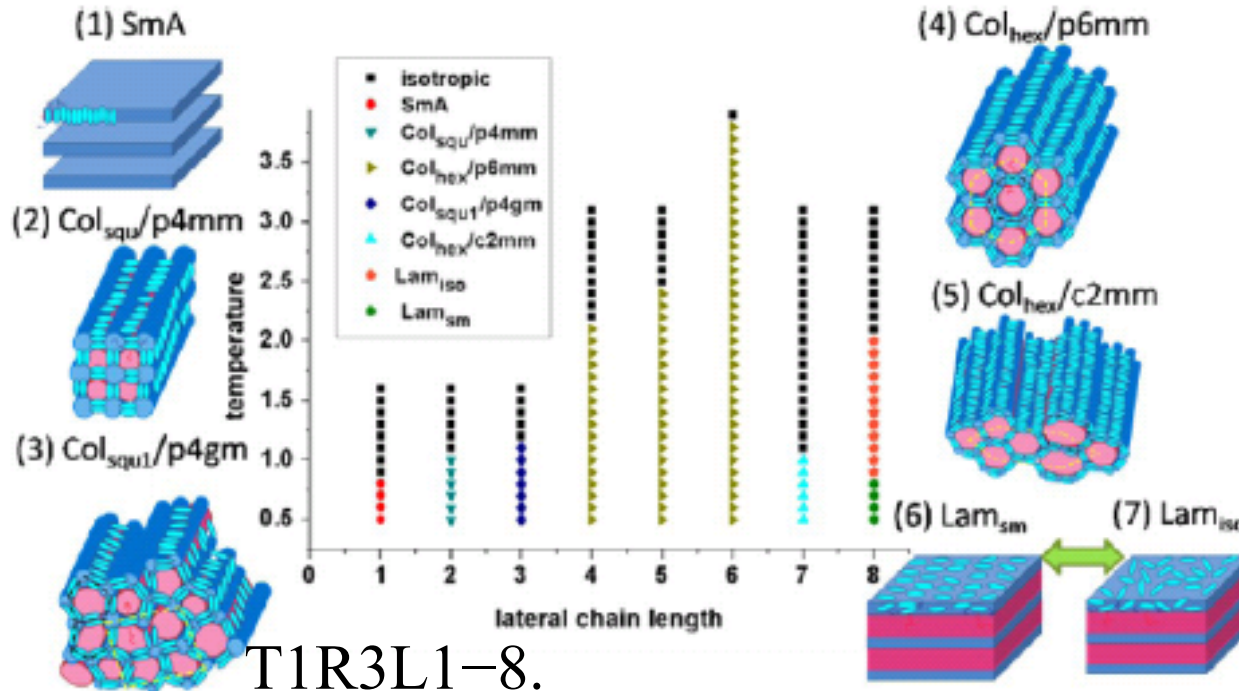
Semirigid mesogen

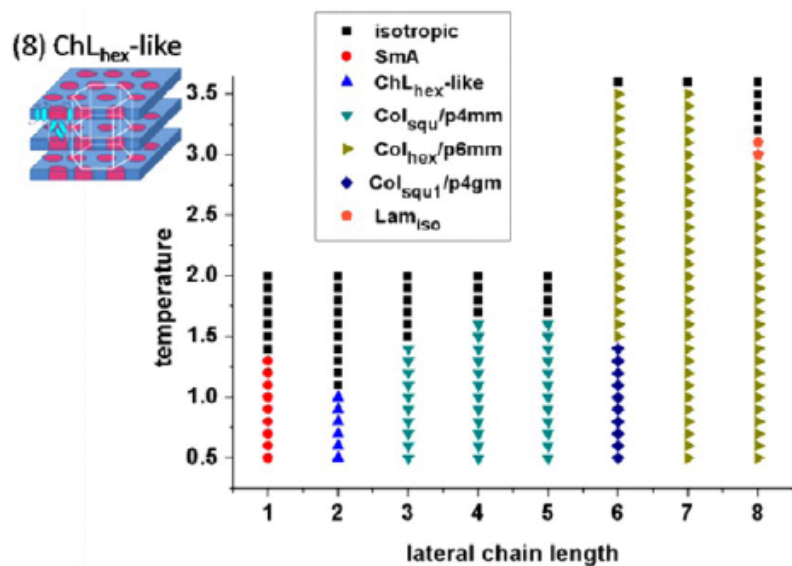


T2R3L3

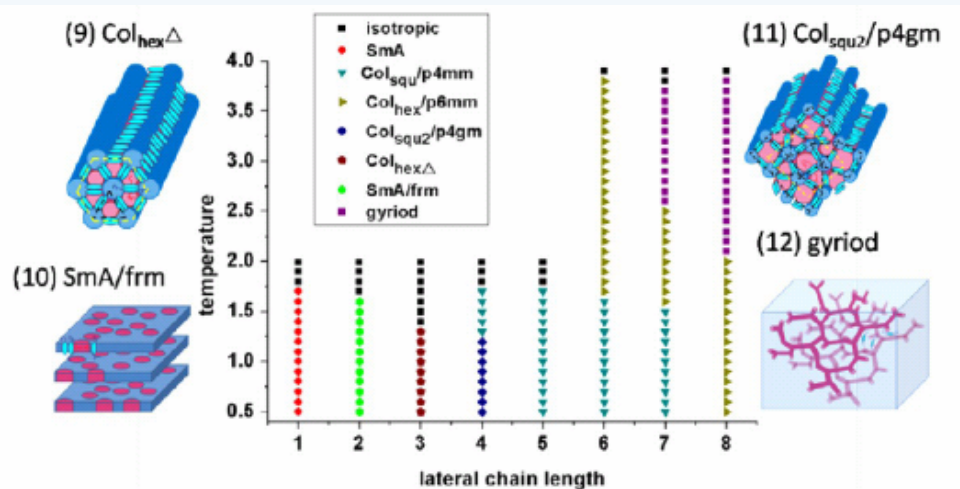
T-Shaped Ternary Amphiphile model

● Phase Behavior of T-Shaped Ternary Amphiphiles Possessing Rodlike Mesogens





T2R3L1-8.



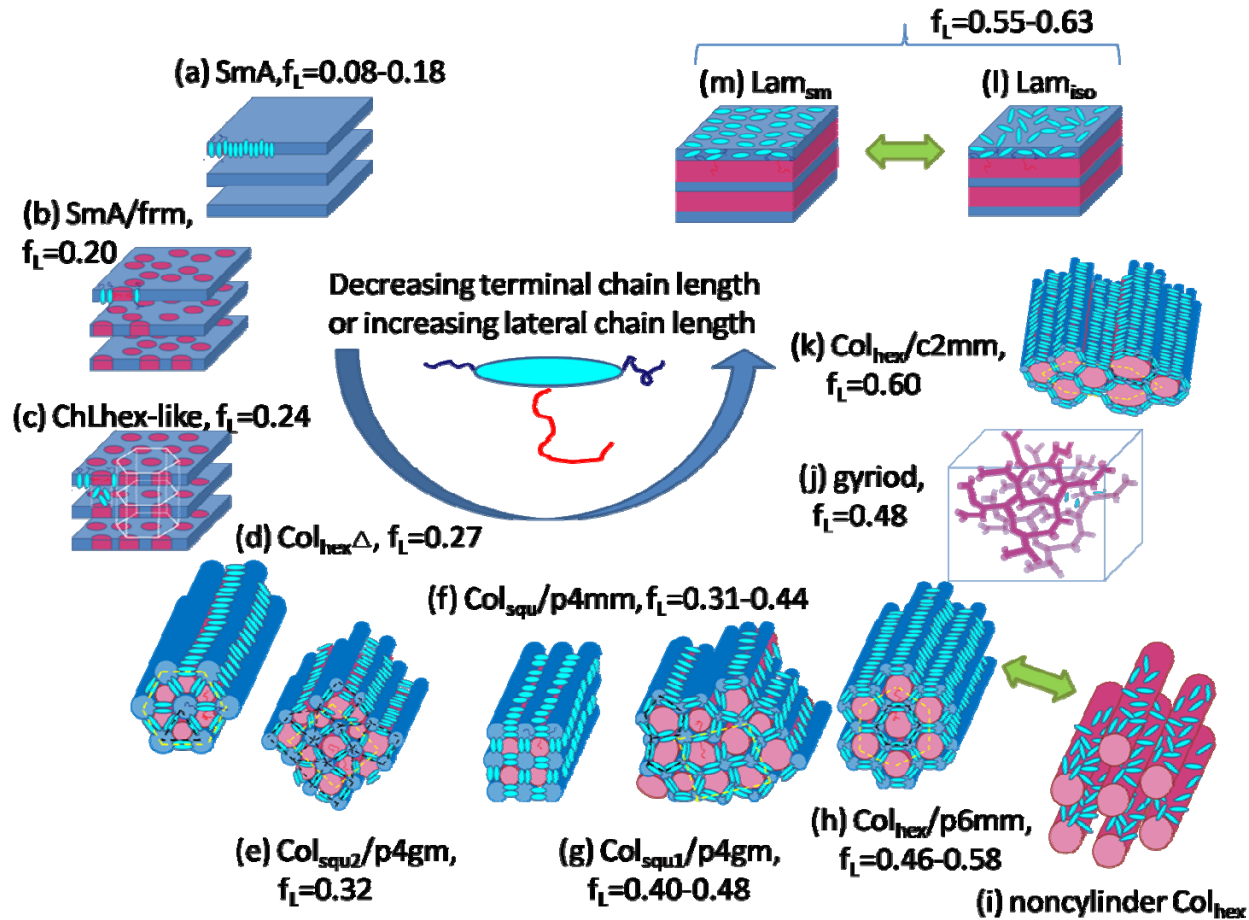
T3R3L1-8.

The volume fraction of lateral chains in TLCs is an important variable to influence the liquid crystalline behavior

$$f_L = \frac{(d_L^{\text{prob}})^3 \times N_L}{(d_R^{\text{prob}})^3 \times N_R + (d_T^{\text{prob}})^3 \times N_T + (d_L^{\text{prob}})^3 \times N_L}$$

J. Phys. Chem. B, 117, 9106 (2013)

the universal phase diagram as a function of effective volume fraction of lateral chains



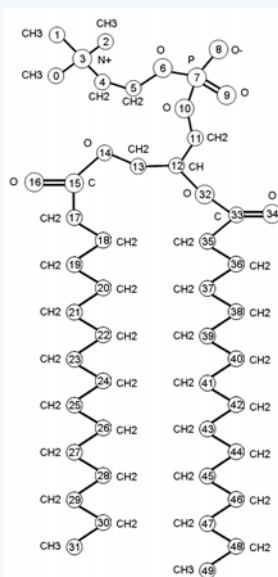
- quantitatively compared with the experimental results
- a unique view to understand the universal phase behavior in real TLCs systems



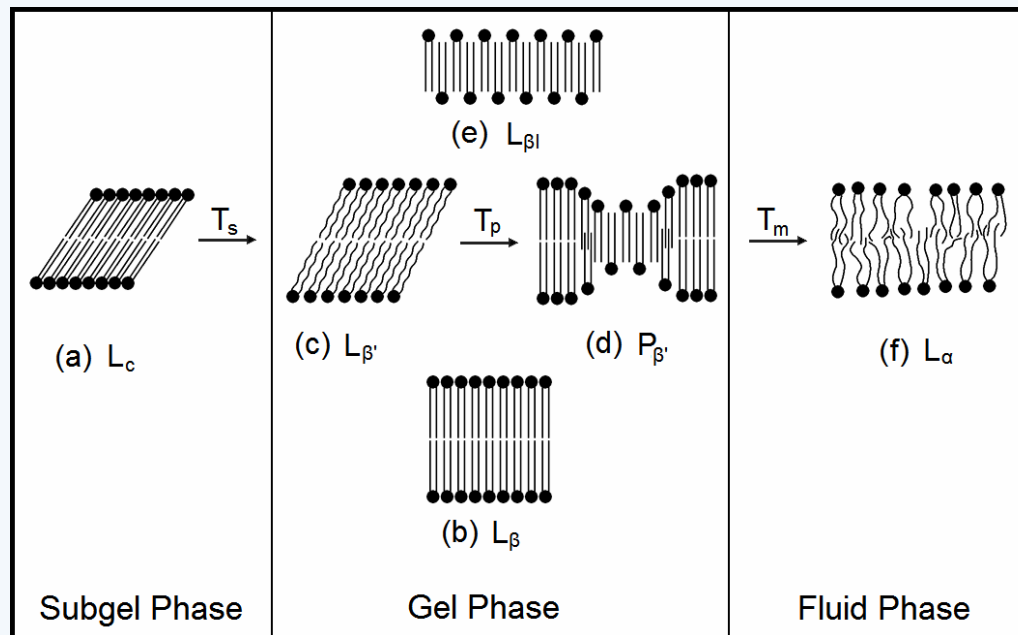
II. Application—4. lipid

- **Phase Behavior of Lipid Bilayers**
- **Protein-Mediated Vesicle Fusion**

► II. Application—4 lipid



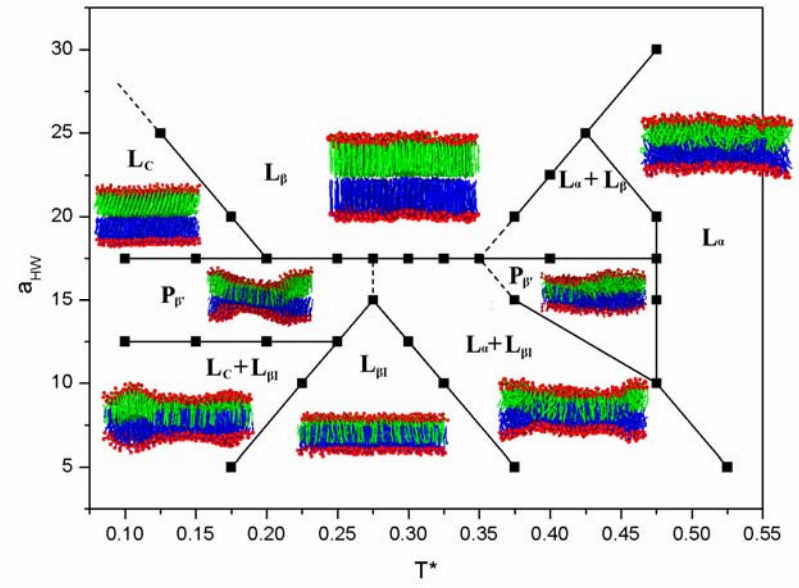
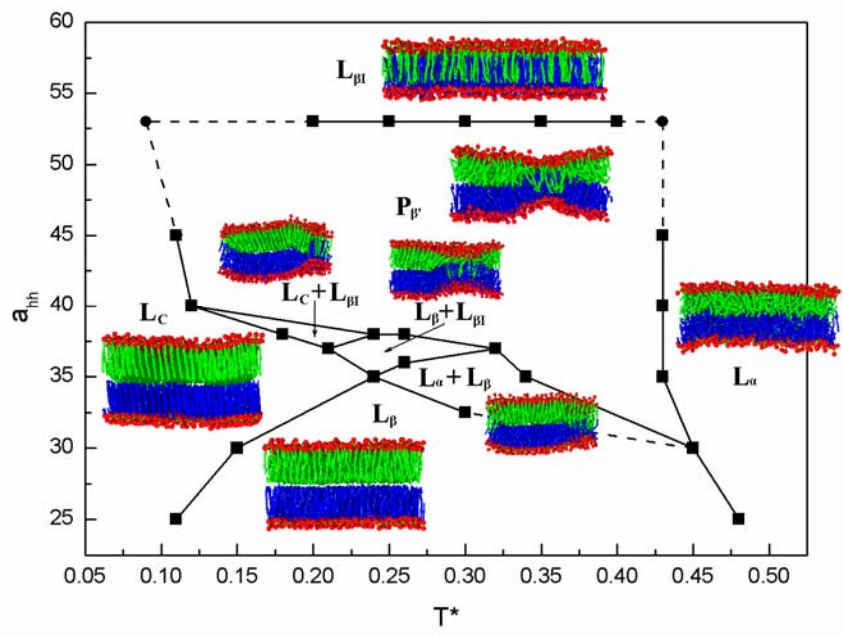
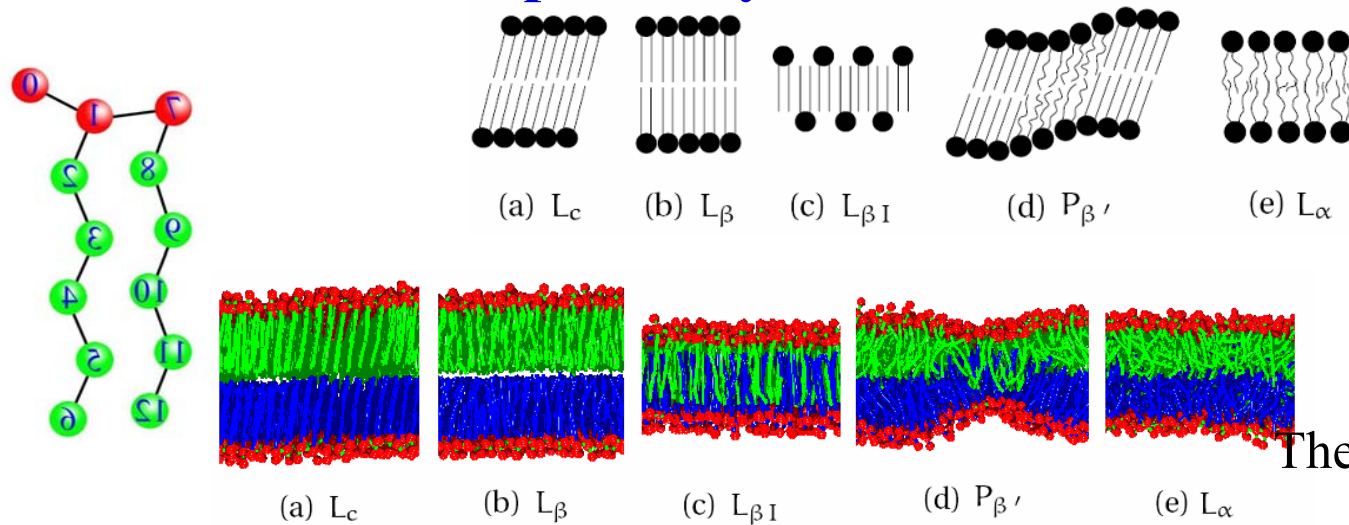
DPPC



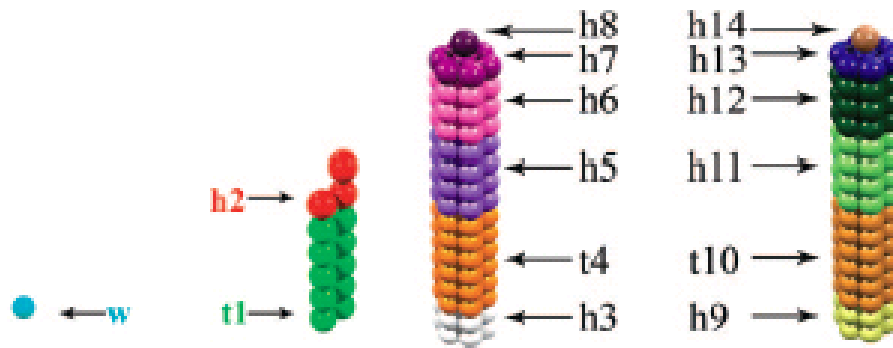
Our interest

A detailed understanding of the gel phase and their relevant transitions is therefore of great biological interest. On the other side, the lipid molecules are restricted in the membrane plane, which makes the lipid bilayer a perfect model for studying two-dimensional phase transition.

Phase Behavior of Lipid Bilayers

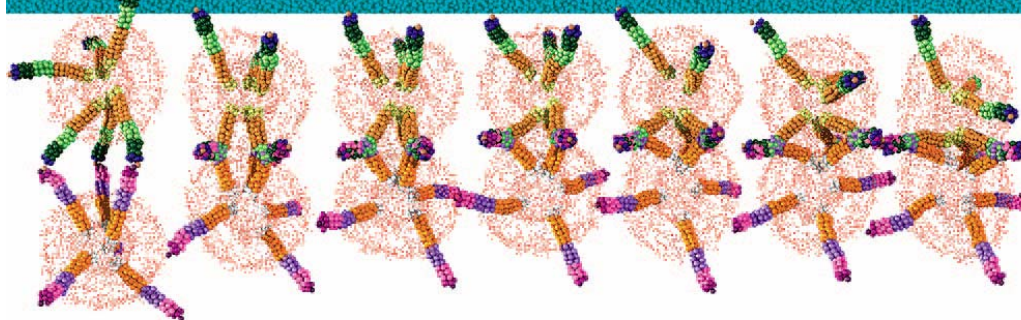
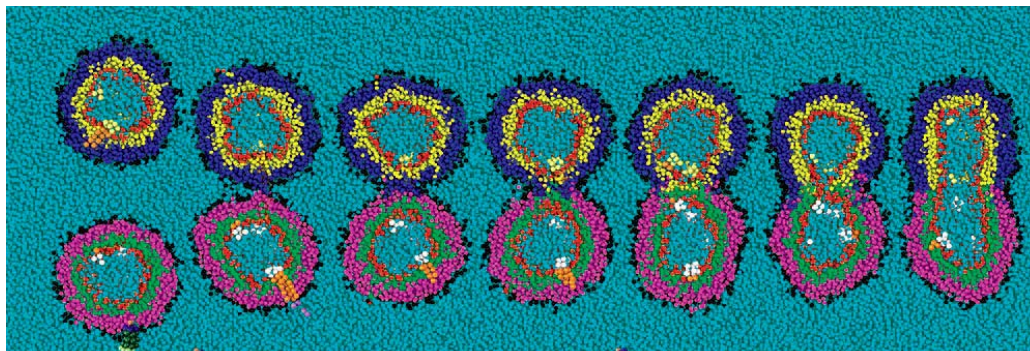


➤ Protein-Mediated Vesicle Fusion



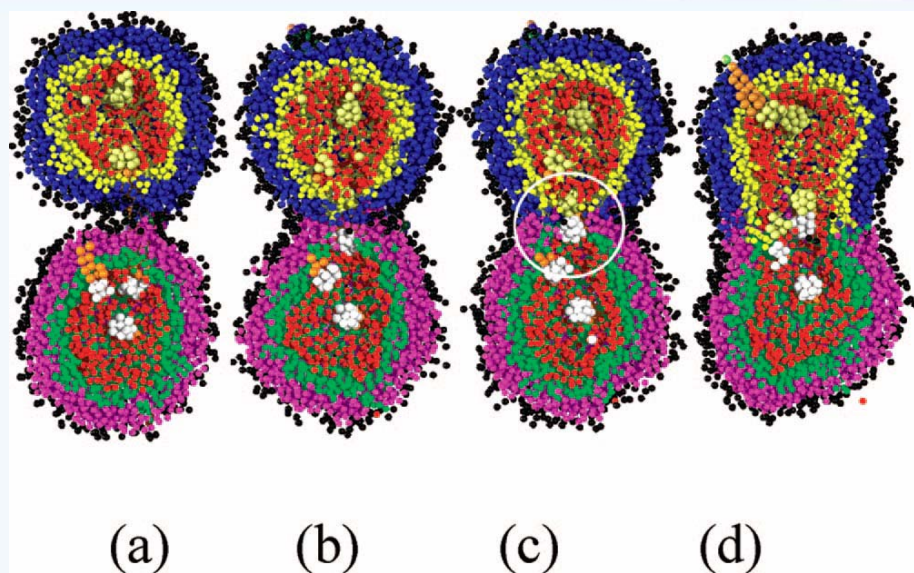
(a) water (b) lipid (c) t-protein (d) v-protein

a simple model system to
mimic the complicated
protein-mediated vesicle
fusion involving extensive and
cooperative molecular
rearrangement!!!

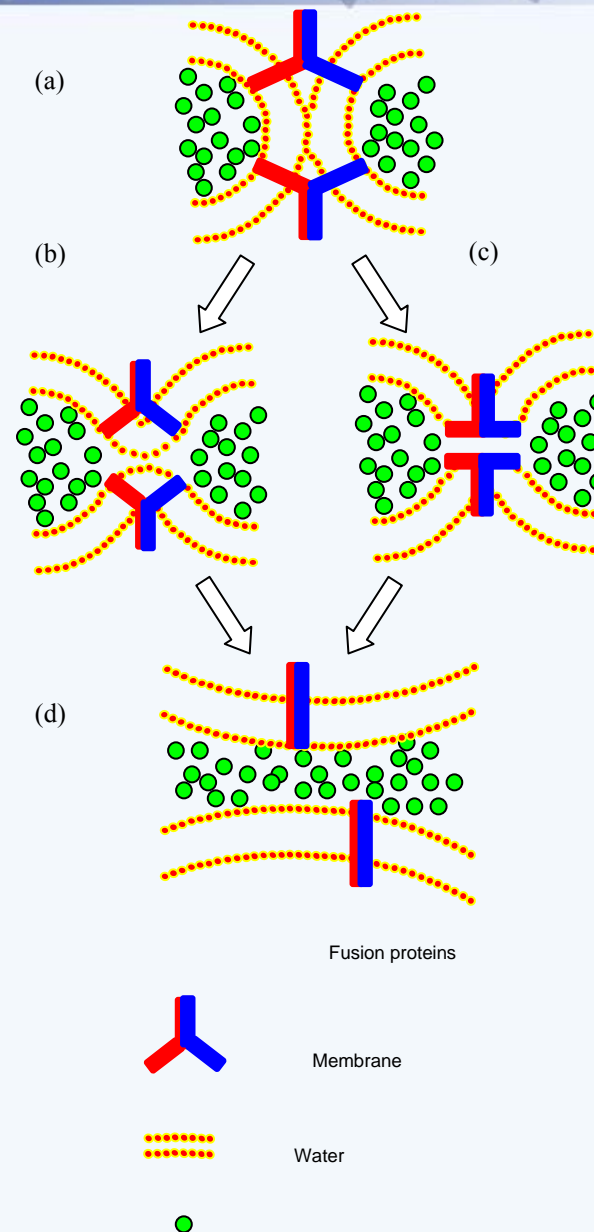


(a) (b) (c) (d) (e) (f) (g)

Pathway I of protein-mediated vesicle fusion (scaffold model):
(a) starting state; (b) outer leaflets contact; (c) stalk; (d) inner leaflets contact; (e) hemifusion diaphragm; (f) fusion pore appears; (g) full fusion



Pathway II of protein-mediated vesicle fusion (protein-pore model). The protein-lined pore (encircled) originates directly from the radially expanding stalk, which allows lipid and water to penetrate into it. Then, the pore expands to accomplish vesicle fusion: (a) outer leaflets contact; (b) stalk; (c) protein-lined fusion pore appears; (d) full fusion.





III. Application of DPD thermostat in--- 1. (NE)MD with LJ potential (spherical model system)

—2. (NE)MD with GB potential (non-spherical model system)



Application of DPD thermostat in 1. (NE)MD with LJ potential

Thermostats: to mimic the experiment conditions, to study the temperature dependent processes, to remove the excess heat generated by external fields

DPD thermostat and soft conservation potential are completely independent!!

DPD thermostat {

- stochastic, a large simulation time step
- conserve both the global and local momentums so that hydrodynamic behavior
- Galilean invariance, the solution of the thermal motion from the total motion is not necessary in NEMD

well-known thermostats : Gaussian, Nosé-Hoover, Langevin

great applications in studying the dynamics of various complex fluids under non-equilibrium conditions!!!!



Stochastic thermostats in NEMD

Langevin Eq

$$m_i \ddot{\mathbf{r}}_i = \bar{\mathbf{F}}_i - \zeta \dot{\mathbf{r}}_i + \bar{\mathbf{F}}_i^{rand}$$

Langevin PBT

$$m_i \ddot{\mathbf{r}}_i = \bar{\mathbf{F}}_i - \zeta (\dot{\mathbf{r}}_i - \gamma r_{iz} \bar{\mathbf{i}}) + \bar{\mathbf{F}}_i^{rand}$$

Langevin YZ

$$m_i \ddot{r}_{iy} = F_{iy} - \zeta \dot{r}_{iy} + F_{iy}^{rand}$$

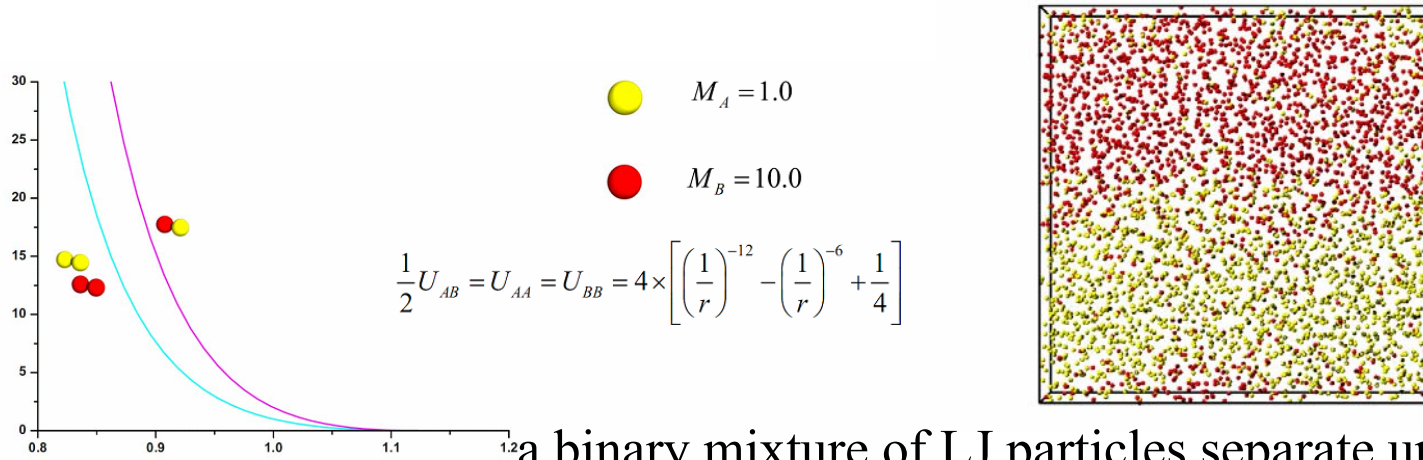
$$m_i \ddot{r}_{iz} = F_{iz} - \zeta \dot{r}_{iz} + F_{iz}^{rand}$$

DPD

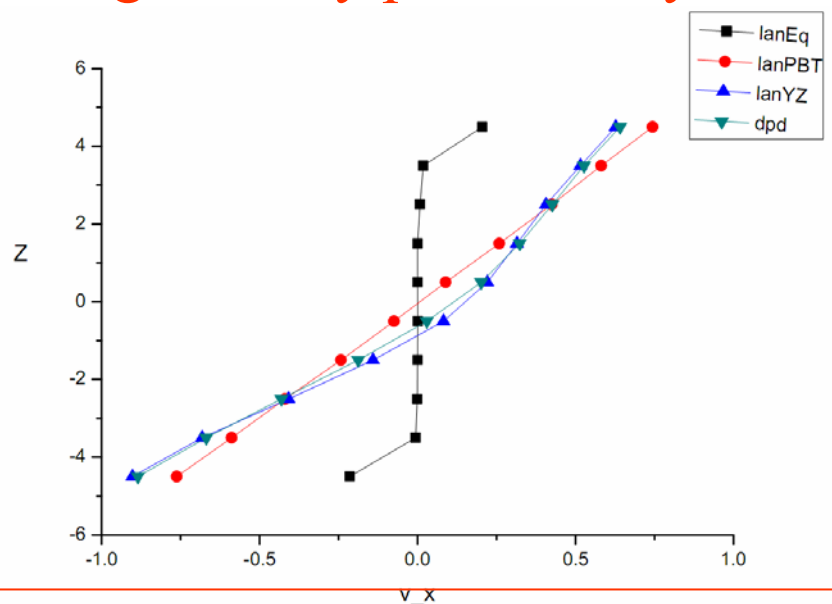
$$m_i \ddot{\mathbf{r}}_i = \bar{\mathbf{F}}_i + \bar{\mathbf{F}}_i^D + \bar{\mathbf{F}}_i^R$$

$$\bar{\mathbf{F}}_i^D = \sum_{i \neq j} \bar{\mathbf{F}}_{ij}^D, \bar{\mathbf{F}}_{ij}^D = -\zeta w^D(r_{ij}) (\hat{\mathbf{r}}_{ij} \cdot \dot{\hat{\mathbf{r}}}_i) \hat{\mathbf{r}}_{ij}$$

$$\bar{\mathbf{F}}_i^R = \sum_{i \neq j} \bar{\mathbf{F}}_{ij}^R, \bar{\mathbf{F}}_{ij}^R = \sigma w^R(r_{ij}) \theta_{ij} \hat{\mathbf{r}}_{ij}$$

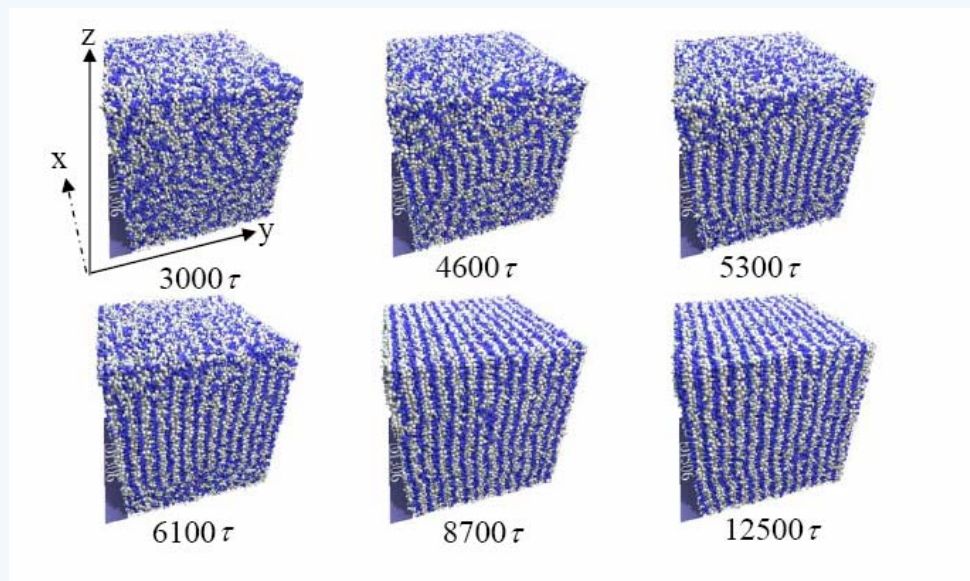
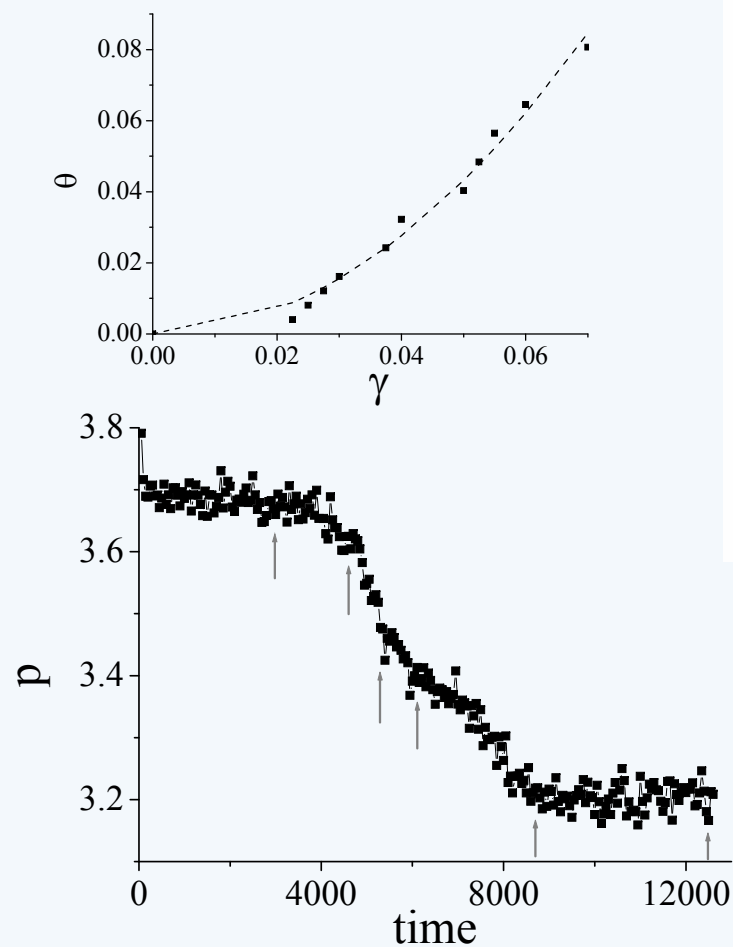


a binary mixture of LJ particles separate under shear flow
 the streaming velocity profile of system should not be linear!!!



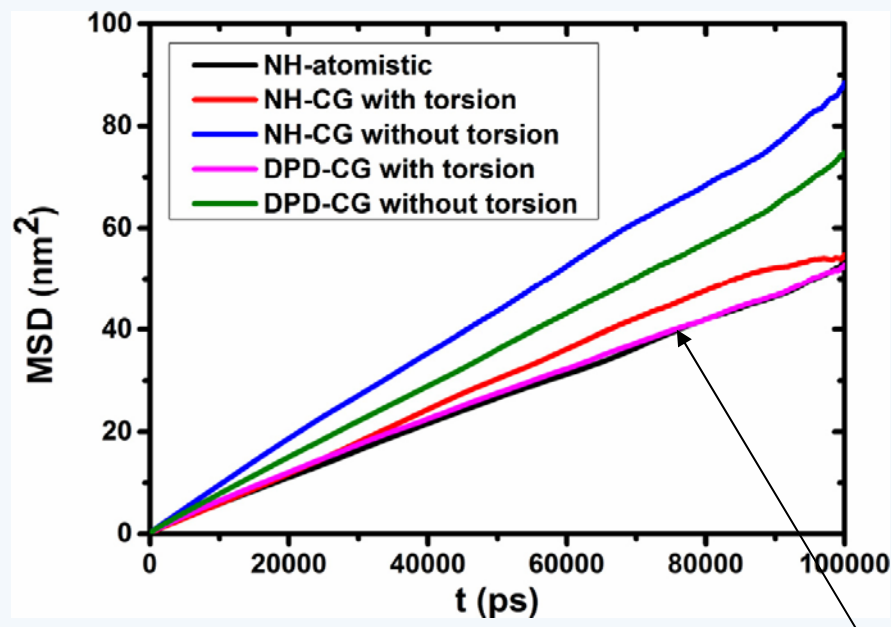
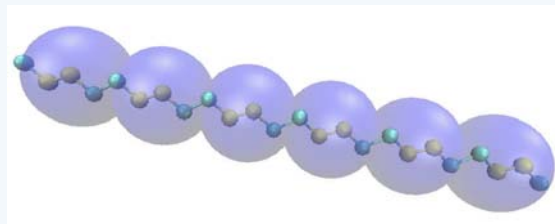
DPD thermostat is an ideal thermostat for NEMD simulations since it avoids profile biasing of NEMD simulation in a very natural and simple way, and thus suitable to study nonlinear phenomena in the nonequilibrium systems.

● DPD thermostat in NEMD of diblock copolymer melts

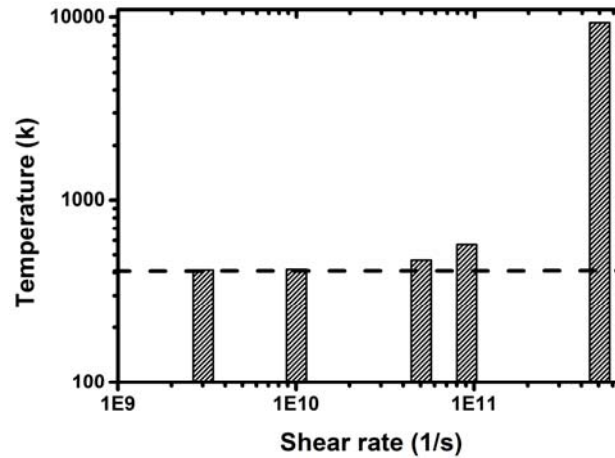


HX GUO , *J. Chem. Phys*, 127(5),054902-1-10 (2007); 125(21), 214902-1-9 (2006)
J. Chem. Phys, 124(5), 054902-1-11 (2006)

trans-1,4-PB

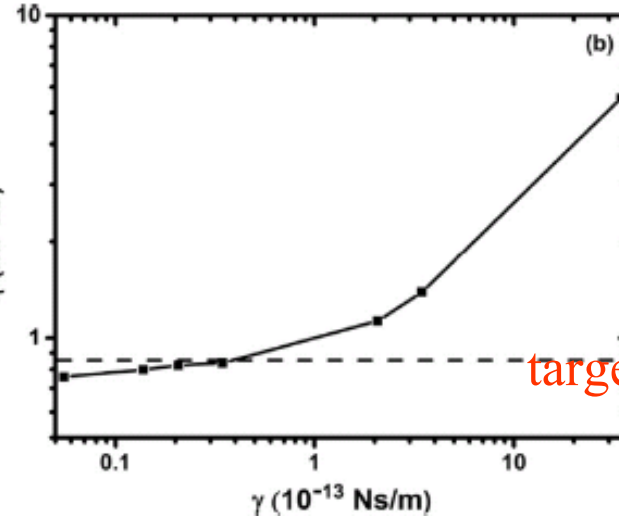
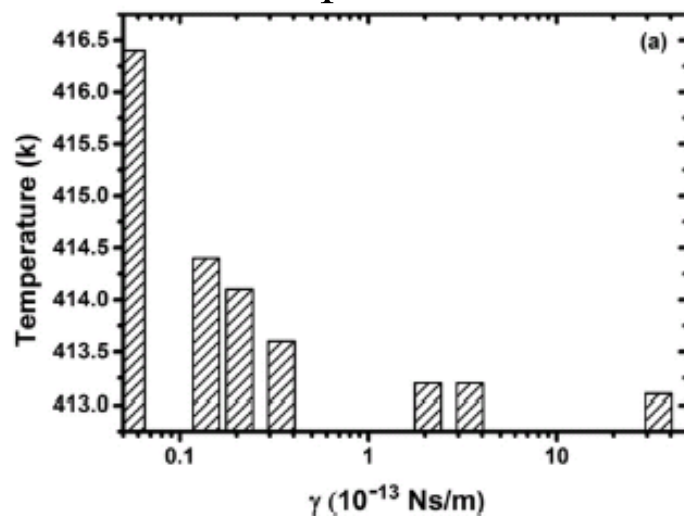


a standard DPD thermostat to remedy the removed degrees of freedom and reduced friction during CG and CG model can completely mimic the dynamics of the atomistic model



target temperature 413K

with the dissipation force factor of $\gamma = 0.056 \cdot 10^{-13}$ Ns/m



target shear viscosity

as a function of γ at a typical moderate shear rate of $1 \cdot 10^{10}$ 1/s ($Wi = 31.8$)

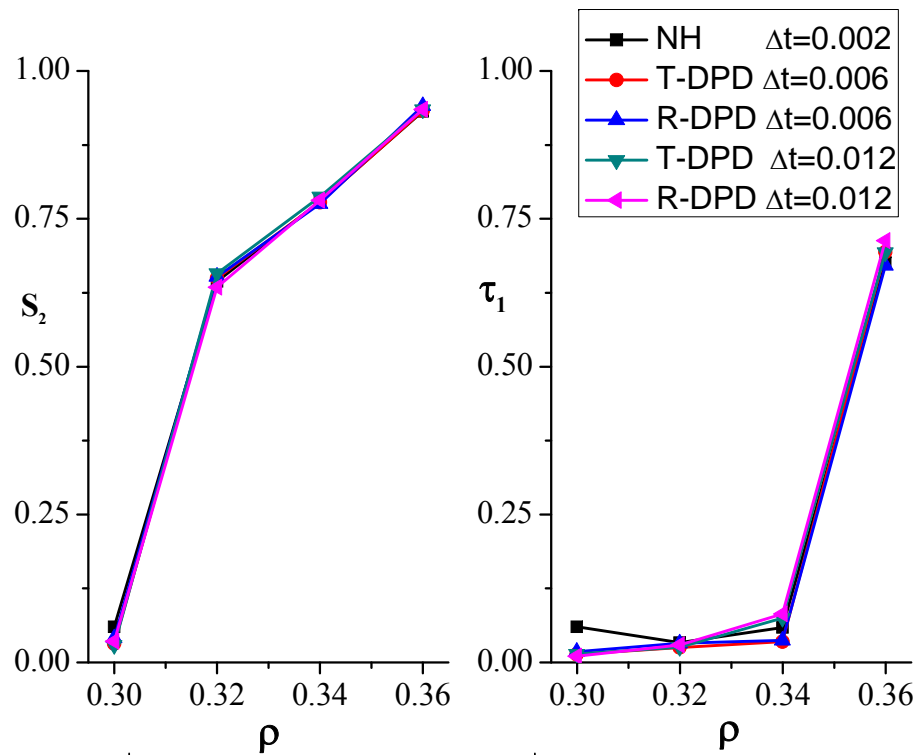
standard DPD thermostat approach is applicable to the steady shear flow simulation at low and moderate rates and the relevant dissipative factor derived by matching diffusion coefficients can only quantitatively reproduce the shear viscosity at low shear rates.



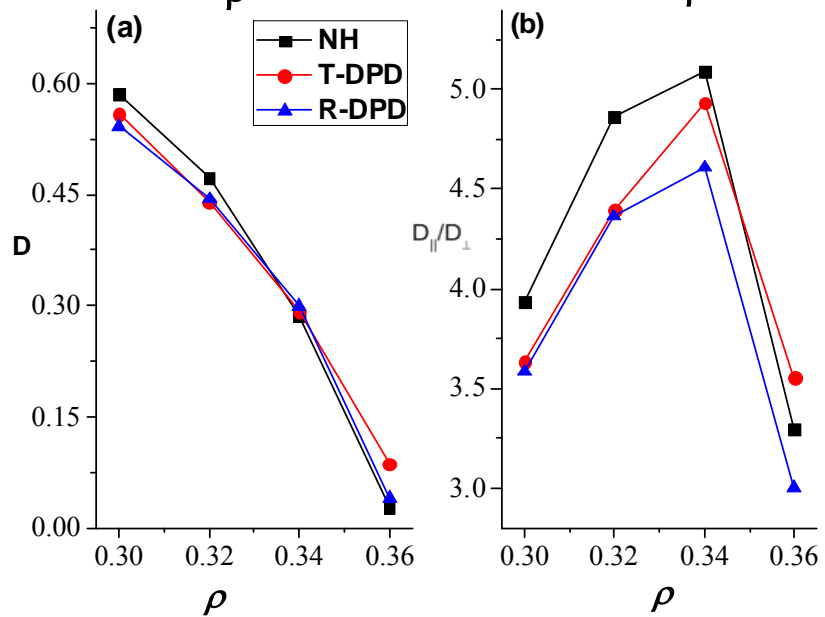
➤ Application of DPD thermostat in--- 2. (NE)MD with GB potential (non-spherical model system)

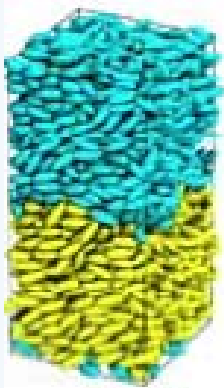
- Translational DPD (T-DPD) Thermostat
- Rotational DPD (R-DPD) Thermostat

T=0.95	$\Delta t=0.002$	$\Delta t=0.006$		$\Delta t=0.012$	
TEM	NH	T-DPD	R-DPD	T-DPD	R-DPD
0.30	0.949 ± 0.012	0.946 ± 0.019	0.948 ± 0.019	0.935 ± 0.019	0.946 ± 0.019
0.32_{ρ}	0.949 ± 0.013	0.946 ± 0.017	0.948 ± 0.018	0.937 ± 0.018	0.950 ± 0.020
0.34	0.949 ± 0.017	0.949 ± 0.019	0.954 ± 0.019	0.940 ± 0.018	0.930 ± 0.019
0.36	0.949 ± 0.018	0.949 ± 0.019	0.948 ± 0.019	0.945 ± 0.019	0.942 ± 0.020



orientational order parameter S_2
 translational order parameter τ_1

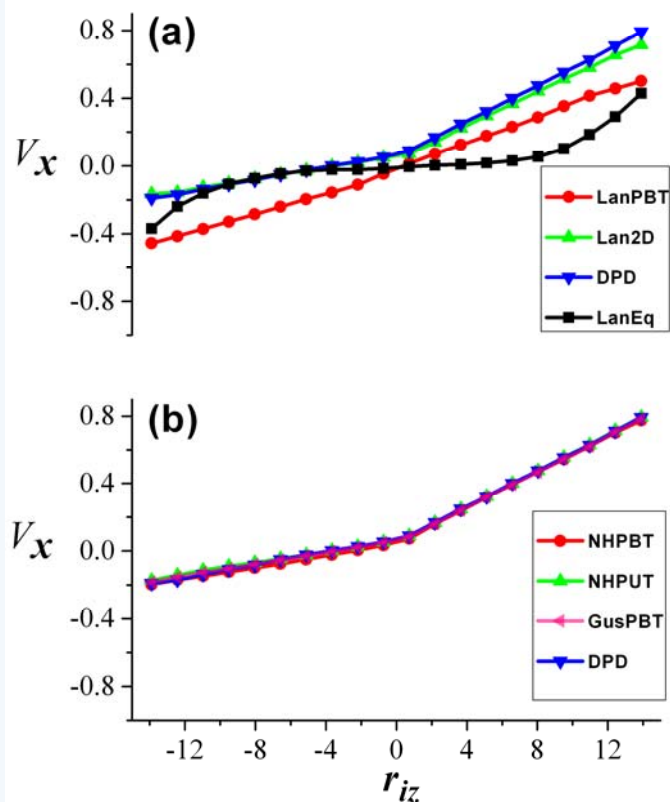




a binary mixture of LJ particles separate under shear flow

the mass ratio of particle B to A is set to 10

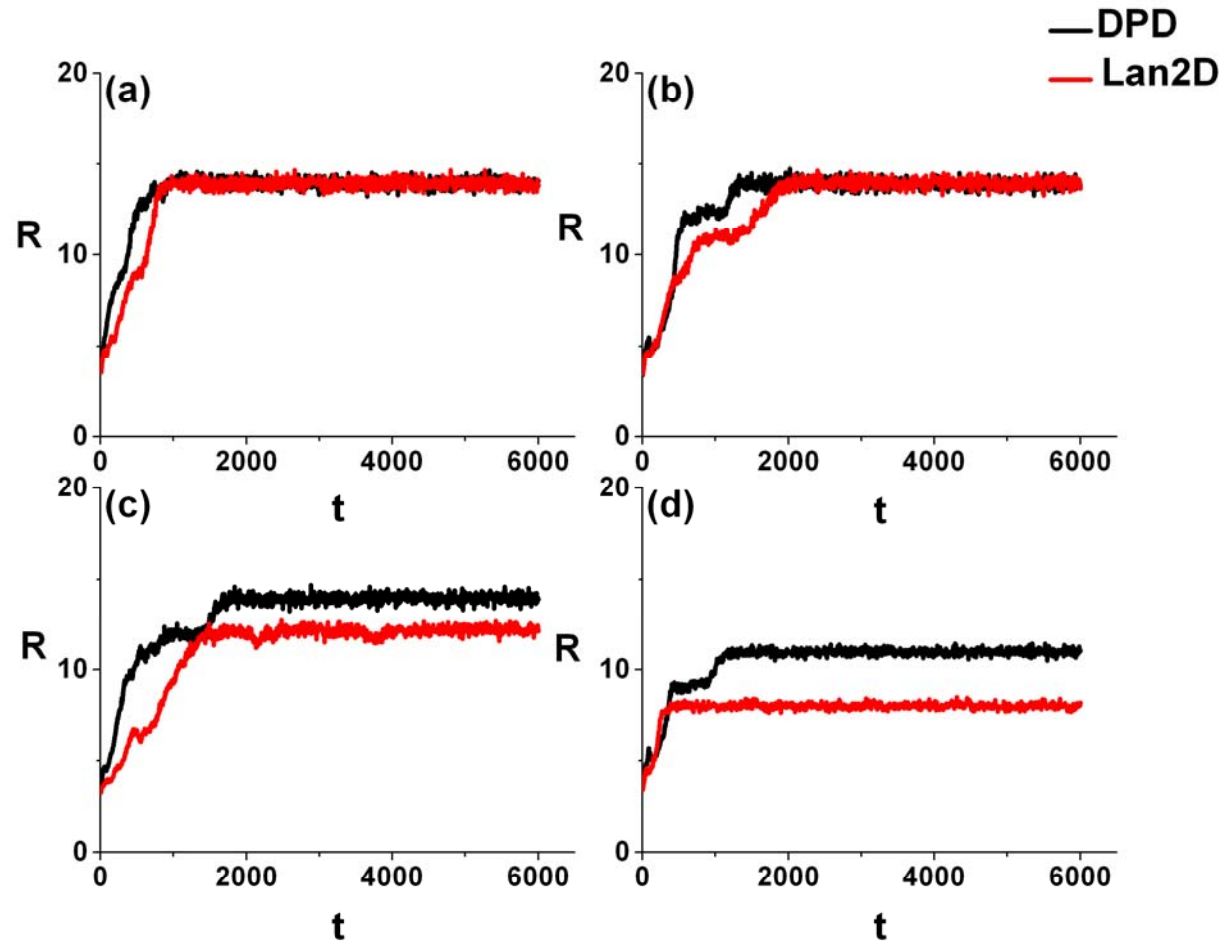
streaming velocity profile should not be linear!!!



Lan2D or DPD thermostat,
the streaming velocity profile displays the
expected broken line feature

GusPBT and NosPBT: we assumed
the streaming velocity profile linear
which conflicts with the above
simulation results, these two thermostats
are not reliable.

NosPUT and DPD thermostats works
without linear profile hypothesis
and can achieve the correct streaming
velocity profile.



DPD thermostat conserves both global and local moments of system,
it wouldn't screen hydrodynamic interactions
promotes the phase separation.



➤ Application of DPD thermostat in--- 2. (NE)MD with GB potential (non-spherical model system)

- translational T-DPD and rotational R-DPD thermostats : used in the GB system independently and both can achieve the thermostating effects.
- a time step of $\Delta t = 0.012$ and a dissipative coefficient of $\zeta t = 0.01-0.1$
- avoids profile biasing of NEMD simulation
- unscreen hydrodynamic interactions



IV. Efficient and large-scale dissipative particle dynamics simulations on GPU

speeding up the computation of DPD to study large-scale systems at reasonable computational cost is one of the important subjects in computational chemistry and computational material science!!!



IV. Efficient and large-scale dissipative particle dynamics simulations on GPU

- ✓ a systematic framework for implementing DPD on GPU is still lacking
- takes advantage of the superior computational performance of GPUs
allows for highly efficient and large-scale DPD simulations**
- ✓ To develop highly efficient and large-scale GPU-based DPD,
- the scheme for GPU implementation should also be designed and optimized according to the nature of DPD simulation technique, optimizing techniques specific for DPD should be included.**

Rozen et al. have developed GPU-based DPD, suffer the low speed of the scattered data access in the linked lists



Implementation of DPD simulation on a GPU

- (a) Set up the initial conditions for the simulation
- (b) Perform the first stage of numerical integration during a single δt
- (c) Update the neighbor list.
- (d) Perform the second stage of each numerical integration step.

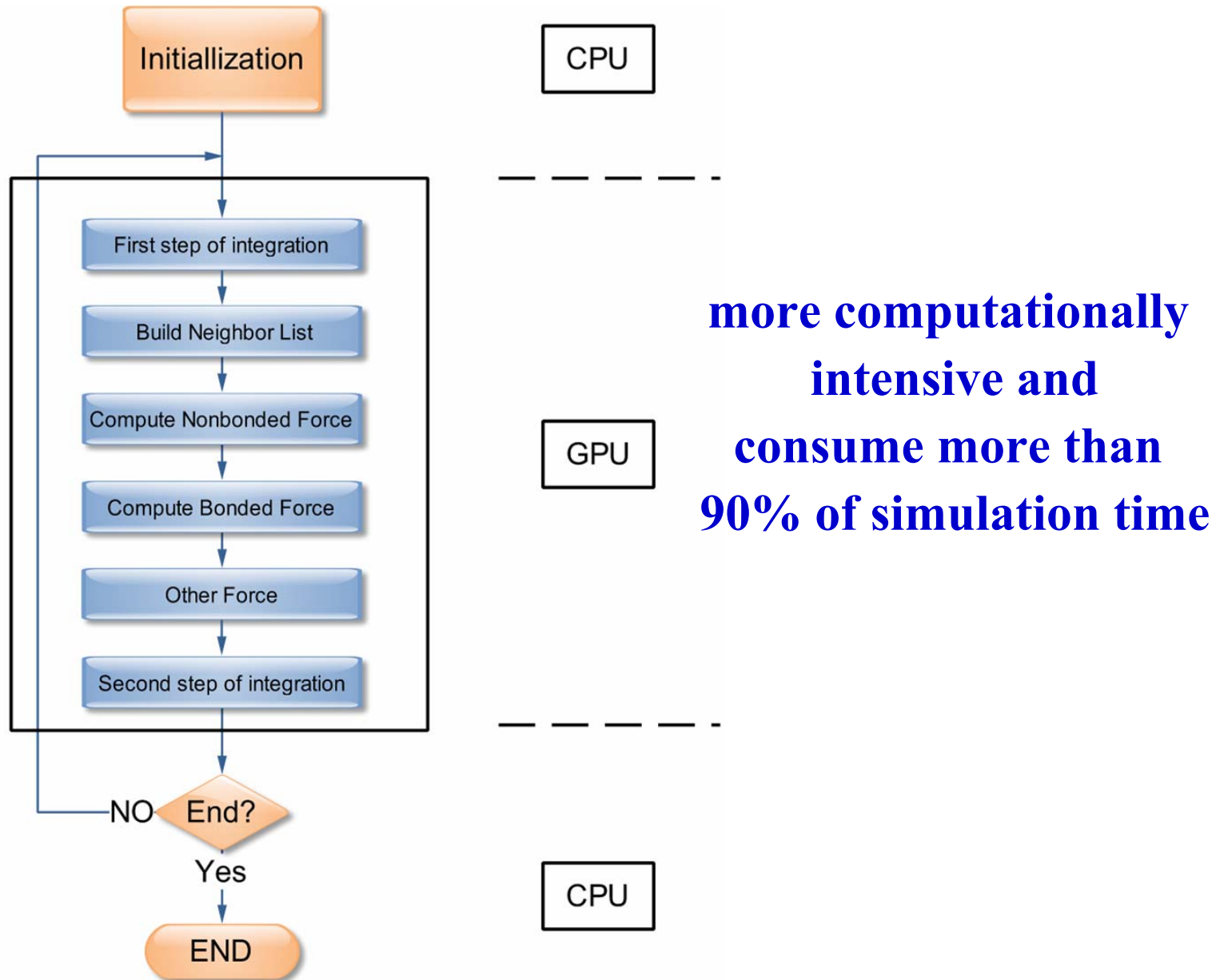
Firstly, the new non-bonded interactions between any pair of DPD particles are evaluated

Then the new bonded interactions such as harmonic bond forces and angle forces are calculated

Finally, the velocity of each particle is updated

(e) Loop over steps (b)-(d) until the simulated system reaches an equilibrium state or long enough to study nonequilibrium phenomena

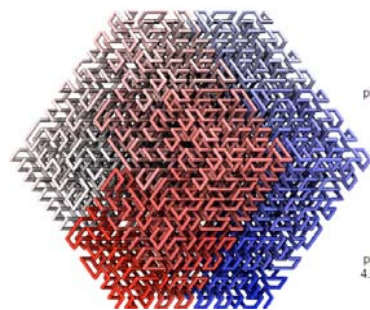
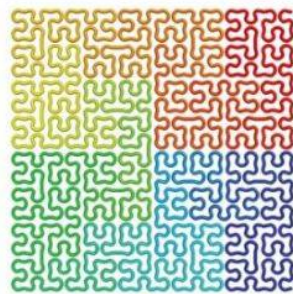
Implementation of DPD simulation on a GPU



some optimizations on *Neighbor list construction*

(i) the neighbor list is constructed without a surrounding shell and updated in each step.

(ii) a modified particle reordering technique is introduced to improve the device memory access efficiency for generating neighbor list



Random
pair: 50.4 ms
—
Sorted
pair: 12.3 ms
4.2x speedup!

Storing the data as cell index

1. Effective

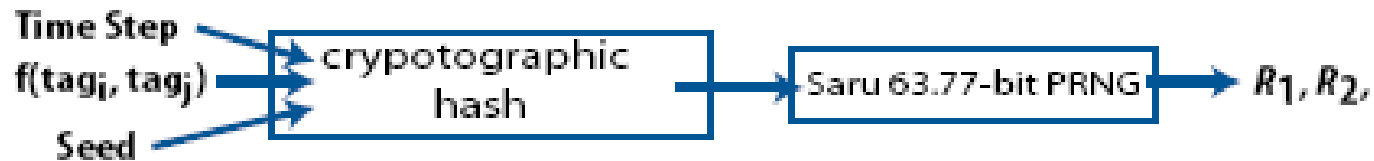
2. Easy to implementation

(iii) we loop over all particles directly during the neighbor list generation without using shared memory and looping over cells first



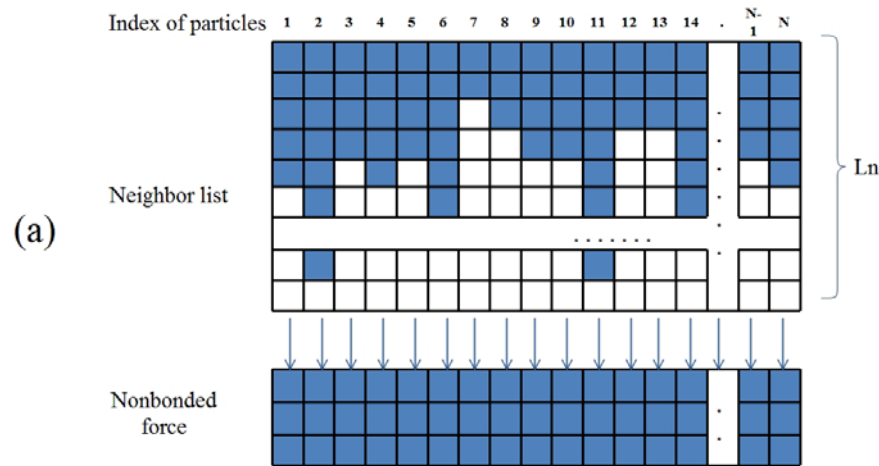
Non-bonded force computation

the uniformly distributed random numbers generated by *sarua* (*novel hash-based PRNG*) are transformed into the normally distributed ones by the Box-Muller method.



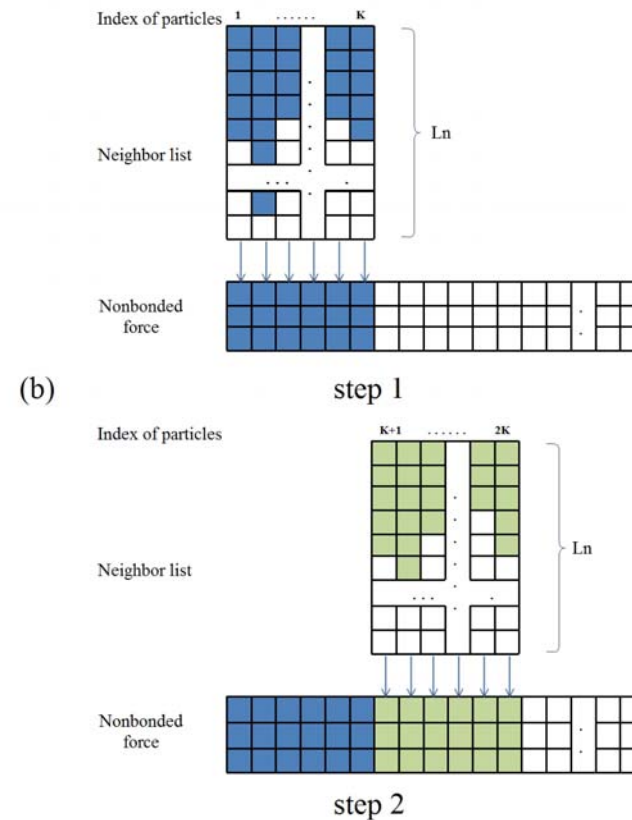
In this way, not only the same stochastic forces can be obtained for the $i-j$ and $j-i$ pairs, but also less GPU time is taken.

Algorithms for large-scale simulations



(a) In the normal GPU-based non-bonded force calculations, a complete neighbor list matrix with a fixed size is used.

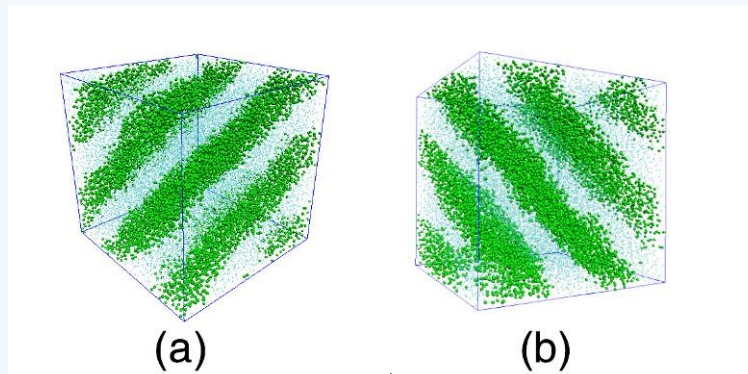
Tesla C2050 GPU
with 3 Gigabyte (GB)
device memory



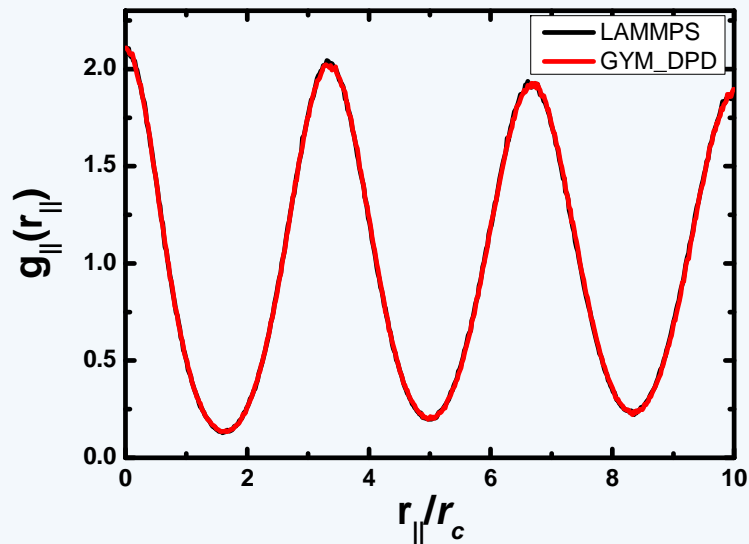
(b) using a novel **divide-and-conquer** algorithm to reduce the memory requirement

→ $N_{max} \sim 1.5 \cdot 10^7$

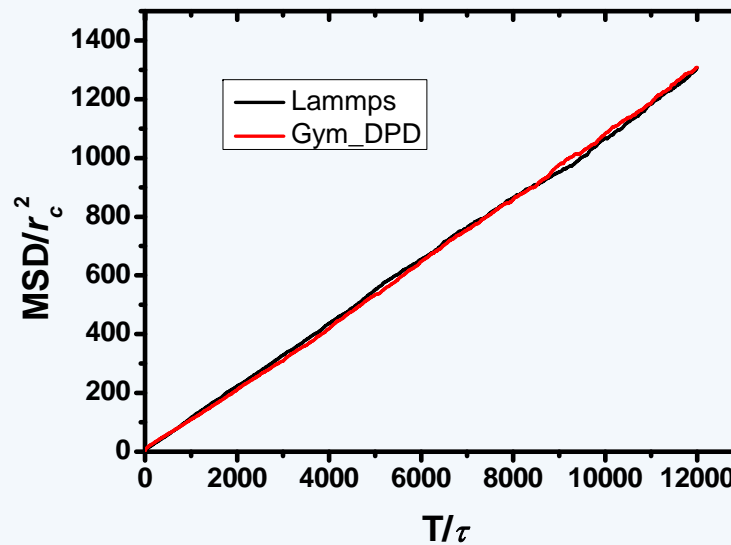
Accuracy



(a) C2050 GPU (b) single E5645 CPU



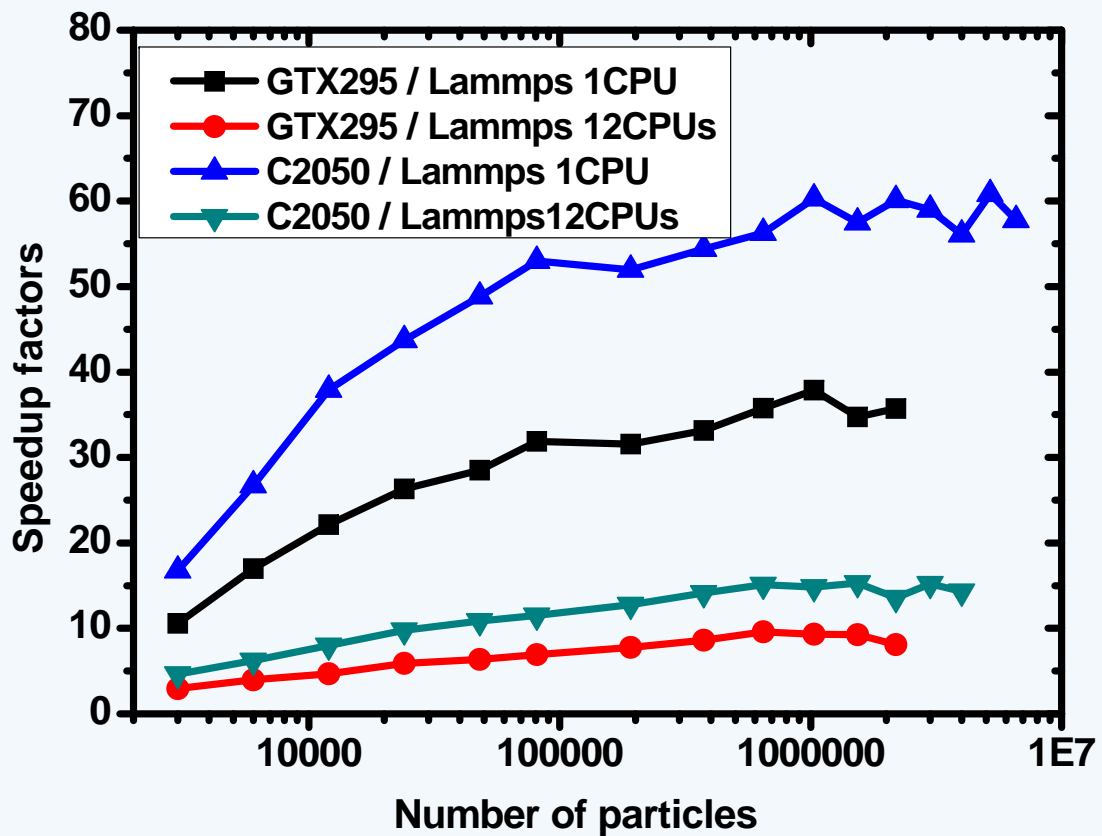
$g_{||}(r_{||})$ profiles



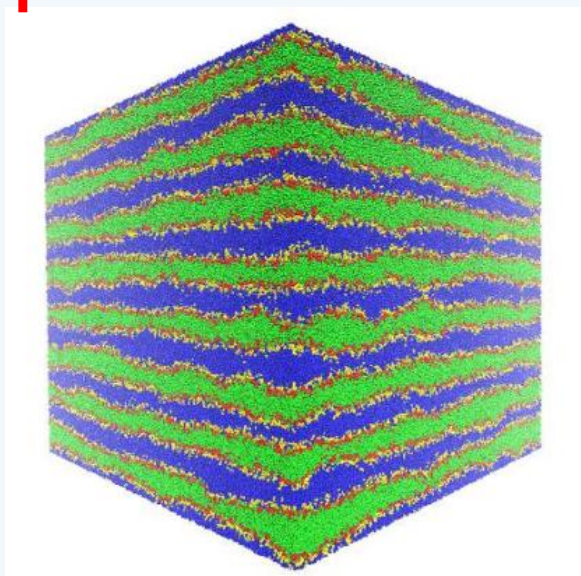
MSD



speedup

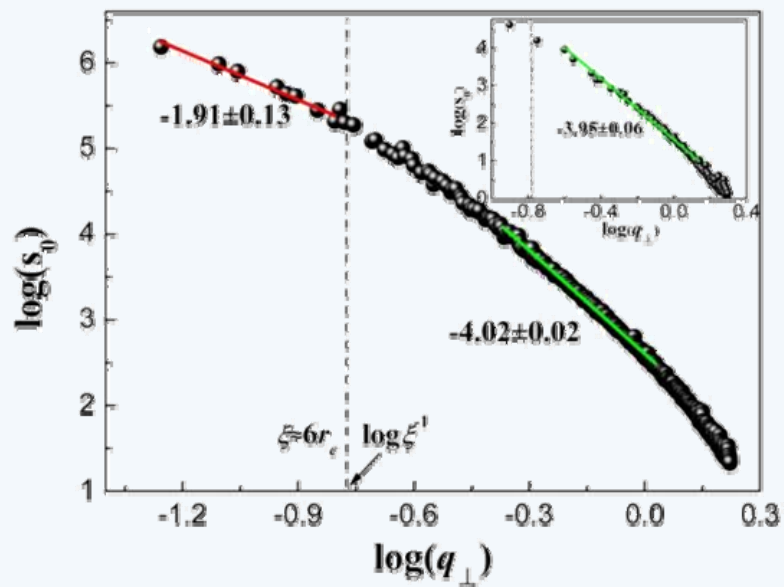


Application:



(144*144*160 11, 059, 200 beads)

Polymer, 54:2146-2157(2013)





1. a complete implementation for the highly efficient and large-scale DPD simulation on a GPU.
2. This implementation is designed and optimized according to the nature of DPD simulation technique and fully takes advantage of the computational power of current GPUs. the GPU-based implementation can predict the results correctly and provide nearly 60 times speedup over LAMMPS on a single Central Processing Unit (CPU) core.
3. using a novel divide-and-conquer algorithm to reduce the memory requirement in simulation, our implementation has the capability to perform large-scale DPD simulations with some ten millions of particles on a single current GPU.



V. Conclusion

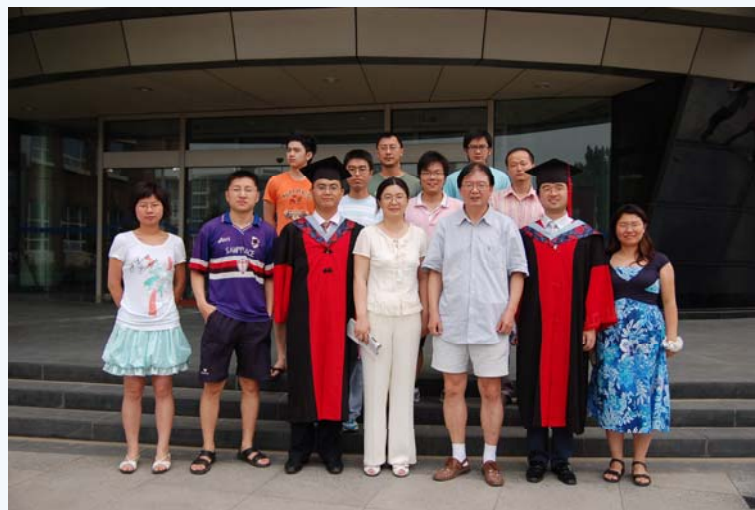
- An efficient mesoscopic simulation technique
- An efficient thermostat: local, hydrodynamic conserved, Galilean invariant



Acknowledge



Group members:
Dr. Huang, manxia
Dr. Bai, Zhiqiang
Dr. Liu, Xiaohan
Dr. Zhang, Zunmin
Dr. Yang, Keda
Dr. Wu, Shaogui





Thank You

& Questions?